



# **Review Iron Sulfide Materials: Catalysts for Electrochemical Hydrogen Evolution**

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**Abstract:** The chemical challenge of economically splitting water into molecular hydrogen and oxygen requires continuous development of more efficient, less-toxic, and cheaper catalyst materials. This review article highlights the potential of iron sulfide-based nanomaterials as electrocatalysts for water-splitting and predominantly as catalysts for the hydrogen evolution reaction (HER). Besides new synthetic techniques leading to phase-pure iron sulfide nano objects and thin-films, the article reviews three new material classes: (a)  $FeS_2$ -TiO<sub>2</sub> hybrid structures; (b) iron sulfide-2D carbon support composites; and (c) metal-doped (e.g., cobalt and nickel) iron sulfide materials. In recent years, immense progress has been made in the development of these materials, which exhibit enormous potential as hydrogen evolution catalysts and may represent a genuine alternative to more traditional, noble metal-based catalysts. First developments in this comparably new research area are summarized in this article and discussed together with theoretical studies on hydrogen evolution reactions involving iron sulfide electrocatalysts.

Keywords: iron sulfide; nanomaterials; electrocatalysts; hydrogen evolution; water-splitting

# 1. Introduction

On our planet, atomic hydrogen H predominantly exists chemically bonded in natural molecules and products such as water, petroleum, and coal. In contrast, molecular hydrogen H<sub>2</sub> does not occur naturally in large amounts on Earth and is only produced by certain microorganisms [1] in small quantities, which escape Earth's gravity into the atmosphere due to the low molecular weight. Despite the low natural abundance of hydrogen gas  $H_2$  on Earth, it has indisputably been identified as a major energy carrier of the future as H<sub>2</sub> has the highest gravimetric energy density of all known substances (lower heating value of  $\sim 120 \text{ kJ/g}$ ) [2], is non-poisonous and, most importantly, hydrogen gas can be derived from water. Using water, which is often termed as "the renewable fuel", as a sustainable and abundant energy source has been a dream of novelists and scientists for a long time [3] and for several decades researchers of various disciplines have been working on energetically favorable and cheap processes to catalyze the decomposition reaction of water into clean molecular hydrogen and oxygen. The "Steam-Iron" process  $(3Fe + 4H_2O \rightarrow Fe_3O_4 + 4H_2)$  is one of the oldest commercial methods for producing hydrogen from water and was practiced at the beginning of the 20th century to generate small quantities of pure hydrogen [4]. Remarkably, a recent study has found that traces of HS<sup>-</sup> in the process forming FeS on the iron catalyst surface, can significantly enhance the hydrogen evolution reaction (HER) [5].

In general, the water-splitting process consists of two half-reactions: the water oxidation (oxygen evolution reaction, OER) and the water reduction (hydrogen evolution reaction, HER). Regardless of the pH of the solution, the voltage required for the splitting of H<sub>2</sub>O into H<sub>2</sub> and  $\frac{1}{2}$  O<sub>2</sub> is 1.23 V (25 °C, 1 atm). However, in reality one must apply voltages higher than the theoretical value of 1.23 V to achieve electrochemical water-splitting. This excess potential (overpotential,  $\eta$ ) is required to

overcome intrinsic activation barriers at the electrodes and solution resistance [6]. Consequently, an energy-efficient water-splitting system should have a low  $\eta$  value and much effort is made towards developing powerful electrode materials with low overpotential.

