### Thermally Activated Delayed Fluorescence in Cu(I) Complexes Originating from Restricted Molecular Vibrations\*\*

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Abstract: The mechanism of thermally activated delayed fluorescence (TADF) in molecules in aggregated or condensed solid states has been rarely studied and is not well understood. Nevertheless, many applications of TADF emitters are strongly affected by their luminescence properties in the aggregated state. In this study, two new isomeric tetradentate Cu(I) complexes which simultaneously show aggregation induced emission (AIE) and TADF characteristics are reported for the first time. We provide direct evidence that effectively restricting the vibrations of individual molecules is a key requisite for TADF in these two Cu(I) complexes through in-depth photophysical measurements combined with kinetic methods, single crystal analysis and theoretical calculations. These findings should stimulate new molecular engineering endeavours in the design of AIE-TADF active materials with highly emissive aggregated states.

**R**ecent developments of thermally activated delayed fluorescence (TADF) materials have attracted tremendous attention as they can be used as highly efficient emitters for organic light emitting diodes (OLEDs).<sup>[1]</sup> TADF is a promising mechanism for converting electricity into light with an internal quantum efficiency (IQE) of nearly 100%, which utilizes the up-conversion from triplet excitons to singlet states by reverse intersystem crossing (RISC).<sup>[2]</sup> However,

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a stubborn problem for TADF materials in practical applications is that most TADF emitters suffer from aggregation-caused quenching (ACO) in the condensed phase. Therefore, effective TADF can be realized in a doped host-guest system only with careful control of their concentrations to restrict the nonradiative deactivation processes. Aggregation-induced emission (AIE) materials are, therefore, able to respond to these challenges. AIE molecules were first reported by Tang et al,<sup>[3]</sup> and while they show weak emission in dilute solution, upon aggregation they emit strongly, due to restriction of intramolecular rotation. Recently, Yasuda et al. reported a series of o-carborane derivatives that simultaneously show AIE and TADF characteristics.<sup>[4]</sup> OLEDs utilizing these molecules as a nondoped emission layer exhibited maximum EQEs as high as 11%. Thus, AIE-TADF active materials are promising luminescent materials and are widely used in numerous optoelectronic applications especially for non-doped OLEDs. However, the mechanism of TADF in aggregated molecules is still unclear and has not been studied in detail, although it is essential for many of the desired applications of these emitters. Fundamental factors that lead to enhanced emission of dyes in viscous and solidstate environments are of ongoing interest.<sup>[5]</sup>

Cu(I) complexes with TADF characteristics have been used in OLEDs and some derivatives show impressive EL performance.<sup>[6]</sup> Recently, dual-emissive Cu(I) complexes which can simultaneously show TADF and phosphorescent emission have been demonstrated.<sup>[7]</sup> Cu(I) complexes have the practical advantages of the relatively low-cost and high abundance of copper, in contrast to rare metals such as iridium.

The aim of the present study is to explore the coexistence of TADF and AIE in the new copper complexes 1 and 2. We demonstrate that both complexes show almost no emission in solution, but strong luminescence and obvious TADF is observed with an extremely short delayed lifetime in their aggregated state. To our knowledge, these combined photophysical phenomena have not been previously reported in Cu(I) complexes, although the coexistence of both TADF and AIE in all-organic molecules has been reported in a few cases.<sup>[8]</sup> The relationship between AIE and TADF is still not clear. Herein, we address the pertinent question: does aggregation induce or promote TADF? This topic is of fundamental importance for a better understanding of the mechanism of AIE-TADF and further designing this kind of material for use in OLEDs. The kinetics of TADF are investigated in two Cu(I) complexes in pristine film state and doped in PMMA matrices. The PMMA matrices of different molecular weight, doped with complexes 1 and 2, have been used in solid-state 'dilution' experiments to probe the role of intramolecular vibrations on the observation of TADF. The similar TADF behavior presented in both high molecular weight host (H-PMMA) and in pristine films clearly reveals that effectively restricting specific vibrations of individual emitter molecules is the origin of TADF for these Cu(I) complexes. Thus, a guideline for designing AIE-TADF active materials is presented.



*Figure 1.* Chemical structure of isomeric complexes **1** a) and **2** b). The steady steadystate fluorescence from 320 to 80 K in pristine film for **1** c) and **2** d), respectively. The steady steady-state fluorescence from 320 to 80 K in L-PMMA for **1** e) and **2** f), respectively. The steady-state fluorescence from 320 to 80 K in H-PMMA for **1** g) and **2** h), respectively. Temperature dependence of the integrated luminescence intensity as a function of temperature (K) for **1** i) and **2** j), integrated intensity normalized at 80 K.

