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Many-body dissipative particle dynamics simulations of micellization of sodium alkyl sulfates†

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We present a study of micelle formation in alkyl sulfate surfactants using the simulation method of many-body dissipative particle dynamics (MDPD). We parametrise our model by tuning the intermolecular interactions in order to reproduce experimental values for the chemical potential and density at room temperature. Using this approach, we find that our model shows good agreement with experimental values for the critical micelle concentration (CMC). Furthermore, we show that our model can accurately predict CMC trends, which result from varying properties such as surfactant tail length and the salt concentration. We apply our model to investigate the effect of aggregation number on various micellar properties, such as the shape of individual micelles and the fraction of bound counterions. We show that micelles become aspherical at large aggregation numbers, in line with experimental predictions, and that longer tail surfactants are generally more spherical at all aggregation numbers compared to those which are shorter. We find excellent agreement between our simulations and experimental values for the degree of counterion binding, a factor that is crucial to accurately studying micellar shape, but one that is typically overlooked in the existing literature.

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1 Introduction

Surfactants are key components in a huge variety of everyday products, including detergents and personal care items. Surfactants are surface-active amphiphilic molecules that are capable of forming micelles in solution when the concentration is above a critical micelle concentration (CMC). At higher concentrations, lyotropic liquid crystal phases can be produced,^{1–3} however, most commercial products are at concentrations in the micellar range: the phase region of interest in this article.

One of the most widely studied surfactants is the anionic surfactant sodium dodecyl sulfate (SDS), which is present in a large range of products such as shampoos and cleansers. Due to its widespread use in commercial products, there has been a great deal of experimental research dedicated to studying the size and shape of SDS micelles,^{4–15} as well as determining the CMC value.^{16–19} Therefore, the creation of an accurate model for studying and understanding SDS aggregation behaviour is of great interest, particularly to aid the design of commercial

products. We developed our model with the primary objective of studying SDS. However, this model can be easily extended to other surfactants. Consequently, we also investigate the variation in the length of the surfactant tail, allowing us to also study sodium octyl sulfate and sodium hexadecyl sulfate; together with the effects of additional salt.

Simulating surfactant solutions is challenging, primarily because of the time scales involved in micelle formation. While molecular dynamics (MD) simulations for SDS micelles can be found in existing literature,^{20–25} these studies are typically limited to simulating single preassembled micelles. These limitations motivated the development of coarse-grained simulation methods, such as dissipative particle dynamics (DPD).²⁶ DPD is computationally faster than traditional MD techniques, which opens up the possibility of simulating larger length and time scales. This speed-up is partly because molecules are simulated using ‘beads’, which represent groups of atoms as opposed to simulating individual atoms, but is also due to the form of forces which represent the intermolecular interactions between beads. The forces between beads are designed in DPD to enhance diffusion, leading to shorter equilibration periods. DPD has been shown across the literature to be a highly useful tool for modelling surfactant systems, in particular for predicting critical micelle concentrations^{27–29} and for studying micellar shape and size.^{28,30,31} DPD has also been shown to be useful for studying systems at higher concentrations, including surfactants which form liquid crystal structures.^{31–34}

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One of the drawbacks of the DPD method is that there is not yet a force field which can be readily applied to any molecule of choice. There are several existing parametrisation attempts in the literature,^{26,35–47} including for SDS,^{28,29,32} but these approaches are typically limited to small subsets of molecules. These parametrisation attempts have also had varying success in being able to make qualitative predictions, which agree with experimental data since, for example, the aggregation number is typically underestimated in existing models.^{28,29}

Since its inception, traditional DPD methods have been extended and modified to make them more versatile and applicable to a greater range of systems. This includes the adaptation of many-body dissipative particle dynamics (MDPD),⁴⁸ which we will use in this work. MDPD is a variation of standard DPD, in which the conservative interaction force between beads is altered from being purely repulsive to one with long-range attractive and short-range repulsive components. This makes the potential more comparable to a molecular dynamics pair potential, such as the Lennard-Jones potential. Adapting the potential in this way has benefits, including the ability to simulate coexisting liquid and vapour phases. It also allows for the possibility of reproducing realistic chemical potentials since it allows negative potential energies to be calculated. In Section 2 of this article we highlight why this makes MDPD an attractive choice for studying micelles.

This article is structured as follows. We first present a brief overview of the many-body dissipative particle dynamics method, before presenting our approach to parametrising surfactant simulations. We then calculate the CMC of SDS molecules and show how this can be influenced by varying the surfactant tail length and the addition of salt. Finally, we calculate the aggregation number and present a short study of how the aggregation number influences micelle shape.

2 Many-body dissipative particle dynamics

2.1 Background and overview

In standard DPD, nonbonded beads are acted upon by three forces, which are typically referred to as the conservative, random, and dissipative forces. For the conservative force, the interaction between beads i and j is given by

$$\mathbf{F}_{ij}^C = \begin{cases} a_{ij} \left(1 - \frac{r_{ij}}{r_C}\right) \hat{\mathbf{r}}_{ij} & \text{for } r_{ij} < r_C \\ 0 & \text{for } r_{ij} \geq r_C \end{cases} \quad (1)$$

where $0 < a_{ij}$ and can vary between different bead types. r_{ij} is the distance that separates the beads and r_C is the interaction cut-off, commonly taken to be $r_C = 1$.

The remaining two non-bonding forces are the dissipative (or drag) force \mathbf{F}_{ij}^D and a random pairwise force \mathbf{F}_{ij}^R , which act as the simulation thermostat. The dissipative force is calculated using

$$\mathbf{F}_{ij}^D = -\gamma \omega^D(r_{ij})(\hat{\mathbf{r}}_{ij} \cdot \mathbf{v}_{ij}) \hat{\mathbf{r}}_{ij}, \quad (2)$$

and the random force using

$$\mathbf{F}_{ij}^R = \sigma \omega^R(r_{ij}) \zeta_{ij} \hat{\mathbf{r}}_{ij} \Delta t^{-1/2}, \quad (3)$$

where ω^D and ω^R are weight functions, γ is a friction coefficient and σ is the noise amplitude. \mathbf{v}_{ij} is the velocity between beads i and j , Δt is the time step and $\zeta_{ij}(t)$ is a randomly fluctuating Gaussian variable with zero mean and unit variance. It was shown by Español and Warren⁴⁹ that the relationship between the weight functions and amplitudes must obey the following relations

$$\omega^D = [\omega^R]^2 \quad (4)$$

$$\sigma^2 = 2\gamma k_B T, \quad (5)$$

where k_B is the Boltzmann constant and T is the temperature. In this work, we set $k_B T = 1$ and set the values of constants to be $\sigma = 3$ and $\gamma = 4.5$.

Individual DPD beads can be chemically bonded to create coarse-grained molecules. The bonding force is calculated via the potential U_{ij}^B , which is represented as a harmonic spring using

$$U_{ij}^B = \frac{C}{2} \sum_j (r_{ij} - l_0)^2, \quad (6)$$

where l_0 is an equilibrium bond length and C is a constant. The molecules are also subject to angle constraints to maintain a realistic rigidity. These angle potentials take the form

$$U_{ijk}^A = \frac{D}{2} \sum_j (\theta_{ijk} - \theta_0)^2, \quad (7)$$

where D is a constant, θ_{ijk} is a bond angle between beads i, j and k , and θ_0 is an equilibrium bond which we set to $\theta_0 = 180^\circ$. For constants C and D we set $C = 150$ and $D = 5$ (DPD units).

In this paper, we choose to use an adaptation of DPD called many-body dissipative particle dynamics (MDPD). In MDPD, the form of the forces used is identical to the standard case for all except for the conservative force. The conservative force is altered to include both attractive and repulsive components, taking the form

$$\mathbf{F}_{ij}^C = \begin{cases} a_{ij} \left(1 - \frac{r_{ij}}{r_C}\right) + B(\rho_i + \rho_j) \left(1 - \frac{r_{ij}}{r_d}\right), & \text{if } r_{ij} < r_d, \\ a_{ij} \left(1 - \frac{r_{ij}}{r_C}\right), & \text{if } r_d < r_{ij} < r_C \\ 0, & \text{if } r_{ij} > r_C \end{cases} \quad (8)$$

where ρ are local densities for each particle, and are calculated as

$$\rho_i = \sum_{j \neq i} \frac{15}{2\pi r_d^3} \left(1 - \frac{r_{ij}}{r_d}\right)^2, \quad \text{for } r_{ij} < r_d. \quad (9)$$

Similarly to traditional DPD, the cutoff r_C is usually taken as $r_C = 1$ and the cut-off for the density dependent component r_d is taken as $r_d = 0.75$.⁵⁰ In contrast to standard DPD, we define $a_{ij} < 0$, with this term therefore acting as an inter-particle attraction. The constant B , on the other hand, takes a positive value ($0 < B$), producing a repulsive, density-dependant term.



The no-go theorem⁵¹ imposes the condition that B and r_d parameters are the same for all particle types and in this work we chose to make the common choice of setting $B = 25$ and $r_d = 0.75$.^{50,52,53}

2.2 Electrostatics

Charged systems are also subject to an additional electrostatic force \mathbf{F}_{ij}^E acting on the beads. In molecular dynamics simulations, it is typical to treat charged atoms as point charges. However, in standard DPD simulations, it is usually the case that researchers^{28,29,31,54} choose to use a smeared charge approach instead, where the charge is spread over a finite volume. Because of the soft-repulsive nature of the DPD conservative force, treating DPD beads as having a point charge can result in artificial ion-pair formation where oppositely charged ionic beads may collapse on top of each other, forming infinitely strong ion pairs. When treated as point charges, research⁵⁵ has shown that the percentage of artificial pairs formed decreases as the magnitude of the conservative force repulsion between beads increases (corresponding to an increasing value of $|a_{ij}|$). Therefore, at sufficiently high values of a_{ij} , it is suggested that one can still use point charges as opposed to requiring smeared approaches.

Due to the different form of the conservative force (eqn (8)) it is unknown whether smeared charges are required in MDPD simulations. We have not found any previous work investigating artificial ion pair formation in the existing MDPD literature. Therefore, before studying micelle solutions, we first present a short investigation into the behaviour of charges in MDPD in Section 4 of this article. It will be shown that we can safely treat our beads as point charges for studying surfactant aggregation. Therefore, the electrostatic force in the study of micellar solutions will be calculated using the standard Coulombic potential

$$U_{ij}^E = \frac{\Gamma}{4\pi} \frac{q_i q_j}{r_{ij}} \quad (10)$$

where q_i and q_j are the charges on two beads, $\Gamma = e^2/(k_B T \epsilon_r \epsilon_0 r_C)$ is a dimensionless electrostatic coupling parameter, where ϵ_0 is the permittivity of free space, ϵ_r is the relative permittivity of water, and e is the elementary charge.