A key step in hydrogen gas formation from an aqueous solution is the reduction of H<sup>+</sup> on a catalyst's surface with an electron to form surface-adsorbed hydrogen. Subsequently, for the catalyst to be efficient, fast liberation of the hydrogen gas from the catalyst surface is essential. In this context, the free energy of hydrogen adsorption,  $\Delta G_H$ , is a significant measure for the catalytic activity of a hydrogen-evolving electrocatalyst [7]. Ideally, for a catalyst this energy (which must be invested to release molecular hydrogen from the catalyst surface) should be nearly zero. Such ideal  $\Delta G_H$  values are observed for elemental platinum and consequently, platinum-based materials have been found to be the most efficient hydrogen evolution catalysts [8]. Due to the high cost and low natural abundance of platinum in the Earth crust ( $3.7 \times 10^{-6}$ %), which is orders of magnitudes lower than that of non-precious metals such as Ni (0.0089%) or Fe (6.8%), much effort is devoted to find alternative, noble metal-free hydrogen evolution catalysts [6].

In recent years, significant progress has been made in developing innovative, less-expensive catalysts for HER whereby especially new nanomaterial-based catalysts exhibit remarkable HER efficiencies [9]. Inspired by natural photosynthesis, in which water and carbon dioxide are converted into molecular oxygen and carbohydrates, a significant research focus has been on the development of artificial photocatalysts, which could facilitate the light-driven generation of hydrogen from water. Significantly, also the biochemical processes taking place during photosynthesis in photosystems I and II are based on the redox chemistry of non-noble metals such as manganese and calcium or the Rieske iron sulfur protein, which contains a [2Fe-2S] cluster [9]. Other prominent examples from nature are hydrogenases, a group of metalloenzymes that catalyze the oxidation of  $H_2$  into protons and electrons as well as the reverse reaction, the generation of  $H_2$  [10]. Inspired by these enzymes, researchers also work on the development of molecular iron sulfur compounds to employ them as "artificial enzymes" in catalysis [11]. Even though significant progress is made in this research area, often such molecular species are prone to decomposition in solution.

Thus, in line with these systems, there has been a growing interest in the biosynthesis of related nanomaterials and the most extensive research has been focused on metal sulfides. For example, bacteria, fungi, or other microorganisms can convert sulfate into reduced sulfide species, which react with metal cations to generate metal sulfide structures [12]. However, typically during these processes it is difficult to control the purity and shape of the forming nanostructure.

Therefore, mimicking such biological processes and developing more sophisticated artificial synthetic procedures for transition metal oxide or sulfide nanostructures has been the aim of many research studies. In this context, in particular the development of transition metal dichalcogenide (TMDC) materials ( $MX_2$ , M: e.g., Mo, Co, Fe, Ni; X = O, S, Se, Te) has attracted the attention of many research groups. Due to superior electrochemical performance, high stability in acid conditions and low fabrication costs TMDC materials exhibit excellent properties as water-splitting catalysts [13–17]. The performance of TMDC catalyst materials heavily depends on their electronic structure and to reach superior catalytic properties it is essential to pull down the Fermi level or decrease the band gap of the structures. Typical strategies to do this include doping of the TMDC lattice with impurity atoms or compounds and intentional creation of structural defects. Theoretical studies show that the performance of TMDC-based HER electrocatalysts severely depends on the materials' electronic energy and that  $\Delta G_H$  is an indicator to predict the catalytic activity based on the materials' electronic structure [18].

Due to the high natural abundance, very low cost and low toxicity of iron and sulfur, the TMDC FeS<sub>2</sub> and other iron sulfide structures are very desirable catalyst materials. However, while iron sulfide structures have been thoroughly studied and have been shown to be good semiconductors for photovoltaic devices [14,19,20], iron sulfide-based catalysts for HER are much less developed. Compared to benchmark precious metal-free HER catalysts such as MoS<sub>2</sub> [21] or transition metal

phosphides [22], typically iron sulfide catalysts are less efficient and prone to corrosion. For example, a reactivity study on  $Fe_xS_y$  films shows their tendency to transform to oxidation products [23]. Thus, regarding the obvious benefits of iron sulfide materials there is an urgent need to develop more efficient and stable iron sulfide catalyst materials for HER. It is to be hoped that the stability of new iron sulfide catalysts can be increased through rational design of nanostructured materials and surfaces.