The chemical structures of two new isomeric complexes 1 and 2 are shown in Figures 1a) and 1b). Their synthesis and characterization data are presented in the Supporting Information. The UV/vis absorption and emission spectra of 1 and 2 in degassed solution (CH<sub>2</sub>Cl<sub>2</sub>) and pristine film are depicted in Figure S6. Upon photoexcitation, 1 and 2 are almost non-emissive in pure CH<sub>2</sub>Cl<sub>2</sub> solution at room temperature. However, bright emission is observed in the pristine film state with a short luminescence decay time. This short-lived luminescence is in contrast with conventional Cu(I) complexes, where luminescence is observed with relatively long decay times (phosphorescence decay times of several 100 µs up to a few ms).<sup>[9]</sup> The luminescence decays of complexes 1 and 2 in their pristine film state are even shorter than the recently reported decay times in mono-nuclear Cu(I) complexes with TADF.[2c,6a-6c,7] This implies that 1 and 2 may be outstanding TADF materials with high reverse intersystem crossing rate  $(k_{RISC})$  by efficiently eliminating the triplet-triplet annihilation and triplet-polaron quenching in their pristine film state.<sup>[9]</sup> The detailed photophysical data of **1** and **2** are summarized in Table S1. The energy levels of the complexes calculated from their electrochemical and photophysical data are listed in Table S2.

To investigate the effect of molecular vibrations on the emission of **1** and **2** in their aggregated state, steady-state and time-resolved photoluminescence experiments were performed in their pristine film state, and in composite films of both complexes dispersed in PMMA hosts. Two PMMA samples, one with low molecular weight, L-PMMA (Mn = 15,000), and the other with high molecular weight, H-PMMA (Mn = 960,000), were used to fabricate thin films of **1** and **2**. Thermal gravimetric analysis (TGA) data showed that H-PMMA has a significantly higher Tg (130 °C) than L-PMMA (86 °C) consistent with H-PMMA being a stiffer material. The temperature dependence of the steady-state emission spectra in these films is compared with the temperature dependence of the emission in pristine films of **1** and **2** that are used as references. This allows the roles of thermal vibrations and intermolecular interactions on the excited state dynamics in these two complexes to be determined.

The two PMMA hosts with different molecular weight appear to be able to suppress molecular vibrations to different degrees. This results in marked differences of the temperature dependence of the photoluminescence of **1** and **2**, collected in pristine films: Figures 1c) and 1d); in L-PMMA Figures 1e) and 1f); and in H-PMMA Figures 1g) and 1h). Additionally, in Figures 1i) and 1j) the temperature dependence of the photoluminescence in the three samples shows clear differences in the temperature region above 170 K, but agree extremely well at temperatures below 170 K.

In the pristine film, the photoluminescence spectra of **1** and **2** show a dominant TADF mechanism between 320-170 K (Figures 1c and 1d). In this temperature range the emission intensity first increases and then decreases with decreasing temperature, showing a typical bell-shape that is characteristic of the equilibrium between the singlet and triplet states, mediated by the  $k_{RISC}$ , which drives the up-conversion of triplets back to the singlet manifold, and the thermal vibrations (internal conversion: IC) that quench the excited state population.<sup>[10]</sup>

Below 170 K, the luminescence intensity is dominated by phosphorescence from the lowest T<sub>1</sub> state, and is thus controlled by the decrease in the internal conversion, and as this continues decreasing with temperature, the steady-state emission slowly increases as observed in Figure 1i). Note also the change observed in the emission spectra in Figures 1c) and 1d). With decreasing temperature  $\lambda_{max}$  shifts to longer wavelengths. This is consistent with the emission being dominated by TADF at high temperatures, i.e. emission from the singlet excited state, and by phosphorescence at low temperatures, i.e. emission from the low energy triplet state. Remarkably, this behavior is not observed for the films of 1 and 2 in L-PMMA host. Here, the emission comes mainly from the triplet state and the contribution of TADF is practically non-existent. In this host the vibrational quenching dominates. The emission intensity, therefore, increases with decreasing temperature in the entire temperature range, and more markedly from 320 K to 170 K, where the effect of suppressing vibrations is more pronounced. This is presented in Figures 1e) and 1i).

In H-PMMA (Figures 1g and 1h), as in L-PMMA, the intermolecular interactions between 1 (or 2) molecules are practically non-existent. This, therefore, indicates that the H-PMMA host is more able to suppress vibrations, in the same way as the intermolecular interactions between 1 (or 2) molecules do in the pristine film. This enables the reverse intersystem crossing to become operative and to activate the TADF emission. However, in L-PMMA this is not the case, and the internal conversion dominates in the entire temperature range.

Below 170 K the temperature dependence of the photoluminescence is similar in all samples (Figures 1i and 1j). This shows that once  $k_{RISC}$  is suppressed, and as the effect of vibrations

decreases due to lowering the temperature, the differences in the host are not so important to the outcome of the emission. These experiments highlight the fact that the nonradiative pathways play a crucial role in the triplet harvesting mechanism in complexes 1 and 2. Moreover, in the case of 1, and to lesser extent in 2, the intermolecular interactions present in the pristine films, due to the molecular packing, are able to effectively suppress vibrations which otherwise will quench TADF.