2.3 Interaction parameters a_{ij}

Self-interaction parameters a_{ii} are often calculated for DPD simulations by aiming to reproduce experimental densities^{28,56} or compressibilities.^{29,38,55,57} In this work, we choose to define our self-interaction parameters by matching to experimental values for the density as well as chemical potentials of pure water and octane.

Various approaches have been proposed to calculate cross-interaction parameters a_{ij} , including matching to experimentally derived quantities such as partition coefficients⁴⁴ or solubility parameters.^{58,59} Other approaches include matching to quantities obtained from molecular dynamics simulations, such as the radial distribution.⁶⁰ In this work, we will base our parameterisation on reproducing experimental activity coefficients at

infinite dilution γ^∞ (IDACs), an approach which has also been used by various previous authors.^{29,38,57,61}

The chemical potential can be split into two components⁶² for computational purposes:

$$\mu = \mu^{\text{excess}} + \mu^{\text{ideal}} \quad (11)$$

where the ideal part is calculated using $\mu^{\text{ideal}} = k_B T \ln \rho \Lambda^3$ where Λ is the thermal de Broglie wavelength and ρ is the number density. We calculate the excess potential energy μ^{excess} via the Widom insertion method, using the equation

$$\mu^{\text{excess}} = -k_B T \ln \left\langle \exp \left(-\frac{\Delta U}{k_B T} \right) \right\rangle, \quad (12)$$

where the change in potential energy ΔU due to the insertion of a single bead is measured. The average over multiple insertion attempts throughout the simulation domain can be used to calculate the excess chemical potential per bead.

The total chemical potential of substance A in solution can be represented in terms of its activity coefficient γ_A ⁶³ as

$$\mu_A = \mu_A^\ominus + k_B T \ln(\gamma_A x_A) \quad (13)$$

where μ_A^\ominus is the chemical potential of pure component A, μ_A is the chemical potential of A in the solution (where the units of chemical potential μ are energy per molecule), and x_A is the mole fraction of A in the solution. Supposing that the solution consists of A at infinite dilution in a solvent of type B, then combining eqn (11) and (13) leads to the following expression

$$\ln(\gamma^\infty) = k_B T \ln \left(\frac{\rho_B}{\rho_A} \right) + \mu_A^{\text{excess}} - \mu_B^{\text{excess}} \quad (14)$$

where the excess chemical potentials can be calculated using eqn (12). Note that a full derivation of eqn (14) can be found in the ESI.†

Since in previous works (using traditional DPD), the self-interactions are usually matched to density or compressibility, the individual chemical potentials μ are usually incorrect and do not match experimental values. Instead, the focus is on matching to the IDAC alone. However, in any case, due to the form of the conservative force (eqn (1)), in DPD it would be impossible to correctly reproduce the chemical potentials of pure components anyway. To correctly reproduce the chemical potentials of pure components, this will usually require negative μ^{excess} values (due to the negative chemical potential of most liquids). It is clear from eqn (12) that, in order to generate $\mu^{\text{excess}} < 0$, in general, the insertion due to a single bead should result in a negative change in the potential energy ($\Delta U < 0$). In liquids such as water, this is due to the attractive forces between water molecules.

However, since the conservative force in standard DPD is purely repulsive, this makes it impossible to generate these negative excess chemical potential values. This is one of the reasons we chose to use MDPD for our simulations. The presence of the attractive term in MDPD means that negative values of μ^{excess} can be generated, therefore making it possible to reproduce the correct pure chemical potential values. In this work, we match not only the experimental values for $\ln(\gamma^\infty)$, but



also the pure chemical potentials of the solute and solvent, and also the experimental densities.

3 Parameterisation approach

In this work, we focus on a DPD model for anionic surfactant sodium dodecyl sulfate (SDS). As discussed in Section 2, we base the parameterisation of the tail and water beads on experimental densities, chemical potentials, and IDACs. An illustration of the coarse graining used for SDS is shown in Fig. 1, such that a single DPD tail bead represents four carbon atoms, and the head group is represented by a single negatively charged bead. The counterion is represented by its own positively charged bead, in which the sodium atom is hydrated with water. The stages for parameterisation are as follows:

1. Set tail bead self-interactions a_{TT} based on the experimental chemical potential of octane, which we determine by performing Widom insertion calculations.

2. Define r_{C} (length scale for conversion into real units) based on matching the experimental density of octane (703 kg m^{-3}) to the density of the DPD fluid at the value of a_{TT} determined in the previous step.

3. Define the coarse graining of water and a_{WW} based on matching to the density (using the value of r_{C} from the previous step) and chemical potential of water. Note that the two unknowns (a_{WW} and the level of water coarse graining N , *i.e.* water molecules per coarse-grained bead), can in theory be solved exactly using the two equations (density and chemical potential).

For the charged beads (the SO_4^- head group and sodium, Na^+ , counterion), we choose to set the a_{ij} interactions to be the same as those with water. This decision is based on the assumption that charged beads will mostly be surrounded by water molecules in solution; therefore, at short range, the beads will interact similarly to water beads. All simulations described are performed using the DL_MESO software package⁶⁴ and we perform all simulations using a time step of $\Delta t = 0.01$.

3.1 Tail self-interactions (a_{TT})

The tail self-interactions (a_{TT}) are based on reproducing the chemical potential of octane. This calculation is performed by simulating a cubic box with periodic boundary conditions,

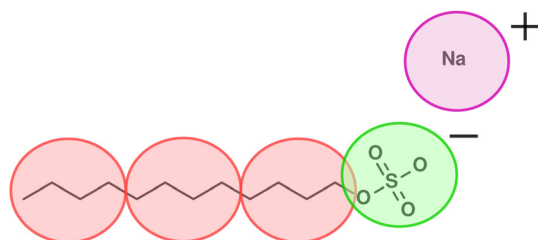


Fig. 1 Coarse-graining of an SDS molecule such that one tail bead (T) represents four carbon atoms and the head group is represented by a single negatively charged bead (H). The counterion is represented as a partially hydrated positively charged bead (I).

which is entirely filled with octane beads. Therefore, before performing the Widom insertions to calculate the chemical potential, one must first know the choice of a_{TT} relates to the resulting bead density of the system. Knowing the bead density of octane as a function of a_{TT} will allow us to correctly set the box sizes for the Widom insertion calculation.

To find a general relationship between a_{TT} and the bead density ρ , we perform a parameter sweep in the range $-26 \leq a_{\text{TT}} \leq -18$, with an interval of $\Delta a_{\text{TT}} = 1$. We simulate two bonded beads to represent octane with an equilibrium bond length of $l_0 = 0.6$ (note that we make our initial choice of bond length based on results from our previous work,⁶¹ more details of which can be found in the ESI†). These simulations are conducted in the const-NVT ensemble, in a box with periodic boundaries with dimensions $10r_{\text{C}} \times 10r_{\text{C}} \times 100r_{\text{C}}$. The beads are initialised with a random initial configuration using $n = 10\,000$ simulation beads (5000 molecules). For this number of beads, following equilibration the simulations produce coexisting regions of liquid and vapour, an example of which is shown in Fig. 2. Therefore, from this, the bead density of the liquid phase can be determined.

Fig. 3a shows the relationship between a_{TT} and the bead density, ρ_{T} . We fit a general expression in the form of a second-order quadratic, such that the density can be calculated using the expression:

$$\rho_{\text{T}} r_{\text{C}}^3 = -(8.42 \times 10^{-3})|a_{\text{TT}}|^2 + (5.17 \times 10^{-1})|a_{\text{TT}}| - 2.78. \quad (15)$$

Following this, we perform a separate set of simulations to calculate the chemical potential. Once again, we perform a parameter sweep in the range $18 \leq a_{\text{TT}} \leq 26$, with an interval of $\Delta a_{\text{TT}} = 1$. These simulations are conducted in a cubic box of length L , with periodic boundaries and beads initialised with a random initial configuration. The simulation is also conducted in the const-NVT ensemble.

For each simulation, we use $n = 50\,000$ simulation beads (25 000 molecules), and the dimensions L of the box are calculated individually for each value of a_{TT} . The value of L is chosen to generate the correct density as determined by eqn (15). This approach produces a domain completely filled with liquid so that Widom insertions can be performed without concern about potential insertions into a vapour phase. Note that for inserting the octane molecules during the Widom insertion, we choose to use a bond length of $l_0 = 0.6$ for all insertions.

For each simulation box (corresponding to different a_{TT}) we calculate the excess chemical potential μ^{excess} by Widom insertion using eqn (12). We convert the energy calculated in DPD units to real units using $k_{\text{B}}T = 4.11 \times 10^{-21} \text{ J}$ at room temperature.

The relationship between the calculated excess potential energy and a_{TT} is shown in Fig. 3b, and we can fit the general relationship for calculating the chemical potential in units of J,



Fig. 2 Coexisting liquid octane and vapour when $a_{\text{TT}} = 25$.