The chemistry of iron sulfides is very complex, with the formation of diverse iron sulfide phases such as FeS,  $Fe_{1-x}S$ ,  $FeS_2$ ,  $Fe_3S_4$ , etc. being very sensitive to factors such as temperature, pressure or pH, and is still not fully understood [24]. The recent discovery of an entirely new nanoparticulate iron sulfide phase,  $FeS_{nano}$ , illustrates the complexity of the field.  $FeS_{nano}$  contains tetrahedral iron sites and is balanced by monosulfide and polysulfide sulfur species, which together dramatically affect the stability and enhance the reactivity of  $FeS_{nano}$  making it highly desirable as a potential catalyst material [25]. Moreover,  $FeS_{nano}$  has been identified as a crucial precursor for the iron sulfide material mackinawite (FeS), a metastable intermediate structure, which itself can be a precursor to the formation of the more stable  $Fe_3S_4$  and  $FeS_2$  phases. Due to the small size of constituent particles and highly disordered domains, mackinawite nanostructures can be highly reactive and therefore useful for the design of new catalyst materials [26].

In general, the fabrication of effective iron sulfide-based catalysts requires the synthesis of materials exhibiting defined iron sulfur ratios (phase purity) and well-defined shapes (spheres, cubes, wires, etc.) [27]. Regarding catalysis, both phase purity and morphology are of major significance for controlling and improving the performance of the catalyst material. The chemistry and syntheses of iron sulfide nanostructures [28] and thin-films [29,30], with their enormous potential for photovoltaic devices [31,32] and electrode materials [33] have been extensively reviewed in the literature. Therefore, the following article does not provide a comprehensive collection of research developments in this area, but rather highlights the latest milestones to spark the scientific communities' interest for iron sulfide electrocatalysts. While the main focus is on iron sulfide-catalyzed HER, some examples of OER and iron sulfide-catalyzed overall water-splitting processes will be discussed. The first part of the article highlights new synthetic approaches to various iron sulfide nanostructures (e.g., Fe<sub>3</sub>C<sub>4</sub>, etc.) and reflects how much their electrochemical activity depends on the structure and phase purity of the materials. In the next section, the concept of iron disulfide-doped titanium dioxide structures and the impact of FeS<sub>2</sub>-doping on the photocatalytic performance of TiO<sub>2</sub> will be discussed. Furthermore, the review summarizes recent milestones in the area of iron sulfide-2D carbon nanomaterial hybrid HER catalysts and concludes with a few examples of metal-doped iron sulfide electrocatalysts. Appendix A of this review contains Table A1 of all discussed iron sulfide materials, summarizing experimental procedures and HER performance data.

#### 2. Iron Sulfide Phases as Electrocatalysts

#### 2.1. Iron Monosulfide, FeS

Recently Zhang and co-workers discovered the potential of iron monosulfide FeS as a low-cost pre-electrocatalyst for water-splitting [34]. Through treatment of an iron foam (IF) with thiourea under hydrothermal conditions they obtained FeS nanosheets, which grew on the iron foam. This material FeS@IF alone (just like FeS or IF) shows only very poor catalytic activity for HER. (In this article X@Y denotes material composites or core–shell particles consisting of two different phases X and Y, whereby X is covering the inner phase Y). Once "electrochemically activated" at the beginning of the electrolysis process, Fe@Fe oxysulfide nanoparticles are generated in situ on the FeS nanosheets. Remarkably, this Fe@FeO<sub>x</sub>S<sub>y</sub> material exhibits a high activity to catalyze both HER and OER simultaneously (Figure 1). At small current densities, the water-splitting output of Fe@FeO<sub>x</sub>S<sub>y</sub> is comparable with that of the benchmark system Pt/C-IrO<sub>2</sub> and at larger current densities it exhibits even better outputs.