Time resolved spectroscopy was used to follow the excited state dynamics in 1 and 2. The luminescence decays of 1 and 2 at 300 K in pristine films and dispersed in L-PMMA are shown in Figures S7a) and S7b). Both luminescence decays of 1 at 300 K are characterized by a fast component around 5 ns and 7.9 ns in pristine film and L-PMMA. respectively. These fast components are followed by a long bi-exponential decay, with time constants of 0.9 us and 5 us in the pristine film, and of 9 and 30 us in L-PMMA. The lifetime in L-PMMA is thus longer than in the pristine film. This reflects the effect of TADF in the pristine film, which at 300 K more rapidly quenches the triplet state, whereas in L-PMMA the emission occurs mainly directly from the triplet state, e.g. phosphorescence. These observations are consistent with the more intense emission in the pristine film below 1 µs, where TADF dominates. Complex 2 shows similar behavior (Figure S7b). Importantly, this reveals also how the presence of TADF accelerates the excited state decay in Cu(I) complexes and minimizes the roll-off effect in devices.



*Figure 2.* Temperature dependence of emission decays in L-PMMA for 1 a) and 2 b), respectively. Temperature dependence of emission decays in pristine film for 1 c) and 2 d), respectively. Power dependence of DF emission in pristine film for 1 e) and 2 f), respectively.

The decays obtained as a function of temperature for the pristine films, L-PMMA and H-PMMA films, and crystalline samples are entirely consistent with the relative importance of TADF. In pristine films, H-PMMA and crystalline samples both complexes 1 and 2 exhibit clear temperature dependence for the delayed fluorescence which is characteristic of the TADF mechanism (Figures 2c, 2d and S8). Here with decreasing temperature the emission intensity decreases and the lifetime increases, consistent with a slower depopulation of the triplet state. In contrast, in the L-PMMA host (Figures 2a and 2b) the photoluminescence decay is dominated by the nonradiative constants. This is observed in the temperature dependence of the prompt fluorescence component which increases in intensity with decreasing the temperature, and the increase of the phosphorescence lifetime in both compounds, as a direct response of suppressing the nonradiative decay pathways. Note that no TADF component is observed on these decays. In Figures 2e) and 2f) the power dependence of the delayed fluorescence for both compounds confirms the origin of the mechanism responsible for the long-lived fluorescence as TADF (slope 1) and not triplet-triplet annihilation (TTA) (slope 2).<sup>[10]</sup> These experiments were performed on pristine films, where the TADF component is stronger.

The two isomeric complexes 1 and 2 differ only in the substitution pattern on the tetrazolate ring, and this is shown to have a profound effect on the supramolecular structure and photophysical properties. It is instructive to compare the single-crystal X-ray structures of the two complexes. They both exhibit highly-distorted tetragonal coordination, typical of Cu(N^N)(POP)<sup>+</sup> complexes (Figures 3, S10 and S11).<sup>[6c]</sup> Crystals of 1 revealed multiple modes of intra- and intermolecular interactions leading to a fascinating 3D supramolecular structure (Figure 3). Important interactions are summarized in Table S5. Generally, the presence of strong intra- and intermolecular interactions would supply a more rigid environment and thereby effectively limit molecular distortions in the excited state, leading to bright emission through restricting the non-radiative processes.<sup>[6d,11]</sup> Furthermore, due to the small  $\Delta E(S_1-T_1)$  value obtained from photophysical experiments, and from the DFT calculations (see below), TADF is triggered in the pristine aggregation state of complex 1, which is similar to the situation in H-PMMA. It is worth noting that intermolecular aromatic stacking has been shown to offer a charge-transfer pathway and to enhance the carrier-transport ability, which is essential for excellent electroluminescence materials.<sup>[12]</sup>

In the crystal structure of 2, multiple modes of intra- and intermolecular interactions lead to a one-dimensional supramolecular chain structure (Figure S11).<sup>[13]</sup> Due to the relatively weak intermolecular interactions in 2, compared with 1 (Figure 3) complex 2 showed less distortion of the tetragonal coordination in the ground state. For the excited state, Riesgo et al. stated that the distortion of Cu complexes is directly related to their "Stokes-like" shift, that is, the difference in wavelength between absorption and emission spectra.<sup>[14]</sup> Indeed, the smaller Stokes-like shift for 1 compared with 2 (Figure S6 and Table S1) indicates that excitedstate distortion is suppressed more effectively in 1 which, in turn, means that for 1 the non-radiative process is suppressed. It could, therefore, be expected that the PLQY for 1 is substantially higher than 2. Indeed PLQYs of 47.1% and 9.4% were measured in the crystalline state for 1 and 2, respectively (Table S1).

Emission spectra of complexes 1 and 2 in amorphous pristine films and crystalline forms were collected in vacuum and in the presence of oxygen, showing only small variations of the luminescence intensity once oxygen has been removed (Figure S9). This indicates that oxygen quenching is not the cause for the lower PLQY of both complexes in their amorphous films, when compared with their crystalline forms, and in particular for the lower PLQY of complex 2 when compared with complex 1. Instead, the origin of the lower PLQY in 2 is due to the strong intermolecular interactions that are prevalent in complex 1 as indicated by the highly-ordered three-dimensional supramolecular chain structure observed in 1, but not in 2.



*Figure 3.* a) The one-dimensional supramolecular chain structure in 1. (Purple dashed lines represent intramolecular  $\pi \cdots \pi$  interactions and green dashed lines represent two kinds of intermolecular  $\pi \cdots \pi$  interactions). b) The cross-section view of the one-dimensional supramolecular chain structure unit in 1. c) The three-dimensional supramolecular structure of 1. Hydrogen atoms are omitted for clarity.