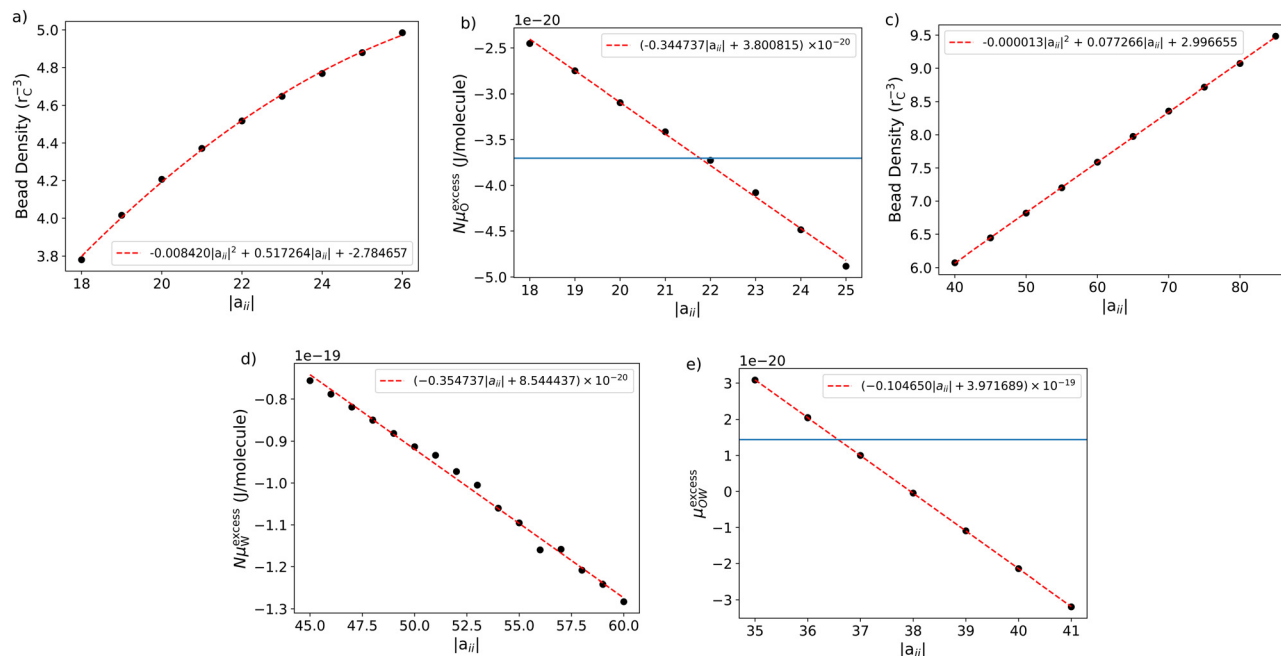


Fig. 3 Illustration of the fitting used in the parameterisation of the tail and water beads: (a) bead density of octane at various values of a_{TT} ; (b) excess chemical potential of octane as a function of a_{TT} ; (c) bead density of water at various values of a_{WW} ; (d) excess chemical potential of water as a function of a_{WW} (note that this is also a function of the degree of coarse-graining for water N); (e) excess chemical potential calculated due to the insertion of octane into water (DPD units) as a function of a_{OW} . Note that the horizontal lines in figure (b) and (e) represent the target values.

$$\mu_O^{\text{excess}} = (-3.45 \times 10^{-21})|a_{TT}| + 3.80 \times 10^{-20}. \quad (16)$$

The experimental excess chemical potential of octane is $-5.33 \text{ kcal mol}^{-1}$ ⁶⁵ (or $-3.703 \times 10^{-20} \text{ J per molecule}$); therefore, we determine that the value of a_{TT} which produces the correct excess chemical potential is $a_{TT} = -21.775$.

At the value $a_{TT} = -21.775$, eqn (15) can be used to determine a bead density of $\rho = 4.486$. Matching to the experimental density of octane (703 kg m^{-3} ⁶⁶) at room temperature allows us to determine that $r_C = 8.45 \text{ \AA}$. Note that this produces a bond length in real units of about 5.07 \AA . If we assume that the C–C–C angles are tetrahedral (approximately 109.5°) and that the experimental C–C bond is 1.543 \AA ,⁶⁷ the separation between four carbon atoms (or two adjacent beads) is calculated at around 5.04 \AA , meaning that the equilibrium bond length of $0.6r_C$ is an appropriate choice.

3.2 Water self-interaction a_{WW}

In this section, we perform a similar series of calculations for water beads (as for octane). In order to find a relationship between the density and value of a_{WW} , we perform a parameter sweep in a similar way to that performed in Section 3.1 for octane, however, the simulation for water contains no bonded beads. The relationship between a_{WW} and ρ_W is shown in Fig. 3c. We fit a general expression in the form of a second-order quadratic, such that the density can be calculated using the expression:

$$\rho_W r_C^3 = -(1.29 \times 10^{-5})|a_{WW}|^2 + (7.73 \times 10^{-2})|a_{WW}| + 3.00 \quad (17)$$

Similarly, to calculate the chemical potential, we now perform a parameter sweep over different a_{WW} values. Once again

we perform const-NVT simulations in a cubic simulation box where the entire domain is filled with water beads, this time setting the box size to reproduce the density in eqn (17).

The experimental value for the excess chemical potential of water is taken as $-6.33 \text{ kcal mol}^{-1}$ ⁶⁵ (or $-4.4 \times 10^{-20} \text{ J per molecule}$). It should be noted that because one bead represents more than one molecule, this must be taken into account in the calculation. Therefore, the calculated chemical potential depends on the coarse graining level, N , chosen for the water molecules. This means that we can, in theory, find the values of N and a_{WW} that reproduce the experimental value for the density of water at room temperature, as well as the chemical potential, where N could (in principle) take a non-integer value.

The relationship between the excess chemical potential of water and a_{WW} is shown in Fig. 3d. The relationship can be fit using a first-order equation (in units of J)

$$N\mu_W^{\text{excess}} = (-3.55 \times 10^{-21})|a_{WW}| + 8.54 \times 10^{-20} \quad (18)$$

where N is degree of coarse graining and μ_W^{excess} is in units of J per molecule. We solve the simultaneous equations for density and chemical potential to find $N = 2.72$ and $a_{WW} = -57.793$ as the optimal values to reproduce the density and chemical potential.

3.3 Cross interaction (a_{TW})

The cross-interaction, a_{TW} , is calculated by reproducing the experimental IDAC of octane in water, which has an experimental value of $\ln(\gamma^\infty) = 16.02$ ⁶⁸ at room temperature. Using eqn (14), we find the target value $\mu^{*,\text{excess}} = 3.5$ (units of $k_B T$). A sweep over a_{TW} parameter, calculating $\mu^{*,\text{excess}}$ in each case, gives the results shown in Fig. 3e. By fitting a first-order equation of the form (in units of J)



$$N\mu_{\text{WW}}^{\text{excess}} = (-1.05 \times 10^{-20})|a_{\text{WW}}| + 3.97 \times 10^{-19}, \quad (19)$$

we determine a cross-interaction value of $a_{\text{CW}} = -36.576$. A summary of the interactions calculated in this section can be found in Table 1.

4 Electrostatics study

Studies have shown that in traditional DPD, when the short-range repulsion between oppositely charged beads is high enough ($a_{ij} \gtrsim 100$ in eqn (1)), point charges can comfortably be used with no ion-pair formation.⁵⁵ In MDPD the short-range force is a combination of a repulsive contribution and an attractive contribution, and therefore, the magnitude of the repulsion at short range differs from that of standard DPD. In the limit of when $r_{ij} \rightarrow 0$, the magnitude of force approaches $|\mathbf{F}_{ij}^{\text{C}}| \rightarrow [a_{ij} + B(\rho_i + \rho_j)]$. Therefore, we investigate the effect that varying a_{ij} has on the average value of $\langle a_{ij} + B(\rho_i + \rho_j) \rangle$ for each pair of interacting beads. Increasing $|a_{ij}|$ has the effect of increasing local densities, making it possible that the net force is sufficiently repulsive to prevent ion collapse.

We perform these calculations using data from the water bead parameter sweep, described previously in Section 3.2. We create an additional Python script to determine the magnitude of the average $\langle a_{ij} + B(\rho_i + \rho_j) \rangle$. For a given trajectory output at time t , we compose a list of local densities for every bead in the system using eqn (9), and find the average local density per bead $\langle \rho_i \rangle = \rho_{\text{L}}$. We then calculate

$$\langle F_{ij} \rangle_{i,j} = \langle a_{ij} + B(\rho_i + \rho_j) \rangle_{i,j} \quad (20)$$

Since a_{ij} is the same for all beads in our simulation, $a_{ij} = A$, and we replace with the average local densities.

$$\langle F_{ij} \rangle_{i,j} = A + B\langle(\rho_i + \rho_j)\rangle_{i,j} = A + 2B\rho_{\text{L}} \quad (21)$$

Finally, we average over multiple outputs at different times t . Fig. 4 shows the results of this calculation, and shows that the net repulsive force at short distances is significantly larger than that used in standard DPD (in which the magnitude of the force tends to $|\mathbf{F}_{ij}^{\text{C}}| \rightarrow a_{ij}$ at short distances). Fig. 4 also shows that the magnitude of repulsion increases with $|a_{ij}|$, and we observe that at our proposed a_{WW} value, the magnitude of repulsion at short distances is likely to be large enough to avoid ion pair collapse.

To verify that our choice of parameters and point charges does not lead to artificial ion pair formation, we performed test simulations on simple salt solutions. We perform our simulations in a box with periodic boundary conditions and dimensions $7r_{\text{C}} \times 7r_{\text{C}} \times 50r_{\text{C}}$. The domain is populated with 500 positively charged beads, 500 negatively charged beads, and 2000 neutral beads representing water. We initialise with random placement throughout the entire simulation box, and after a short time, the beads form coexisting vapour and liquid phases. Following this equilibration period, we start data collection.

Fig. 5 shows the radial distribution function between the oppositely charged beads, where we confirm that there is no artificial pair formation, even at this relatively high salt

Table 1 Final $|a_{ij}|$ values for MDPD simulations of alkyl sulfate surfactants

	Tail (T)	Head (H)	Ion (I)	Water (W)
Tail (T)	21.775	—	—	—
Head (H)	36.576	57.793	—	—
Ion (I)	36.576	57.793	57.793	—
Water (W)	36.576	57.793	57.793	57.793

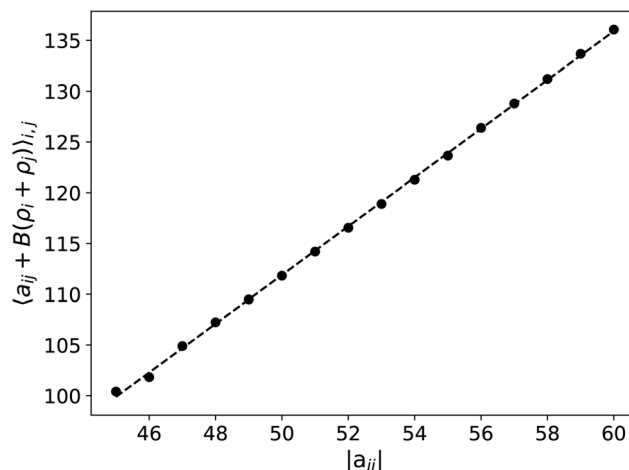


Fig. 4 Magnitude of the conservative force in MDPD in the limit of $r_{ij} \rightarrow 0$. Force is presented in DPD units.

concentration. If there was any ion collapse in the system, we would expect to see this reflected at short distances in the value of $g(r)$.

5 Critical micelle concentration (CMC)

In this section, we calculate the CMC for sodium alkyl surfactants of varying tail length (Section 5.1) and for SDS solutions with added salt (Section 5.2). Although the CMC is fairly well defined experimentally, there is no universally agreed best approach to calculating the CMC from simulations. This means that a variety of approaches have been used by various authors, and in this section, we begin by comparing a selection of the most common methods.