Air-stable FeS nanoparticles dispersed on Nafion films, invented by Giraud, Tard and co-workers [35,36], have attracted much attention for HER reactors. FeS nanoparticles are prepared

by solvothermal decomposition of the single-source precursor  $Fe_2S_2(CO)_6$  at 230 °C in the presence of octylamine. Subsequently, vitreous carbon rotating disk electrodes are coated with the catalyst ink, which is prepared by dispersion of the FeS nanoparticles in Nafion. The system achieves HER electrocatalysis of neutral water at room temperature with a mild over potential, and no structural decomposition of the catalyst material or decrease in activity is observed for at least six days. A decrease in overpotential occurs during the first 24 h indicating a rise in hydrogen evolution activity, which is likely due to some modifications of the catalyst surface. Remarkably, with no particular storage care these electrodes were found to be stable for electrochemical studies for more than 6 months.



**Figure 1.** FeS nanosheets grown on iron foam act as pre-catalyst for generation of  $Fe@FeO_xS_y$ , which is a highly active electrocatalyst and splits water simultaneously into molecular hydrogen and oxygen. Reprinted with permission from reference [34].

## 2.2. Iron Disulfide, $FeS_2$

Iron disulfide (pyrite, FeS<sub>2</sub>) is the most prominent phase among the iron sulfides. Due to its unique properties (suitable band gap of E = 0.95 eV and extremely high absorption coefficient  $\alpha > 105 \text{ cm}^{-1}$ ) it has been extensively investigated as a photovoltaic semiconductor [19] and high-performance electrode material [37]. The physiochemical properties of FeS<sub>2</sub> depend critically on its purity and shape; therefore, much effort has been devoted to controlling these factors. Various synthetic procedures for pyrite and applications of pyrite materials have been comprehensively reviewed in the literature [19,28,38,39]. These works show that a major challenge remains the access to single crystalline, high-purity FeS<sub>2</sub> materials with a precise stoichiometric ratio, as Fe<sub>1-x</sub>S impurities are commonly observed in FeS<sub>2</sub> structures. Structural defects coming along with these impurities (S vacancies and interstitial atoms) significantly diminish the photoelectric conversion efficiency of FeS<sub>2</sub>-based devices [28].

# 2.2.1. FeS<sub>2</sub> Wires and Discs

Besides phase purity also the shape of FeS<sub>2</sub> catalysts effects their performance and Ren, Leonard and co-workers demonstrate in a study how strong the HER activity indeed depends on the morphology of the nanomaterials. Following a hot-injection procedure and starting from FeI<sub>2</sub>, elemental sulfur and octadecylamine (which acts as reducing and capping agent), they describe the synthesis of low-dimensional, hyperthin 1D FeS<sub>2</sub> wires, and 2D FeS<sub>2</sub> discs. Notably, both FeS<sub>2</sub> nanomaterials can be obtained following the same synthetic pathway by only changing the ratio of iron precursor and sulfur [40,41] (Figure 2A). The authors determined the exchange current density (a measure of kinetics for HER) of the 1D FeS<sub>2</sub> wires and 2D FeS<sub>2</sub> discs and remarkably, 2D FeS<sub>2</sub> discs clearly exhibit the higher electrocatalytic activity for HER compared to 1D FeS<sub>2</sub> wires and other tested FeS<sub>2</sub> morphologies (Figure 2B). In fact, the electrochemical performance of the investigated 2D FeS<sub>2</sub> discs is comparable to that of platinum in neutral pH conditions [40].



**Figure 2.** (**A**) Schematic representations of 1D and 2D FeS<sub>2</sub> structure formation. a: Fe nanoparticles (formed in the absence of sulfur), b: FeS<sub>2</sub> wires (formed in a 1:6 Fe/S ratio reaction) and c: FeS<sub>2</sub> discs (formed in a 1:24 Fe/S ratio). (**B**) Experimental linear sweep voltammograms at 1 mV/s for different FeS<sub>2</sub> morphologies (solid lines) and the corresponding best-fit single-electron Butler-Volmer equations (dashed lines). Reprinted with permission from reference [40].