To further probe the interesting optoelectronic phenomena discussed above, we performed (Time-Dependent) Density Functional Theory ((TD)DFT) calculations on complexes 1 and 2. The electronic structures based on the optimized geometries of 1 and 2 plotted in Figure S12 show the clearly separate electronic occupations in the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO) for each complex, facilitating a small  $\Delta E(S_1-T_1)$  which is a characteristic of TADF materials.<sup>[15]</sup>

Since the value of  $\Delta E(S_1-T_1)$  is usually used to assess the ability of reverse intersystem crossing from  $T_1$  to  $S_1$ ,<sup>[16]</sup> it was evaluated here for every structure (IM, LPM and CM) presented in Figure 4 to explain the change of luminescence properties in the different situations mentioned above. The nomenclature IM, LPM and CM defines the optimized isolated molecule, the simulated molecule in low PMMA (L-PMMA) host, and the molecule in the crystal structure, respectively. As mentioned above, the complex in L-PMMA host has more freedom to rotate the phenyl group than in the crystal structure. Therefore, an important dihedral angle  $\theta$ , which corresponds to the nearest phenyl group to the tetrazole ring, as a representative angle, was changed and scanned for 1 and 2. The results collected in Table S6 show: (i) both the IM and CM structures have a smaller  $\Delta E(S_1-T_1)$  value, further verifying that TADF may occur in isolated molecules (solution state) and in the aggregation state (crystal structure). However, the huge difference in optimized geometries between ground state and S<sub>1</sub> along with T<sub>1</sub> collected in Table S7 also illustrate the large nonradiative decay of single molecules of both complexes in solution. This is crucial for the quenching of TADF in the solution state. (ii) With the increase of  $\theta$ ,  $\Delta E(S_1-T_1)$  becomes larger for both 1 and 2, suggesting the rotation of the nearest phenyl group to the tetrazole ring raises  $\Delta E(S_1-T_1)$  relative to the crystal structure, which might greatly reduce the ability for transfer from the  $T_1$  to the  $S_1$  state. Thus for 1 and 2 phosphorescent emission, rather than TADF, appears in the L-PMMA environment where the molecules are free to rotate their phenyl groups. Hence, the relationship between restriction of molecular vibrations and the luminescent properties of the two Cu(I) complexes are summarized in Figure 4b.



*Figure 4.* a) Diagram of three models (**IM**, **LPM** and **CM**) for **1** and **2**. The **IM**, **LPM** and **CM** represent the optimized isolated molecule, simulated molecule in low PMMA host material, and the molecule in the crystal structure, respectively. Hydrogen atoms are omitted for clarity. b) The relationship between restriction of molecular vibrations and luminescent properties of the two Cu(I) complexes.

Moreover, a singlet-triplet gap around 0.2 eV, as predicted in the calculations, is in excellent agreement with the energy difference determined from the onset of the emission spectra in pristine films (see Figures 1c) and 1d)).

In summary, we have studied two new AIE-TADF active isomeric Cu(I) complexes. Solid-state 'dilution' experiments investigated their photophysical properties at different degrees of suppression of molecular vibrations, revealing phosphorescence in low MWt PMMA (L-PMMA), but efficient TADF in high MWt PMMA (H-PMMA). The similar TADF behavior in H-PMMA and in the pristine aggregated state (neat thin films) suggests that TADF behavior is strongly influenced by effectively restricting the vibration of individual molecules through intermolecular interactions. TD-DFT calculations are in excellent agreement with the observed photophysical properties of the complexes. Our findings may stimulate new molecular engineering endeavours in the design of AIE-TADF active materials with highly emissive aggregated states. Moreover, the extremely short delayed lifetime in pristine film state suggests that both complexes 1 and 2 are candidates for non-doped OLEDs.

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# **Communications**

#### **Photochemistry**

G. Li, R. S. Nobuyasu, B. Zhang, Y. Geng, B. Yao, Z. Xie, D. Zhu\*, G. Shan, W. Che, L. Yan,
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Thermally Activated Delayed Fluorescence in Cu(I) Complexes Originating from Restricted Molecular Vibrations



**Does aggregation induce or promote TADF?** Two new isomeric AIE-TADF active Cu(I) complexes have been studied. We provide direct evidence that aggregation is able to promote TADF by suppressing vibrational quenching in these two Cu(I) complexes. Phosphorescence and TADF could be realized respectively in doping with two different matrix PMMA films due to different degrees of restricting molecular vibrations. These findings could open a new avenue in the design of AIE-TADF active materials.