One common approach is to simulate at concentrations well above the CMC and calculate the CMC from the number of free surfactants.^{28,29,38,69} Another proposed method is to simulate at lower concentrations and define CMC as the concentration at which half of the surfactant material is free (as both monomers and submicellar aggregates) and half exists as micelles.^{28,70,71} Of course, different authors may define the size at which an aggregate can be considered a micelle differently, complicating direct comparisons between parameterisation schemes. Other proposed approaches include defining the CMC as the concentration at which a significant change in viscosity occurs,⁷² or when there is a significant change in the relationship between aggregation and concentration, *i.e.*, where one performs a linear fit of the submicellar concentration in regions both



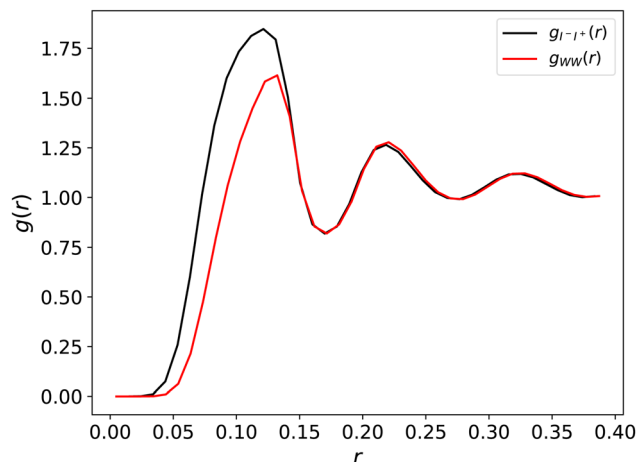


Fig. 5 Radial distribution function between the oppositely charged salt beads and water beads.

above and below an estimated CMC, where the crossing is defined as the CMC.^{27,73}

One further complication that arises for ionic surfactants is that there is a strong dependence of the number of free monomers on the concentration of micelles. In other words, the number of free monomers cannot be considered independent of the overall concentration.^{28,29,69} Therefore, to calculate the CMC concentration C_{CMC} from the free monomer concentration C_{F} , one should make use of the following relation⁷⁴

$$\log C_{\text{F}} = (1 + \alpha) \log(C_{\text{CMC}}^0) - \alpha \log \left(\frac{(1 - \alpha)(C_{\text{T}} - C_{\text{F}}) + C_{\text{F}} + C_{\text{S}}}{1 - VC_{\text{T}}} \right) \quad (22)$$

where C_{T} is the total surfactant concentration, C_{S} represents any additional counterions added as salt, V is the surfactant molar volume, and α is the degree of counterion binding to micelles. Note that C_{CMC}^0 in this equation represents the CMC in the absence of salt.

Throughout the literature, different approaches have been taken to choose a suitable value of α . One approach is to select a value of α derived from experiment,²⁹ while others determine the value directly from the simulations using the radial distribution function and an appropriate choice of cutoff distance.⁶⁹ However, it has been reported that the calculated CMC can be relatively insensitive to the precise value used.^{28,69} It will be shown in Section 6 that we also find this to be the case. In this work, we choose to determine the CMC using three of the different approaches discussed. These include:

- Method A – defined as the concentration at which half of the surfactant molecules are in micelles ($C_{\text{F}} = C_{\text{T}}/2$);
- Method B – *via* calculating the free concentration and using eqn (22);
- Method C – by performing linear fits to the relationship between aggregation and concentration and identifying a cross-over point.

5.1 Pure aqueous surfactant systems

To test our model, we simulate SDS surfactants and also investigate the impact of varying the length of the hydrocarbon chain. This allows us to check the robustness of our parameterisation scheme. We calculate the CMC for molecules containing 8, 12, and 16 carbon atoms (which we henceforth refer to as S8S, S12S and S16S). In practice, this involves varying the number of tail beads from 2–4 in our coarse-grained representation. The molecules are initially placed at random in a cubic box with periodic boundary conditions, subject to constant pressure.

We set the box pressure to be equivalent to that of pure water, which we calculate to be $P = -0.37$ (in DPD units). It should be noted that when the liquid phase is in coexistence with a very low-density vapour (as in our simulations), the pressure which generates liquid phase density ρ_{L} can be estimated⁵⁰ as $P(\rho_{\text{L}}) = 0$. Therefore we expect setting $P = 0$ would generate very similar results. However, to ensure we generate the exact same density described in Section 3, we run a bulk NVT simulation of our water beads a cubic simulation box at this density, to determine a precise value for P . We discuss setting the pressure of the system in more detail in the ESI.†

For each of the three different molecules, we choose a different box size owing to the differences expected in their CMC where we want to ensure there will be sufficient free monomers in the simulation boxes, choosing a total of $n = 24\,000$ (S8S), $n = 300\,000$ (S12S) and $n = 340\,000$ (S16S) beads in each case. Note that although it would be desirable to have a larger simulation box in the S16S case, we choose a moderate-size simulation for computational reasons. We allow each simulation to equilibrate until the number of free monomers is approximately constant, which takes considerably more computational effort for the larger molecules as a result of their larger simulation sizes.

In order to determine the number of free monomers, one must define a cutoff to distinguish between micelles and free molecules (*i.e.*, how large does an aggregate have to be to be considered a micelle). We determine this cutoff by plotting a histogram of aggregate sizes, from which we determine it is sensible to define micelles as aggregates containing 8 or more surfactant molecules.

Fig. 6 shows the results of this study, and presents plots of the surfactant molecules which can be considered to reside in micelles as a function of concentration. From this, we can determine the CMC using method A ($C_{\text{F}} = C_{\text{T}}/2$) and method C (using the cross-over point), which are both highlighted in Fig. 6. From the free concentration, we also determine a CMC value using method A (eqn (22)), where we use an experimentally⁷⁴ obtained estimate for the number of bound ions, $\alpha = 0.7$.

Table 2 presents a summary of the different CMC values calculated using different approaches, as well as a comparison with experimental data. Generally, we find that our simulations slightly underpredict the CMC; however, given the magnitude of variation between different length molecules (*i.e.* over an order of magnitude difference in the experimental CMC values between molecules which differ by only a single DPD bead), we



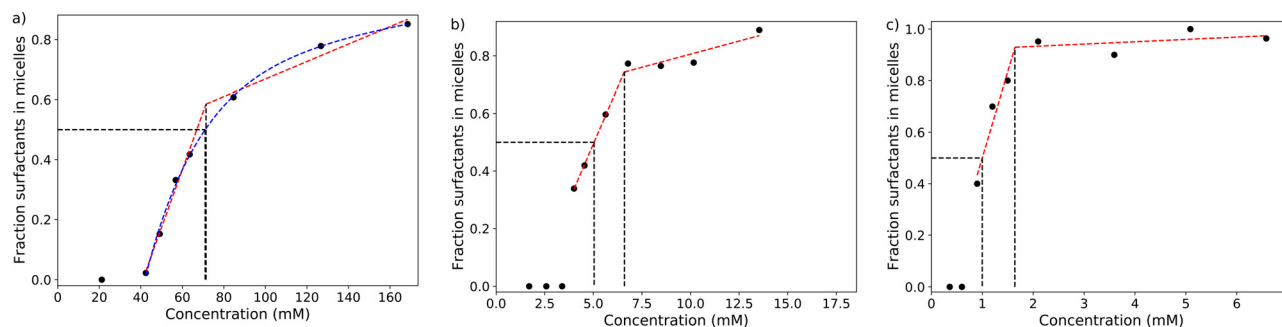


Fig. 6 The fraction of surfactant molecules existing in micelles for S8S (a), S12S (b) and S16S (c). Red linear fits are shown for the purpose of calculating the CMC using method C (as described in the main text). The CMC as determined using method A is also shown, and to identify the point at which $C_F = C_T/2$, a blue line of best fit is shown for the S8S surfactant. Calculated CMC values are highlighted by vertical black lines.

consider this underprediction to be relatively small. Fig. 7 compares the change in the $\log(\text{CMC}/\text{mM})$ as a function of chain length with the experimental data, which calculates the CMC using conductivity measurements. We see that the experimental trend of Aniansson *et al.*,¹⁶ showing a linear decrease in $\log(\text{CMC}/\text{mM})$ with chain length, is accurately reproduced by the simulations.

5.2 Addition of salt

We also investigate the impact of salt, by calculating the CMC of SDS in the presence of sodium chloride at various NaCl concentrations. Experimentally, the CMC is expected to decrease with increasing concentrations of salt.^{18,19}

We perform these calculations in a similar manner to those described in the previous section, keeping the number of water beads the same as that described previously for S12S. We modelled the additional salt using two oppositely charged beads (each possessing a charge $q = |e|$), which represent a single NaCl molecule. The short-range interactions with other beads in the system are the same as those for water, based on the reasoning outlined in Section 4. We simulate two different salt concentrations using 220 and 1500 Na^+ and Cl^- ions, which correspond to salt concentrations 15 and 100 mM respectively.

Fig. 8 shows the percentage of surfactant molecules in micelles for the different salt concentrations, and we see that micelles form more readily at higher salt concentrations as expected. We also observe that there is a relatively short concentration range where the solution transforms from consisting entirely of free monomers to complete aggregation. As the region is much narrower than in the salt-free simulations, this makes determining the CMC using method A more difficult. From Fig. 8, we estimate the CMC value as the point where

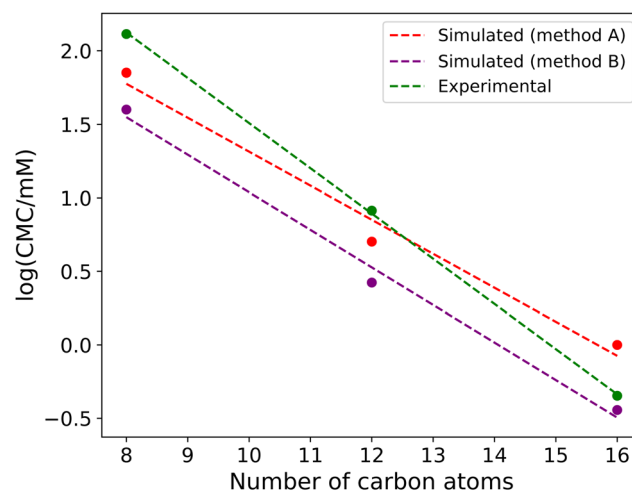


Fig. 7 CMC of alkyl sulfate surfactants as a function of tail length, where simulated results are compared with experimental data obtained from Aniansson *et al.*,¹⁶ who calculate the CMC using conductivity measurements.

there is an abrupt change in the number of surfactants in micelles (similar to using method C). These calculated CMC values are shown in Table 3 alongside experimentally determined values. We also show an approximate lower bound based on the highest concentration at which no aggregation is observed.