# **Supporting Information**

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#### I. Experimental - general information

Materials obtained from commercial suppliers were used without further purification unless otherwise stated. All glassware, syringes, magnetic stirring bars, and needles were thoroughly dried in a convection oven. Reactions were monitored using thin layer chromatography (TLC). Commercial TLC plates were used and the spots were visualized under UV light at 254 and 365 nm. <sup>1</sup>H NMR spectra were recorded at 25 °C on a Varian 500 MHz spectrometer. The chemical shifts ( $\delta$ ) are given in parts per million relative to internal standard TMS (0 ppm for <sup>1</sup>H). <sup>31</sup>P NMR spectra were recorded at 25 °C on a 600 MHz spectrometer. The molecular weights of the complexes were obtained by using matrix-assisted laser desorption-ionization time-of-flight (MALDI-TOF) mass spectrometry. Elemental analysis was obtained using a Flash EA1112 analyzer. UV-vis absorption spectra were recorded on a Shimadzu UV-3100 spectrophotometer. PL efficiencies were measured with an integrating sphere (C-701, Labsphere Inc.), with a 365 nm Ocean Optics LLS-LED as the excitation source, and the laser was introduced into the sphere through the optical fiber. Crystal structure data for complexes **1** and **2** were collected on a Bruker Smart Apex II CCD diffractometer with graphite-monochromated Mo K $\alpha$  radiation ( $\lambda = 0.71073$  Å) at room temperature. CCDC numbers 1524132 for **1** and 1524133 for **2**.

#### Synthesis of (2-pyridyl)tetrazole (L1)

L1 was synthesized according to the literature method.<sup>1</sup> To a 250 mL round-bottomed flask was added the 2-cyanopyridine (20 mmol), sodium azide (1.43 g, 22 mmol), zinc bromide (4.50 g, 20 mmol), and water (40 mL). The reaction mixture was refluxed for 24 h, cooled to room temperature and basified by addition of 2.5 equiv. of NaOH, filtered, acidified by diluted hydrochloric acid to pH = 6.5, and filtered, and the resulting solid was washed with water to give L1 (2.1 g; 77% yield). <sup>1</sup>H NMR (500 MHz, DMSO-d<sub>6</sub>,  $\delta$  [ppm]): 8.79 (m, 1H), 8.22 (d, 1H, *J* = 9.5 Hz), 8.08 (m, 1H), 7.62 (m, 1H); MS: (MALDI-TOF) [m/z]: 149.06 (M<sup>+</sup>) (calcd: for 148.06).

# Synthesisof2-(2-methyl-2H-tetrazol-5-yl)pyridine(L2)and2-(1-methyl-1H-tetrazol-5-yl)pyridine (L3)

L2 and L3 were synthesized according to the literature method.<sup>2</sup>L1 (1.0 g, 6.8 mmol), iodomethane (3.85 g, 27.1 mmol) and sodium hydroxide (0.68 g, 16.96 mmol) were dissolved in dimethylformamide (15 mL). The mixture was stirred for 24 h at room temperature. Water (100 mL) was added to the solution then extracted with dichloromethane (400 mL). The organic layer was

separated and the mixture was then purified by column chromatography on silica gel (200–300 mesh) with petroleum ether/ethyl acetate (10:3 v/v) to give **L3** (0.52 g, 48% yield) and then using the same solvent system (5:8 v/v) as the eluent to give **L2** (0.316 g, 29% yield). <sup>1</sup>H NMR (500 MHz, CDCl<sub>3</sub>-d<sub>6</sub>,  $\delta$  [ppm]): 8.65 (s, 1H), 8.10 (d, *J* = 7.5 Hz, 1H), 7.74 (t, *J* = 7.5 Hz, 1H), 7.28 (d, *J* = 5.5 Hz, 1H), 4.34 (s, 3H) (**L2**); 8.75 (d, *J* = 4 Hz, 1H), 8.33 (d, *J* = 8.5 Hz, 1H), 7.93 (t, *J* = 7.5 Hz, 1H), 7.48 (m, 1H), 4.52 (s, 3H) (**L3**).

#### Synthesis of complex 1

Complex **1** was prepared according to the following general procedure: A mixture of  $[Cu(CH_3CN)_4]BF_4$  (0.32 g, 1.0 mmol) and POP (0.54 g, 1.0 mmol) in CH<sub>2</sub>Cl<sub>2</sub> (10 mL) was stirred at room temperature for 1 h and then the diimine ligand L**2** (0.16 g, 1.0 mmol) was added. The reaction mixture was stirred for another 2 h, and the solvent was then evaporated; the crude product was dissolved in a small amount of dichloromethane and passed through a short column of neutral aluminum oxide. The solvent was removed to obtain complex **1** (0.70 g, 80% yield). <sup>1</sup>H NMR (500 MHz, DMSO-d<sub>6</sub>,  $\delta$  [ppm]): 8.48 (d, *J* = 5.0 Hz, 1H), 8.24 (d, *J* = 5.5 Hz, 1H), 8.16 (t, *J* = 9.0 Hz, 1H), 7.62 (t, *J* = 5.5 Hz, 1H), 7.43–7.32 (m, 14H), 7.13–7.08 (m, 12H), 6.74–6.70 (m, 2H), 4.44 (s, 3H). <sup>31</sup>P NMR (600 MHz, CDCl<sub>3</sub>,  $\delta$  [ppm]): -12.4 (s). MS (MALDI-TOF) [m/z] : 601.1 (1-BF<sub>4</sub>–-L**2**). Anal. Calcd for C<sub>43</sub>H<sub>35</sub>BCuF<sub>4</sub>N<sub>5</sub>OP<sub>2</sub>: C, 60.76; H, 4.15; N, 8.24. Found: C, 60.51; H, 4.23; N, 8.33. Crystals for X-ray analysis were grown by slow evaporation of a solution of complex **1** in CH<sub>2</sub>Cl<sub>2</sub> and MeOH mixture.

#### Synthesis of complex 2

The synthetic procedure of complex **2** was the same as complex **1**, except ligand L**3** (0.16 g, 1.0 mmol) was used instead of L**2**. Yield: 0.72 g, 82%. <sup>1</sup>H NMR (500 MHz, DMSO-d<sub>6</sub>,  $\delta$  [ppm]): 8.48 (d, J = 5.0 Hz, 1H), 8.24 (d, J = 5.5 Hz, 1H), 8.16 (t, J = 9.0 Hz, 1H), 7.62 (t, J = 5.5 Hz, 1H), 7.43–7.32 (m, 14H), 7.13–7.08 (m, 12H), 6.74–6.70 (m, 2H), 4.44 (s, 3H). <sup>31</sup>P NMR (600 MHz, CDCl<sub>3</sub>,  $\delta$  [ppm]): -12.0 (s). MS (MALDI-TOF) [m/z] : 601.1 (**2**-BF<sub>4</sub><sup>-</sup>-L**3**). Anal. Calcd for C<sub>43</sub>H<sub>35</sub>BCuF<sub>4</sub>N<sub>5</sub>OP<sub>2</sub>: C, 60.76; H, 4.15; N, 8.24. Found: C, 60.45; H, 4.18; N, 8.17. Crystals for X-ray analysis were grown by slow evaporation of a solution of complex **2** in CH<sub>2</sub>Cl<sub>2</sub> and MeOH mixture.







Figure S2. <sup>1</sup>H NMR spectrum of complex 2.



Scheme S1. Synthesis of L1, L2, L3, complex 1 and complex 2.



Figure S3. <sup>31</sup>P NMR spectra of complex 1 and complex 2.



Figure S5. MALDI-TOF spectrum of complex 2.

#### **II.** Photophysical properties

**Photophysical Characterization.** Thin films of Cu(I) complexes **1** and **2** in a matrix were prepared by spin-coating a blend of complex : matrix (1:20 w/w). L-PMMA (average Mw ~15,000) and H-PMMA (average Mw ~960,000) were obtained from Sigma-Aldrich. Thin films of pristine Cu(I) complexes were prepared by spin-coating from DCM solutions (10 mg/mL). Steady-state photoluminescence data were collected using a Fluorolog fluorescence spectrometer (Jobin Yvon). Temperature dependence studies were performed using a cryostat with a liquid nitrogen cooling system. Phosphorescence, prompt fluorescence (PF), delayed fluorescence (DF) and decays were recorded using nanosecond gated luminescence and lifetime measurements (from 1 ns to 1 s) with either a high energy pulsed Nd:YAG laser emitting at 355 nm (EKSPLA) or a N2 laser emitting at 337 nm. Emission was focused into a spectrograph and detected on a sensitive gated iCCD camera (Stanford Computer Optics) having a sub-nanosecond resolution. PF/DF time resolved measurements were performed by increasing exponentially the gate and the delay times.



Figure S6. UV/vis absorption and emission spectra of 1 and 2 in degassed solution  $(CH_2Cl_2)$  and pristine film at room temperature.

Complex	$\lambda_{abs}^{a}$ (nm)	$\lambda_{\rm em}^{b,c,d}$ (nm)	$\tau^{b}$ (µs)	${{{ { { { \Phi } } } } }_{{ m{L}}}}^{b,e}}\left(\% ight)$
1	264, 358 <sub>sh</sub>	533, 516, 516	0.90, 5.0	15.5, 47.1
2	269, 386	572, 556, 556	0.95, 4.2	2.9, 9.4

Table S1. Photophysical data for complexes 1 and 2 at room temperature

<sup>*a*</sup> Measured in CH<sub>2</sub>Cl<sub>2</sub> ( $1.0 \times 10^{-5}$  M). <sup>*b*</sup> Measured in the pristine film; error  $\pm$  5%. <sup>*c*</sup> Measured in low molecular weight (L-PMMA) film. <sup>*d*</sup> Measured in the high molecular weight (H-PMMA) film. <sup>*e*</sup> Measured in the crystalline state.

Table S2.	Energy	levels a	nd S <sub>1</sub> -T <sub>1</sub>	gaps of tl	he comple	exes 1 and	12
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Complex	$\Delta E_{g}^{a}$ (eV)	HOMO <sup>b</sup> (eV)	LUMO <sup>c</sup> (eV)	$\mathbf{S}_{1}^{d}(\mathbf{eV})$	$T_1^e$ (eV)	$\Delta E_{ST} (eV)$
1	3.0	-5.5	-2.5	2.75	2.583	0.167
2	2.72	-5.49	-2.77	2.719	2.50	0.219

<sup>*a*</sup> Estimated from the onset wavelengths of the absorption spectra measured in  $CH_2Cl_2$ . <sup>*b*</sup> Calculated from the oxidation potentials obtained by cyclic voltammetry. <sup>*c*</sup>Calculated from the HOMO energy levels and E<sub>g</sub>. <sup>*d*</sup>Estimated from the onset wavelengths of the 298 K emission spectra measured as pristine film. <sup>*c*</sup>Estimated from the onset wavelengths of the 77 K emission spectra measured as pristine film.