Since it was shown in the previous section that the values of the CMC are slightly underpredicted using our model (even in the absence of salt), it is unsurprising that the values in Table 3 are lower relative to experiment. However, the general trend for the simulated results is in excellent agreement with the experimental data. From the data presented by Dutkiewicz and Jakubowska,¹⁸ we know that the CMC C_{CMC} and salt concentration C_S should obey the relationship $C_{\text{CMC}} = a(C_S)^k$ where a and k are fitted constants. From our simulated data, we calculate a gradient value of, $k = -0.35$, which is reasonably good agreement with the experimental value of $k = -0.48$.

Suppose we now once again consider eqn (22). If we use the definition that the CMC is the concentration at which micelles first start appearing, then we can say $C_F = C_T = C_{\text{CMC}}$. If we also

Table 2 CMC (units mM) calculated from the MDPD results using three different methods. Also shown are experimental values. (Note that ref. 16 and 75 report the CMC at the slightly elevated temperature of 30 °C)

Molecule	Method A	Method B	Method C	Exp.
S8S	71.0	39.9	71.4	130 ^{16,17}
S12S	5.04	2.65	6.60	8.2 ^{16,17}
S16S	1.00	0.36	1.64	0.22–0.45 ^{16,75,76}



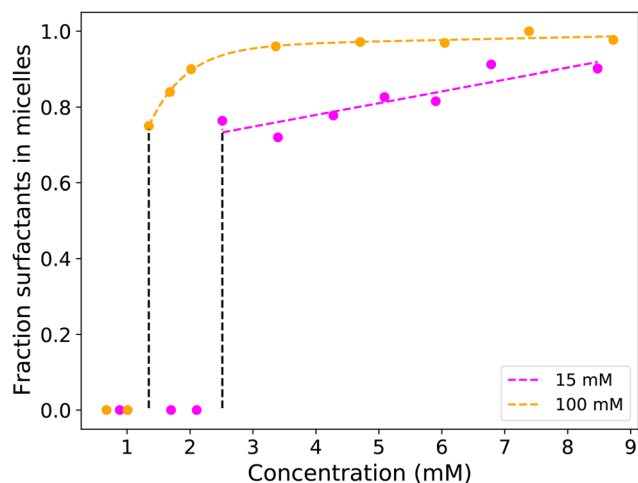


Fig. 8 The fraction of S12S molecules existing in micelles for salt concentrations 15 and 100 mM. Fits are shown for the purpose of calculating the CMC (as described in the main text), where the CMC values are highlighted by dashed black lines.

suppose that the surfactant concentration at the CMC is small ($VC_{\text{CMC}} \ll 1$) then eqn (22) simplifies to

$$\log C_{\text{CMC}} = (1 + \alpha) \log(C_{\text{CMC}}^0) - \alpha \log(C_{\text{CMC}} + C_s). \quad (23)$$

A rearrangement yields

$$\log(C_{\text{CMC}}(C_{\text{CMC}} + C_s)^\alpha) = (1 + \alpha) \log(C_{\text{CMC}}^0) \quad (24)$$

In the limit that $C_{\text{CMC}} \ll C_s$, we can say that $C_{\text{CMC}}(C_{\text{CMC}} + C_s)^\alpha \rightarrow C_{\text{CMC}}(C_s)^\alpha$, and therefore we can find an expression for C_{CMC}

$$C_{\text{CMC}} = [(C_{\text{CMC}}^0)^{(1+\alpha)}](C_s)^{-\alpha}. \quad (25)$$

Note that this is equivalent to the experimentally derived expression $C_{\text{CMC}} = a(C_s)^k$, where the constant a is a function of the saltless CMC and the constant $k = -\alpha$.

6 Aggregation number and micelle shape

In this section, we use our model to determine the equilibrium aggregation number for surfactants S8S, S12S and S16S. For each surfactant, we do this for a range of concentrations: 2.5, 5, 7.5, 10 wt%. The simulations are set up in the same way as described in Section 5, however, here we use much smaller boxes than those that were required for calculating the CMC. We take an approach that has previously been used by other

Table 3 CMC of S12S solutions containing additional NaCl salt, where simulated results are compared with experimental data obtained from Dutkiewicz and Jakubowska,¹⁸ who obtained their CMC values *via* conductivity measurements. Also shown in brackets is an approximate lower bound for the CMC. All data presented in units mM

Salt concentration	Simulated CMC	Experimental CMC
15	2.5 (2.1)	4.2
100	1.3 (1.0)	1.7

authors,²⁸ in which we keep the number of surfactant molecules N_s in the box constant and alter the number of water beads N_w to vary the concentration of the box. We use $N_s = 500$ surfactant molecules, meaning that the smallest simulation box used is for the S8S surfactant at a concentration of 10% ($N_w = 21\,168$), and our largest box is for the S16S surfactant at a concentration of 2.5% ($N_w = 136\,010$). Each simulation is equilibrated for long simulations until the aggregation number is determined to no longer change.

Fig. 9 shows the final aggregation number calculated for each simulation case. We see that the aggregation size generally increases with both tail length and concentration. There is a reasonably broad distribution of micelle sizes in each simulation case, represented by the relatively large error bars shown in Fig. 9. For example, for C16 at a concentration of 10%, we find that 6 micelles form of various sizes in the range 50–113. Experimentally,⁷⁷ S8S is reported to have an aggregation number at around $N_{\text{Agg}} \approx 40$ at 6.4 wt%, which is in good agreement with our simulated results. However, S12S is reported as having $N_{\text{Agg}} \approx 80$ at 2.9 wt%, indicating that our simulated value of around $N_{\text{Agg}} \approx 36$ is a significant underestimate.

Other authors have commonly reported that the aggregation number for micelles formed by ionic surfactants in DPD simulations are underpredicted.^{28,29} One suggestion is that this underprediction is due to problems with equilibration. This stems from the fact that micelles with a net charge would have a significant amount of repulsion between them. Therefore, on the time scale of the simulation, they may not meet sufficiently to come together and aggregate to form larger micelles.

In order to test the impact of the charges on equilibration, we performed simulations in which the charges were initially turned off. We expect that this will allow the surfactant molecules to aggregate significantly more easily, due to the reduced repulsion between head groups. In all cases, we simulate using

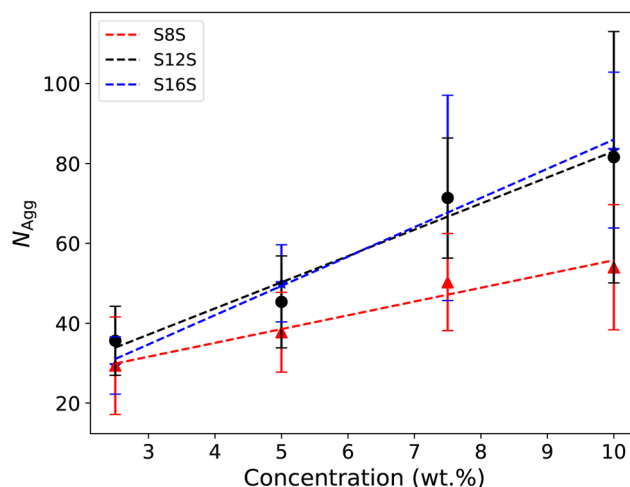


Fig. 9 Final aggregation number of alkyl sulfate surfactants with different tail lengths at different concentrations. Linear lines of best fit are shown for each surfactant type. Error bars represent the standard deviation in micelle sizes.



a total of 24 000 beads, and we vary the number of surfactant molecules N_S in the box.

In the N_S range studied ($20 < N_S < 220$) we observe that the surfactants quickly aggregate into a single, large micelle, often significantly above the experimental aggregation number. An example is shown in Fig. 10a. Note that the formation of large micelles is to be expected because one would expect head group repulsion to be a limiting factor on micelle size. However, this approach allows us to treat this single large micelle as an 'initial state' for studying anionic micelles. It has been shown²⁰ that when simulations are initialised with micelles larger than their experimental values, they can break down into micelles, which are closer in size to those expected experimentally.

When the charges are turned on, the micelle transforms its shape and the counterions move to surround the micelle. However, the micelle does not necessarily break down into smaller aggregates, as one might expect. The S12S and S16S molecules always remain as a single micelle, during the run time, albeit with a significant shape change at high aggregation numbers (see Fig. 10b). This behaviour allows us to study micelles as a function of their aggregation number, and, in particular, study shape changes. However, for S8S surfactants, which have weaker attraction between (shorter) chains, the micelles exhibit different behaviour and break down once the electrostatic interactions are turned on when the initial micelle is above a particular size ($N_S \gtrsim 80$). This suggests that for S8S, large metastable micelles are significantly less stable than for the two longer surfactant molecules. This may also be reflective of the fact that experiments have shown that the width of the distribution of stable micelles increases with chain length.¹⁶

To quantify the non-spherical nature of the large micelles, we calculate the moment of inertia I_m (where $m = x, y, z$). From this, we define the eccentricity:

$$e = 1 - \frac{I_{\min}}{I_{\text{Avg}}} \quad (26)$$

where I_{\min} is the smallest moment of inertia in any dimension and I_{Avg} is the average of all three. Here, $0 \leq e \leq 1$ and a value of $e = 0$ corresponds to a perfectly spherical micelle.

A plot of eccentricity as a function of aggregation number is shown in Fig. 11. For S8S surfactants, we are limited to studying

a reduced range of surfactant sizes due to the breakdown of micelles when initialised at larger aggregation numbers. The bars plotted illustrate the standard deviation of the eccentricity over the measurement period, highlighting the significant fluctuations in shape we observe, particularly as the micelle sizes increase. We observe that the eccentricity increases with decreasing tail length, with S16S micelles being the most spherical at all micelle sizes. For the S12S and S16S surfactants, there appears to be a minima illustrating an optimal value of the aggregation number for which the micelles are most spherical, which is located around 60 for S12S and 90 for S16S. It is of interest that these values are very close to the experimental values for the average aggregation numbers at the CMC (where $N_{\text{Agg}} = 64$ and $N_{\text{Agg}} = 100$ for S12S and S16S respectively¹⁶), and small angle scattering studies^{5,78} and packing parameter calculations^{79,80} indicate that SDS micelles at this concentration should be relatively spherical. Following this plateau, the eccentricity increases with larger aggregation size as the micelles begin to alter their shape. This increase is significantly more rapid for S12S compared to S16S, with the S12S micelles showing a more rapid transition to rod-like shapes. Spin probe experiments have indicated that SDS micelles start to become rod-like at aggregation numbers around 130,⁸¹ which is in agreement with what we observe in our simulations.