**Figure S7.** The fluorescence decays collected in pristine and L-PMMA films at room temperature for complexes **1** a) and **2** b), respectively.



Figure S8. Temperature dependence of emission decays of crystals and H-PMMA films of 1 and 2.



**Figure S9**. Steady state photoluminescence spectra under vacuum and in air, for complexes **1** and **2** in pristine film and in crystal.

#### III. X-ray crystallographic data

The molecular structures of complexes **1** and **2** were confirmed by X-ray crystallographic analysis of single crystals. Diffraction data were collected on a Bruker SMART Apex CCD diffractometer using  $k(Mo-K\alpha)$  radiation (k = 0.71073 Å). Cell refinement and data reduction were made by the SAINT program. The structure was determined using the SHELXTL/PC program. Figure S10 shows Oak Ridge thermal ellipsoid plot (ORTEP) drawings of complexes **1** and **2**. The crystallographic data have been deposited with the Cambridge Crystallographic Data Centre with CCDC deposition numbers 1524132 and 1524133. These data can be obtained free of charge from The Cambridge Crystallographic Data Centre via <u>www.ccdc.cam.ac.uk/data\_request/cif</u>.

	1
Empirical formula	$C_{43}H_{35}BCuF_4N_5OP_2$
Formula weight	850.05
Temperature (K)	293(2)
Crystal system	Monoclinic
space group	P2(1)/c
a /Å	16.358(3)
b /Å	11.479(2)
c /Å	23.685(5)
α /°	90.00
β /°	106.59(3)
γ /°	90
$V/Å^3$	4262.5(15)
Ζ	4
$\rho_{calc}(g/cm^3)$	1.325
$\mu/\text{mm}^{-1}$	0.644
R <sub>int</sub>	0.0623
Goodness-of-fit on $F^2$	1.007
$R_1^{a}, wR_2^{b} [I > 2\sigma(I)]$	0.0702, 0.1866
$R_1$ , w $R_2$ (all data)	0.1113, 0.2074

 Table S3. Crystal data and structure refinement for 1

 $\overline{{}^{a} R_{1} = \Sigma ||Fo| - |Fc|| / \Sigma |Fo|} wR_{2} = \{ \Sigma [w(Fo^{2} - Fc^{2})^{2}] / \Sigma [w(Fo^{2})^{2}] \}^{1/2}$ 

Table S4. Crystal data and structure refinement for 2

	2
Empirical formula	$C_{43}H_{35}BCuF_4N_5OP_2$
Formula weight	850.05
Temperature (K)	296(2)
Crystal system	Orthorhombic
space group	Fddd
a /Å	25.1061(14)
b /Å	30.1001(14)
c /Å	48.205(3)

$\alpha$ /°	90.00
β /°	90
γ /°	90
$V/Å^3$	36428(4)
Z	32
$\rho_{\text{calc}}(g/cm^3)$	1.240
$\mu/mm^{-1}$	0.603
R <sub>int</sub>	0.1324
Goodness-of-fit on F <sup>2</sup>	0.977
$R_1^{a}, wR_2^{b} [I > 2\sigma(I)]$	0.0786, 0.2027
$R_1$ , w $R_2$ (all data)	0.1661, 0.2584
$R_1$ , w $R_2$ (all data)	0.1661, 0.2584

<sup>*a*</sup>  $R_1 = \Sigma ||Fo| - |Fc|| / \Sigma |Fo|$ . <sup>*b*</sup>  $wR_2 = \{ \Sigma [w(Fo^2 - Fc^2)^2] / \Sigma [w(Fo^2)^2] \}^{1/2}$ 

![](_page_17_Figure_2.jpeg)

**Figure S10.** ORTEP drawings of **1** and **2**. The BF<sub>4</sub> counterion, H atoms and solvent molecules have been omitted for clarity. Selected bond lengths (Å) and angles (°): For **1** Cu–N1 2.059(4), Cu–N5 2.120(5), Cu–P1 2.280(14), Cu–P2 2.2136(14); N1–Cu–N5 79.09(18), P1–Cu–P2 119.92(5), N1–Cu–P1 101.63(13), N1–Cu–P2 129.26(13), N5–Cu–P1 105.41(12), N5–Cu–P2 112.82(13). For **2** Cu–N1 2.041(5), Cu–N5 2.098(6), Cu–P1 2.2480(19), Cu–P2 2.2381(18); N1–Cu–N5 79.2(2), P1–Cu–P2 113.63(7), N1–Cu–P1 118.14(17), N1–Cu–P2 118.03(16), N5–Cu–P1 107.78(16), N5–Cu–P2 114.76(16).