In Section 5 we chose to define the degree of counterion binding as $\alpha = 0.7$. Using our simulations of single micelles, we are now able to calculate the degree of binding as a function of micelle size, to investigate whether this was a suitable choice. We calculate α by determining the number of sodium ions within a specified cutoff range R_{Ions} of the micelle head groups. This cutoff range is refined using the first peak of the radial distribution function between the head groups and the counterions. Fig. 12 shows examples of the radial distribution function for different molecules of various sizes. While the magnitude of the $g(r)$ peaks varies significantly, we note that

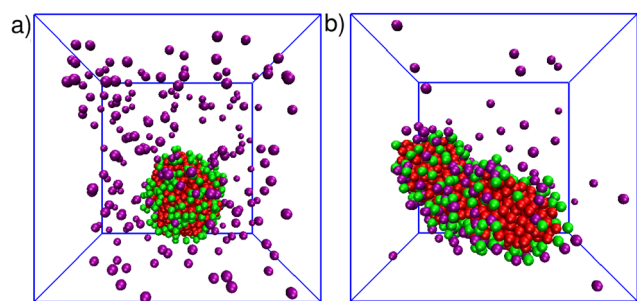


Fig. 10 Micelle formation of SDS molecules when the charges are turned off (a), compared with the shape change that results when they are turned on (b). The example given is using 200 SDS molecules. Water molecules are not shown for clarity and beads are coloured according to type: tail (red), head (green), and sodium ions (purple).

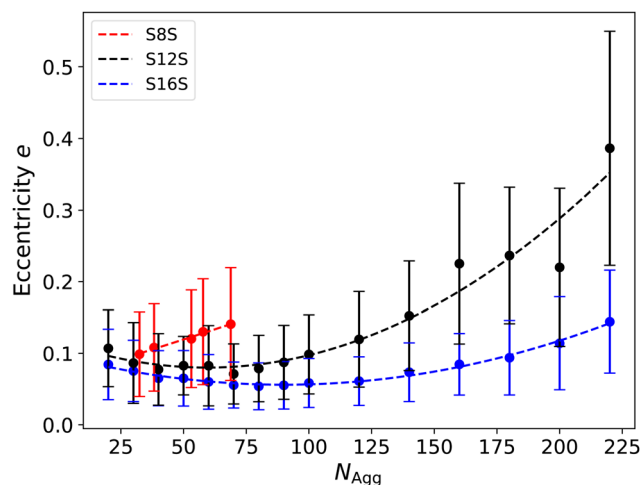


Fig. 11 Eccentricity e as a function of micelle size, where the bars represent the standard deviation over the measurement period. We note that errors in e are typically much smaller than one standard deviation.



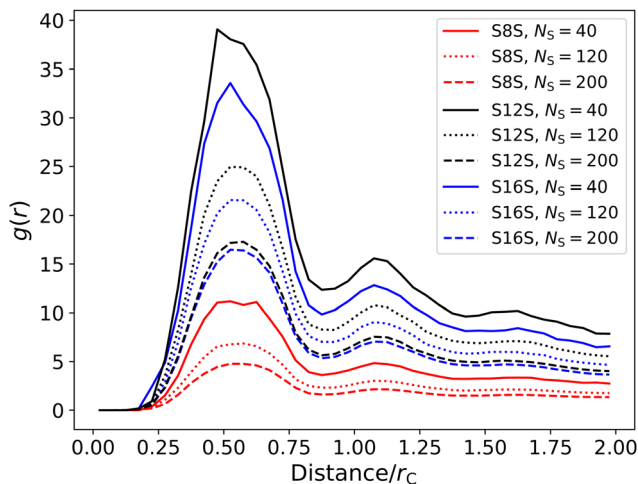


Fig. 12 Radial distribution function between the head groups and counterions for molecules of different size and type.

the local minima between the two first peaks (at a distance of $0.9r_C$) does not significantly vary in position. Therefore, we choose a cutoff distance of $R_{\text{Ions}} = 0.9$ in all calculations for determining α .

In order to determine the number of ions which are bound to the micelle, our aim is to calculate the average number of ions which are within cutoff radius R_{Ions} of the head groups in the micelle (*i.e.* the surface of the micelle). Here we can use the simplification that all surfactants are contained within the micelle (that is, there are no free surfactants). This can be done in one of two ways. The first is to make use of the radial distribution $g(r)$ directly and calculate the number of particles within distance R_{Ions}

$$N_{ij} = \rho_i \int_0^{R_{\text{Ions}}} 4\pi r^2 g_{ij}(r) dr \quad (27)$$

where ρ_i is the bulk density of particles of type i . However, this expression only provides the expected number N_{ij} of type i beads within distance R_{Ions} for a single bead of type j . To find the probability of having at least one bead of type i within this distance, we could use the Poisson distribution to give

$$\alpha = P(n \geq 1) = 1 - e^{-N_{ij}(R_{\text{Ions}})}. \quad (28)$$

In this work, however, we choose to determine the number of bound ions directly from our simulation trajectories instead. For each frame we use a Python script to determine the number of ions which are within at least one cutoff radius of a surfactant head group. We then average over multiple trajectories to calculate an accurate value for α for different micelle sizes. We note that we compare our results calculated *via* this method with the RDF approach discussed above and generate very similar results.

Fig. 13 shows how the number of ions bound to a micelle depends on the aggregation size. Here we focus on the S12S and S16S micelles due to the wide range of micelle sizes we have available for study. While we do not observe a significant dependence on tail length, we see a strong dependence on micelle size. With increasing size, the micelle increases its net

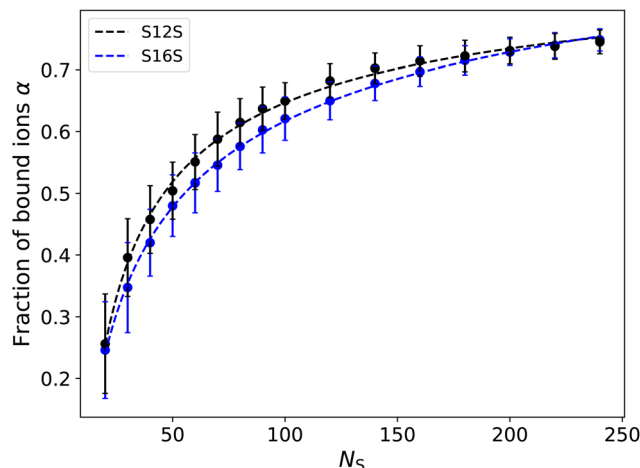


Fig. 13 Fraction of counter-ions bound to the micelle as a fraction of the aggregate size. Bound ions are classed as being those which are within a distance $r < R_{\text{Ions}} = 0.9$ of a head group. Vertical bars represent the standard deviation over the measurement period (note that standard errors are around the size of the points).

charge, likely increasing the percentage of bound ions to aid its electrostatic screening. At larger micelle sizes ($N_s \gtrsim 100$) the number of bound ions is in agreement with experimental values, and therefore in line with what was used in Section 5 to calculate the CMC in this work.

However, since the value of α appears to be affected by varying the concentration/size of the system (particularly at lower sizes), we investigate the impact this has on our previous calculation of the CMC since our micelles are typically smaller in these simulations. However, similarly to others,^{28,69} we observe that varying $\alpha \approx 0.4$ – 0.8 does not have a significant impact on the calculated CMC. For example, using a value of $\alpha = 0.4$ yields an estimate for the CMC of 2.70 mM (compared to a CMC value of 2.65 when using $\alpha = 0.7$). Thereby making our previous choice of $\alpha = 0.7$ a reasonable one.

We also note that it is possible to predict the degree of counter-ion binding using eqn (23), and measurements of the CMC in the presence of salt. We take the experimental value for the CMC of saltless SDS as 8.2 mM^{16,17} (Table 2) and the CMC in the presence of 15 mM and 100 mM of salt as 4.2 mM and 1.7 mM,¹⁸ respectively (Table 3). Using eqn (23) yields estimates for the counter-ion binding of $\alpha = 0.79$ and $\alpha = 0.62$, where the average of these two values $\alpha = 0.71$ almost perfectly matches our choice of $\alpha = 0.7$.

7 Discussion

We discuss here some of the key results and considerations for our model, including the choice to use point charges as opposed to smeared charges; issues related to equilibration and aggregation number prediction; the impact of the tail-water interaction on our parameterisation scheme and the quantities we calculate; and finally, how our results compare with those from other experiments and simulations.



7.1 Use of point charges

As a result of our choice to use MDPD compared to standard DPD, we are able to use point charges as opposed to smeared charges. The smeared charge approach, which is typically used in traditional DPD methods, is usually applied to deal with the fact that the soft repulsion experienced by DPD beads is not strong enough to prevent ion-pair formation. However, in this work, we show that there is no evidence of ion collapse occurring in our MDPD simulations.

The ability to use point charges was hypothesised as a result of our analysis of the strength of the repulsion at short ranges and was confirmed by checking the radial distribution functions from salt solutions at high concentrations. We check that our charged ions in micelle simulations behave correctly by calculating the number of bound ions as a function of aggregation number. This allows us to confirm that the degree of dissociation of the counterions is in excellent agreement with experimental values, which is crucial for studying the shape and size of simulated micelles.

The use of point charges is simpler than smeared charges, and eliminates the issue that there are often several unknowns when charges are smeared over the bead volume. For example, various forms of the smear function have been proposed^{82–85} and various degrees of smearing have been used by different authors.^{29,83} The function and degree of smearing are typically chosen for computational efficiency reasons as opposed to scientific reasoning, with little attention paid to the impact this has on the simulated results. However, we note that the use of smeared charges is perfectly possible within MDPD, is fully implemented within codes such as DL_MESO, and potentially could remove the danger of stability issues in some cases (noting the Fisher–Ruelle stability criterion⁸⁶).

7.2 Equilibration of aggregation number

We observe that our calculated aggregation number equilibrates very slowly, such that it can be difficult to ascertain when equilibration has completed. We also observe that our aggregation numbers are somewhat lower than might be expected for the S12S2 and S16S surfactants. Once formed, charged micelles do not meet on the time scales we look at. The net repulsion between charged micelles is so great, that the micelles never get close enough to aggregate into larger micelles. This creates a difficulty in generating larger aggregation numbers.