![](_page_18_Figure_0.jpeg)

**Figure S11.** The one-dimensional supramolecular chain structure unit in **2** (Purple dashed lines represent intramolecular  $\pi \cdots \pi$  interactions, blue dashed lines represent intermolecular C-H $\cdots \pi$  interactions and green dashed lines represent intermolecular  $\pi \cdots \pi$  interactions).

1		2	
Intramolecular $\pi \cdots \pi$ stacking	3.6 Å	Intramolecular $\pi \cdots \pi$ stacking	3.5 Å
Intermolecular $\pi \cdots \pi$ stacking	3.5 Å	Intermolecular C-H $\cdots \pi$ interaction	2.5 Å
Intermolecular head-to-tail stacking	3.7 Å	Intermolecular head-to-tail stacking	3.9 Å

Table S5. The distances of intra- or intermolecular interactions for 1 and 2

#### **IV. Quantum calculations**

The single molecule geometries in the ground state and the first triplet excited state (T<sub>1</sub>) were optimized by employing the B3LYP exchange-correlation functional. The optimal geometries of the first singlet excited state (S<sub>1</sub>) and the energy gap between S<sub>1</sub> and T<sub>1</sub>, namely  $\Delta E(S_1-T_1)$ , were obtained using TDDFT with the B3LYP functional. For all the calculations, the basis set 6-31G(d,p) was adopted for non-metallic atoms, O, N, C, P, H atoms, and the basis set LAND2DZ was applied on Cu atoms. The Gaussion 09 software was used for all calculations.

![](_page_19_Figure_0.jpeg)

**Figure S12**. The frontier molecular orbitals of **1** and **2** based on optimized ground-state geometries calculated at B3LYP/6-31G(d,p) (LAND2DZ for Cu atoms) level.

Table S6. The $\Delta E(S_1\text{-}T_1)$ (eV) values of three m	odels (IM, LPM and CM) for 1 and 2 obtained
by B3LYP/6-31G(d,p) (LAND2DZ for Cu atoms	s) method. For a definition of $\theta$ see Figure 4.
1	2

1				2	
<b>IM 1</b>	0.136		IM 2	0.1	77
	$\theta = 92$	0.185		$\theta = -6$	0.271
	$\theta = 97$	0.202		$\theta = -11$	0.277
	$\theta = 102$	0.224		$\theta = -16$	0.283
	$\theta = 107$ $\theta = 112$	0.248		$\theta = -21$	0.290
		0.271		$\theta = -26$	0.297
LPM 1	$\theta = 117$	0.291	LPM 2	$\theta = -31$	0.303
	$\theta = 122$	0.307		$\theta = -36$	0.308
	$\theta = 127$	0.319		$\theta = -41$	0.311
	$\theta = 132$	0.326		$\theta = -46$	0.314
	$\theta = 137$	0.325		$\theta = -51$	0.315
	$\theta = 142$	0.316		$\theta = -56$	0.315
CM 1	0.1	85	CM 2	0.2	71

8		0	-				
	1				2		
	$S_0$	$\mathbf{S}_1$	$T_1$		$S_0$	$\mathbf{S}_1$	$T_1$
Cu1-P2	2.379	2.408	2.433	Cu1-P2	2.355	2.465	2.447
Cu1-P3	2.363	2.487	2.456	Cu1-P3	2.379	2.456	2.444
Cu1-N5	2.202	2.169	2.074	Cu1-N5	2.158	2.087	2.022
Cu1-N9	2.215	2.059	1.990	Cu1-N9	2.241	2.119	2.028
P2-Cu1-N9	113.5	143.4	141.8	P2-Cu1-N9	114.0	101.8	101.7
P3-Cu2-N5	119.5	123.1	128.1	P3-Cu2-N5	108.0	96.8	97.0
C20-P2-Cu1-N9	166	159.5	164.9	C20-P2-Cu1-N9	-24.3	-29.4	-26.0
C21-C20-P2-Cu1	-34.1	-38.5	-44.8	C21-C20-P2-Cu1	-123.6	-116.3	-114.0

Table S7. The optimized bond lengths (Å), bond angles (°) and dihedral angles (°) of 1 and 2 at ground (S<sub>0</sub>) state and S<sub>1</sub> state along with T<sub>1</sub> state

#### V. Electrochemical data

Cyclic voltammetry was performed in a gas-tight single-compartment three-electrode cell with a BAS Epsilon Electrochemical Analyzer at room temperature. A glassy carbon disk and a platinum wire were the working and auxiliary electrodes, respectively. The reference electrode was  $Ag/Ag^+$  (0.1 M AgNO<sub>3</sub> in acetonitrile). The CV measurements were carried out in anhydrous and nitrogen-saturated acetonitrile solutions with 0.1 M *n*-tetrabutylammonium perchlorate (TBAP) and 1.0 mM cuprous complex. The ferrocenium/ferrocene couple was used as an internal standard.

![](_page_20_Figure_4.jpeg)

Figure S13. Cyclic voltammograms of ferrocenium/ferrocene couple (as an internal standard) and complexes 1 and 2 measured in acetonitrile in the presence of 0.1 M n-tetrabutylammonium perchlorate (TBAP) as supporting electrolyte with 100 mV/s scan speed.

#### **VI. References for SI**

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