In experimental reality, micelles are in dynamic equilibrium constantly disintegrating and reforming, and it is generally considered that micellar solutions exhibit two relaxation processes^{16,87,88} characterised by relaxation times τ_1 and τ_2 . The first timescale τ_1 relates to the exchange of surfactant monomers between micelles (order of microseconds), while τ_2 relates to micellization-dissolution equilibrium (order of milliseconds). Experiments¹⁶ on sodium alkyl sulfate solutions have shown that, in general, τ_1 and τ_2 decrease with surfactant concentration and also increase with increasing alkyl chain length. This may partially explain our difficulty in equilibrating micelles at lower concentrations and longer tail lengths.

To overcome these time-scale limitations, a potentially interesting approach for future researchers might be the use of a biased Monte Carlo simulation technique, in which one computes micelle free energies as functions of aggregation number in order to determine the equilibrium aggregation numbers.⁸⁹

7.3 Choice of tail–water interaction

In Section 3.3 it was shown that the tail–water interaction is defined based on a Widom insertion approach, with octane being the reference molecule. The choice to use octane was based on the fact that, in our coarse-grained representation, it can be represented by two single-bonded beads. While matching a longer hydrocarbon might be more desirable for calculating the CMC for larger molecules, such as dodecane or even hexadecane, it is worth noting that the Widom insertion calculation is much more complex for longer molecules. Firstly, it would involve more than two beads, and therefore conformation needs to be taken into account when performing insertions. Secondly, a larger molecule would result in finding an accurate value for μ^{excess} very difficult. For a larger molecule, the available space in the simulation box in which the molecule can be inserted is very limited. This means that most insertion attempts result in very high insertion energies ΔU , resulting in $\exp(-\Delta U/k_B T)$ becoming negligibly small most of the time. This means that determining μ^{excess} is statistically difficult and inefficient. For these reasons, we chose to focus on insertions of octane.

However, we observe that the precise value of the CMC calculated is very sensitive to the value of a_{TW} . This has been commented on in conventional DPD by previous authors.²⁹ To illustrate this point, we calculate a value for the CMC of S12S if we increase the value of the tail–water interaction by a small amount, choosing $\Delta a_{\text{TW}} = 0.5$ (such that the new value $a_{\text{TW}} = 37.076$, an increase of about 1.4%). In this case, we calculate a CMC value (using method A) as $C_{\text{CMC}} = 9.1$ mM (see ESI† for details), compared to the value of 5.04 mM calculated using our original parameters. Therefore, we caution that it is possible that small inaccuracies in the calculated value of a_{TW} can lead to large changes in the CMC, and that our exact values of CMC could be improved by matching tail–water interactions to different choices of molecules. However, we also note both the aggregation number and micellar shape are relatively insensitive to the exact choice of a_{TW} , at least to a significantly lesser extent than the CMC.

7.4 Comparison with earlier studies

In this article, we report values for the CMC and aggregation number for sodium octyl sulfate, sodium dodecyl sulfate and sodium hexadecyl sulfate. We also report on the change in the CMC for additional salt for SDS solutions. Finally, we also report the fraction of bound ions α as a function of aggregation number, as well as their eccentricity as a means to quantify micellar shape.

We showed that both without (Table 2) and with (Table 3) salt, our CMC values are underpredicted in comparison with the experimental data, however, the trends of CMC with tail length match experimental data reasonably well. It is an issue



that other authors have also found when using standard DPD approaches,^{28,38} and it is difficult to attribute an exact reason for this. However, we will note that accurate prediction of the CMC is expected to be difficult from theoretical reasons alone. The Gibbs free energy change for micelle formation from 1 mol surfactant is related to the CMC by

$$\Delta G^0 = RT \ln \text{CMC}, \quad (29)$$

where ΔG^0 is the difference between the Gibbs free energy of the micelle and of the monomeric surfactant, which is related to the chemical potentials we determine *via* Widom insertion. However, the main point is that the resulting simulated CMC can become exponentially incorrect, for very small errors in ΔG^0 . This is why it often makes more sense to compare the logarithmic value of the CMC's instead, which has been done in this article, as well as by other authors.^{28,38}

We discussed in the previous section the sensitivity that the CMC has to the tail–water interaction, however, the CMC mismatch is most likely to result from how we treat charges in the system since these have the greatest impact on altering the CMC behaviour. We made no attempt to parameterise the charged beads in our model, and it is highly likely that the treatment of charges will need to be tuned to accurately reproduce the correct ΔG^0 . Up until now, direct parameterisation of charged beads has been largely neglected, with most authors taking the simple approach to assign beads as having the same a_{ij} interactions as water. However, an interesting study by Lavagnini *et al.*⁹⁰ indicated that for salt solutions, even for nonionic surfactants, the calculated CMC values depend largely on the ion–tail repulsion parameter, meaning that this is likely to be a source of error in our approach.

Turning our attention now to the shape of micelles, it was noted in Section 6 that spin probe experiments indicate that SDS micelles start to become rod-like at aggregation numbers which are comparable to those we see in our simulations. There is also a wealth of literature using small-angle neutron scattering (SANS)^{4–10} and small-angle X-ray scattering^{13–15} (SAXS) to investigate the shape and size of SDS micelles. Typically it is concluded that micelles at equilibrium are nearly spherical, but slightly prolate, which is in agreement with what we see in our study. Fig. 11 shows that we never determine an eccentricity of 0 (which would correspond to a spherical micelle), instead usually taking larger values of $0.1 < e$. Interestingly, a molecular dynamics study by Palazzesi *et al.*²⁴ reported an eccentricity value of around $e = 0.15$ for SDS micelles at aggregation numbers of $N = 60$, which is in reasonably good agreement with that we have reported at in this article. Other DPD studies³¹ have also found similar degrees of eccentricity for micelles of around this size.

Finally, we discuss the binding of ions to micelles. It was noted in Section 6 that our calculated fraction of bound ions α is generally in line with experimental measurements⁷⁴ at comparable micelle sizes. It is difficult to compare with experimental data for smaller micelles, as data only exists for micelles at their equilibrium, larger sizes. However, it was shown that the counter ion binding varies very little over SDS micelles in

the range $60 \leq N \leq 110$, which is broadly what we see in Fig. 13.

The degree of counterion binding is a quantity that can also be determined by molecular dynamics simulations of single micelles. It has been shown that α can depend on the choice of force field²¹ (where Tang *et al.*²¹ define bound ions to be sodium atoms which exist within 0.6 nm of the oxygen atoms in the head group). Authors report that for an SDS micelle with aggregation number $N = 60$, α ranges from 45–70% for the six force fields tested, which is generally in line with experimental values. Using a cutoff of $0.9r_c$ (0.76 nm), we determined in our simulations that micelles of the same size have $\alpha = 0.55$, while if we use a cutoff $0.71r_c$ (which translates in real units to a more comparable value to Tang *et al.*²¹) we find $\alpha = 0.48$. This puts our MDPD simulated value in reasonable agreement with those from MD simulations. However, it is still worth noting that this still is not exactly comparable, since the distance used in our calculation is the centre of the bead rather than the oxygen atoms, and if calculated equivalently our values are likely to have even closer matching.

8 Conclusion

In this article, we present a comprehensive parameterisation scheme for MDPD simulations. We conclude that our MDPD model provides good estimates for the CMC as a function of tail length and salt concentration. Our model also allows us to easily study micelles, including their size, shape, and charge, making it a useful tool for studying charged surfactants. We believe that, using the methodology presented here, the model could easily be extended to other ionic and nonionic surfactants, to oils, to mixed surfactant systems and to higher concentration phases.

Author contributions

RH: conceptualization, data curation, formal analysis, investigation, methodology, software, validation, visualization, writing – original draft. CA: project administration, writing – review and editing. MW: funding acquisition, project administration, supervision, writing – review and editing.

Data availability

The data that support the findings of this study are available from the corresponding author, RH, upon reasonable request.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 P. Kékicheff, C. Grabielle-Madellmont and M. Ollivon, *J. Colloid Interface Sci.*, 1989, **131**, 112–132.
- 2 A. S. Poulos, C. S. Jones and J. T. Cabral, *Soft Matter*, 2017, **13**, 5332–5340.
- 3 A. Capaccio, S. Caserta, S. Guido, G. Rusciano and A. Sasso, *J. Colloid Interface Sci.*, 2020, **561**, 136–146.
- 4 S. S. Berr and R. R. M. Jones, *Langmuir*, 1988, **4**, 1247–1251.
- 5 B. Cabane, R. Duplessix and T. Zemb, *J. Phys.*, 1985, **46**, 2161–2178.
- 6 M. Bergstrom and J. S. Pedersen, *Phys. Chem. Chem. Phys.*, 1999, **1**, 4437–4446.
- 7 L. J. Magid, Z. Li and P. D. Butler, *Langmuir*, 2000, **16**, 10028–10036.
- 8 M. Kakitani, T. Imae and M. Furusaka, *J. Phys. Chem.*, 1995, **99**, 16018–16023.
- 9 M. Ludwig, R. Geisler, S. Prévost and R. von Klitzing, *Molecules*, 2021, **26**, 4136.
- 10 B. Hammouda, *J. Res. Natl. Inst. Stand. Technol.*, 2013, **118**, 151–167.
- 11 S. Khodaparast, W. N. Sharratt, G. Tyagi, R. M. Dalglish, E. S. Robles and J. T. Cabral, *J. Colloid Interface Sci.*, 2021, **582**, 1116–1127.
- 12 V. Y. Bezzobotnov, S. Borbely, L. Cser, B. Farago, I. A. Gladkih, Y. M. Ostanevich and S. Vass, *J. Phys. Chem.*, 1988, **92**, 5738–5743.
- 13 S. L. Gawali, M. Zhang, S. Kumar, D. Ray, M. Basu, V. K. Aswal, D. Danino and P. A. Hassan, *Langmuir*, 2019, **35**, 9867–9877.
- 14 J. Lipfert, L. Columbus, V. B. Chu, S. A. Lesley and S. Doniach, *J. Phys. Chem. B*, 2007, **111**, 12427–12438.
- 15 Y. Mirgorod, A. Chekadanov and T. Dolenko, *Chem. J. Mold.*, 2019, **14**, 107–119.
- 16 E. A. G. Aniansson, S. N. Wall, M. Almgren, H. Hoffmann, I. Kielmann, W. Ulbricht, R. Zana, J. Lang and C. Tondre, *J. Phys. Chem.*, 1976, **80**, 905–922.
- 17 F. H. Quina, P. M. Nassar, J. B. Bonilha and B. L. Bales, *J. Phys. Chem.*, 1995, **99**, 17028–17031.
- 18 E. Dutkiewicz and A. Jakubowska, *Colloid Polym. Sci.*, 2002, **280**, 1009–1014.
- 19 H. Khan, J. M. Seddon, R. V. Law, N. J. Brooks, E. Robles, J. T. Cabral and O. Ces, *J. Colloid Interface Sci.*, 2019, **538**, 75–82.
- 20 S. D. Peroukidis, D. G. Mintis, I. Stott and V. G. Mavrantzas, *J. Phys. Mater.*, 2021, **4**, 044001.
- 21 X. Tang, P. H. Koenig and R. G. Larson, *J. Phys. Chem. B*, 2014, **118**, 3864–3880.
- 22 S. Jalili and M. Akhavan, *Colloids Surf., A*, 2009, **352**, 99–102.
- 23 N. Yoshii and S. Okazaki, *Condens. Matter Phys.*, 2007, **10**, 573.
- 24 F. Palazzesi, M. Calvaresi and F. Zerbetto, *Soft Matter*, 2011, **7**, 9148.
- 25 B. J. Chun, J. I. Choi and S. S. Jang, *Colloids Surf., A*, 2015, **474**, 36–43.
- 26 R. D. Groot and P. B. Warren, *J. Chem. Phys.*, 1997, **107**, 4423–4435.
- 27 D. Nivón-Ramírez, L. I. Reyes-García, R. Oviedo-Roa, R. Gómez-Balderas, C. Zuriaga-Monroy and J.-M. MartínezMagadán, *Colloids Surf., A*, 2022, **645**, 128867.
- 28 R. L. Anderson, D. J. Bray, A. Del Regno, M. A. Seaton, A. S. Ferrante and P. B. Warren, *J. Chem. Theory Comput.*, 2018, **14**, 2633–2643.
- 29 R. Mao, M.-T. Lee, A. Vishnyakov and A. V. Neimark, *J. Phys. Chem. B*, 2015, **119**, 11673–11683.
- 30 M. Panoukidou, C. R. Wand, A. Del Regno, R. L. Anderson and P. Carbone, *J. Colloid Interface Sci.*, 2019, **557**, 34–44.
- 31 R. L. Hendrikse, A. E. Bayly and P. K. Jimack, *J. Phys. Chem. B*, 2022, **126**, 8058–8071.
- 32 S. J. Gray, M. Walker, R. Hendrikse and M. R. Wilson, *Soft Matter*, 2023, **19**, 3092–3103.
- 33 T. L. Rodgers, O. Mihailova and F. R. Siperstein, *J. Phys. Chem. B*, 2011, **115**, 10218–10227.
- 34 S. Jury, P. Bladon, M. Cates, S. Krishna, M. Hagen, N. Ruddock and P. Warren, *Phys. Chem. Chem. Phys.*, 1999, **1**, 2051–2056.
- 35 R. D. Groot, *Langmuir*, 2000, **16**, 7493–7502.
- 36 G. Kacar, E. A. J. F. Peters and G. de With, *EPL*, 2013, **102**, 40009.
- 37 M.-T. Lee, A. Vishnyakov and A. V. Neimark, *J. Chem. Theory Comput.*, 2015, **11**, 4395–4403.
- 38 M.-T. Lee, R. Mao, A. Vishnyakov and A. V. Neimark, *J. Phys. Chem. B*, 2016, **120**, 4980–4991.
- 39 A. Maiti and S. McGrother, *J. Chem. Phys.*, 2004, **120**, 1594–1601.
- 40 L. Rekvig, M. Kranenburg, J. Vreede, B. Hafskjold and B. Smit, *Langmuir*, 2003, **19**, 8195–8205.
- 41 K. P. Travis, M. Bankhead, K. Good and S. L. Owens, *J. Chem. Phys.*, 2007, **127**, 014109.
- 42 A. Vishnyakov, M.-T. Lee and A. V. Neimark, *J. Phys. Chem. Lett.*, 2013, **4**, 797–802.
- 43 A. Vishnyakov, R. Mao, M.-T. Lee and A. V. Neimark, *J. Chem. Phys.*, 2018, **148**, 024108.
- 44 R. L. Anderson, D. Bray, A. S. Ferrante, M. G. Noro, I. P. Stott and P. Warren, *J. Chem. Phys.*, 2017, **147**, 094503.
- 45 A. Truskowski, M. Epple, A. Fiethen, A. Zielesny and H. Kuhn, *J. Colloid Interface Sci.*, 2013, **410**, 140–145.
- 46 M. Walker, A. J. Masters and M. R. Wilson, *Phys. Chem. Chem. Phys.*, 2014, **16**, 23074–23081.
- 47 M. Walker and M. R. Wilson, *Soft Matter*, 2016, **12**, 8588–8594.
- 48 I. Pagonabarraga and D. Frenkel, *J. Chem. Phys.*, 2001, **115**, 5015–5026.
- 49 P. Español and P. Warren, *Europhys. Lett.*, 1995, **30**, 191–196.
- 50 P. B. Warren, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2003, **68**, 066702.
- 51 P. B. Warren, *Phys. Rev. E: Stat., Nonlinear, Soft Matter Phys.*, 2013, **87**, 045303.



- 52 Y.-C. Li, H. Liu, X.-R. Huang and C.-C. Sun, *Mol. Simul.*, 2011, **37**, 875–883.
- 53 A. Ghoufi and P. Malfreyt, *J. Chem. Theory Comput.*, 2012, **8**, 787–791.
- 54 R. Hendrikse, A. Bayly and P. Jimack, *J. Chem. Phys.*, 2023, **158**, 214906.
- 55 K. A. Terrón-Mejía, R. López-Rendón and A. G. Goicochea, *J. Phys.: Condens. Matter*, 2016, **28**, 425101.
- 56 D. J. Bray, R. L. Anderson, P. B. Warren and K. Lewtas, *J. Phys. Chem. B*, 2022, **126**, 5351–5361.
- 57 H. Alasiri and W. G. Chapman, *J. Mol. Liq.*, 2017, **246**, 131–139.
- 58 A. Khedr and A. Striolo, *J. Chem. Theory Comput.*, 2018, **14**, 6460–6471.
- 59 K. Shi, C. Lian, Z. Bai, S. Zhao and H. Liu, *Chem. Eng. Sci.*, 2015, **122**, 185–196.
- 60 P. M. Pieczywek, W. Plazinski and A. Zdunek, *Sci. Rep.*, 2020, **10**, 14691.
- 61 R. L. Hendrikse, C. Amador and M. R. Wilson, *Soft Matter*, 2023, **19**, 3590–3604.
- 62 A. Ben-Naim, *J. Phys. Chem.*, 1978, **82**, 792–803.
- 63 I. Levine, *Physical Chemistry*, McGraw-Hill Education, 2009, pp. 295–302.
- 64 M. A. Seaton, The DL_MESO Mesoscale Simulation Package, STFC Computational Science and Engineering Department, 2012, https://www.ccp5.ac.uk/DL_MESO.
- 65 J. Chang, *J. Chem. Phys.*, 2009, **131**, 074103.
- 66 W. Hayduk and C.-F. Wong, *Can. J. Chem. Eng.*, 1990, **68**, 653–660.
- 67 F. H. Allen, O. Kennard, D. G. Watson, L. Brammer, A. G. Orpen and R. Taylor, *J. Chem. Soc., Perkin Trans. 2*, 1987, S1–S19.
- 68 K. Kojima, S. Zhang and T. Hiaki, *Fluid Phase Equilib.*, 1997, **131**, 145–179.
- 69 S. A. Sanders, M. Sammalkorpi and A. Z. Panagiotopoulos, *J. Phys. Chem. B*, 2012, **116**, 2430–2437.
- 70 S. A. Sanders and A. Z. Panagiotopoulos, *J. Chem. Phys.*, 2010, **132**, 114902.
- 71 M. S. Hossain, S. Berg, C. A. S. Bergström and P. Larsson, *AAPS PharmSciTech*, 2019, **20**, 61.
- 72 Y. Ruiz-Morales and A. Romero-Martínez, *J. Phys. Chem. B*, 2018, **122**, 3931–3943.
- 73 M. A. Johnston, W. C. Swope, K. E. Jordan, P. B. Warren, M. G. Noro, D. J. Bray and R. L. Anderson, *J. Phys. Chem. B*, 2016, **120**, 6337–6351.
- 74 B. L. Bales, *J. Phys. Chem. B*, 2001, **105**, 6798–6804.
- 75 H. Lange and M. J. Schwuger, *Colloid Polym. Sci.*, 1968, **223**, 145–149.
- 76 H. Kumar and G. Kaur, *J. Dispersion Sci. Technol.*, 2021, **42**, 970–983.
- 77 R. Ranganathan, L. Tran and B. L. Bales, *J. Phys. Chem. B*, 2000, **104**, 2260–2264.
- 78 A. T. Gubaidullin, I. A. Litvinov, A. I. Samigullina, O. S. Zueva, V. S. Rukhlov, B. Z. Idiyatullin and Y. F. Zuev, *Russ. Chem. Bull.*, 2016, **65**, 158–166.
- 79 T. S. Banipal, H. Kaur and P. K. Banipal, *J. Mol. Liq.*, 2016, **218**, 112–119.
- 80 Z. Ettarhouni, F. Elarbei, A. Jangher and L. Abusen, *Am. J. Eng. Res.*, 2020, **9**, 118–126.
- 81 B. L. Bales, L. Messina, A. Vidal, M. Peric and O. R. Nascimento, *J. Phys. Chem. B*, 1998, **102**, 10347–10358.
- 82 P. B. Warren and A. Vlasov, *J. Chem. Phys.*, 2014, **140**, 084904.
- 83 R. D. Groot, *J. Chem. Phys.*, 2003, **118**, 11265–11277.
- 84 M. González-Melchor, E. Mayoral, M. E. Velázquez and J. Alejandre, *J. Chem. Phys.*, 2006, **125**, 224107.
- 85 P. B. Warren, A. Vlasov, L. Anton and A. J. Masters, *J. Chem. Phys.*, 2013, **138**, 204907.
- 86 M. E. Fisher and D. Ruelle, *J. Math. Phys.*, 1966, **7**, 260–270.
- 87 J. Lang, C. Tondre, R. Zana, R. Bauer, H. Hoffmann and W. Ulbricht, *J. Phys. Chem.*, 1975, **79**, 276–283.
- 88 A. Patist, J. R. Kanicky, P. K. Shukla and D. O. Shah, *J. Colloid Interface Sci.*, 2002, **245**, 1–15.
- 89 J. A. Mysona, A. V. McCormick and D. C. Morse, *Phys. Rev. E*, 2019, **100**, 012602.
- 90 E. Lavagnini, J. L. Cook, P. B. Warren and C. A. Hunter, *J. Phys. Chem. B*, 2022, **126**, 2308–2315.