### 2.2.2. Mesoporous FeS<sub>2</sub> Nanoparticles

The activity of any heterogeneous catalyst depends on its surface area, which should be as large as possible to reach maximum catalytic effects. In this context, mesoporous materials (materials containing pores with diameters between 2 nm and 50 nm) [42] have reached much attention. Suib and co-workers succeeded in synthesizing mesoporous FeS<sub>2</sub> nanoparticles by sulfidation of iron oxide nanoparticles. In a first reaction step Fe<sub>2</sub>O<sub>3</sub> nanoparticles are produced via an inverse micelle sol-gel method and then heated to 150 °C to yield a mesoporous Fe<sub>2</sub>O<sub>3</sub> structure. Subsequently, this oxide substrate is converted with hydrogen sulfide and elemental sulfur into mesoporous FeS<sub>2</sub> nanoparticles [43]. Theoretical calculations indicate that the exposed (201) facets of this mesoporous pyrite material are key for its excellent HER activity at pH 13 (low over potential of 96 mV (10 mA cm<sup>-2</sup>), Tafel slope of 78 mV dec<sup>-1</sup>).

## 2.2.3. FeS<sub>2</sub>/C Electrode Coating

While several studies found FeS<sub>2</sub> electrocatalysts with significant HER activity, reports on FeS<sub>2</sub> materials for OER or overall water-splitting are very limited. Li, Wang and co-workers published a rare example and demonstrated FeS<sub>2</sub>/C nanoparticles as efficient bifunctional catalysts for overall water-splitting [44]. These FeS<sub>2</sub>/C nanoparticles were prepared by suspending FeS<sub>2</sub> nanoparticles (obtained in a hydrothermal process from FeCl<sub>3</sub>·(H<sub>2</sub>O)<sub>6</sub> and diethy-dithio-carbamate trihydrate) and carbon powder in a water/ethanol mixture containing 5% Nafion. Immobilized on nickel foam,

this FeS<sub>2</sub>/C electrode material can serve as electrocatalyst towards both OER and HER. For OER, excellent activity was obtained (240 mV overpotential at 10 mA cm<sup>-2</sup>), which is lower than that of IrO<sub>2</sub> (~310–320 mV [45,46]). Similarly, and with nearly no activity loss after 1000 cycles, 202 mV at 10 mA cm<sup>-2</sup> overpotential was measured for the hydrogen evolution. An overall water-splitting process with this FeS<sub>2</sub>/C electrode is achieved with 1.72 V at 10 mA cm<sup>-2</sup>.

# 2.2.4. FeS<sub>2</sub> Thin-Films

Chi and Liu developed a combined physical vapor deposition (PVD) and thermal vapor sulfurization technique to produce FeS<sub>2</sub> thin-films. In this process, Fe<sub>3</sub>S<sub>4</sub> thin-films are deposited together with elemental sulfur in a tube furnace system and Fe<sub>3</sub>S<sub>4</sub> is sulfurized and converted into pure FeS<sub>2</sub> thin-films without any crystal phase incorporation [47]. Such traditional vapor deposition methods (chemical vapor deposition (CVD) or PVD) for thin-film substrates generally have limitations in film purity, roughness, and process controllability. In contrast, a new powerful technique, Atomic Layer Deposition (ALD) facilitates access to high-quality and high-purity thin-film substrates [48]. ALD provides a layer-by-layer growth process of substrates enabling precise control of both film composition and thickness. In this way, Guo and Wang recently achieved the first ALD synthesis of FeS<sub>2</sub> thin-films [49]. During the ALD process an iron amidinate precursor [iron-bis(*N*,*N*'-di-*tert*-butylacetamidinato)] is treated with a hydrogen sulfide plasma to generate well-crystallized, pyrite FeS<sub>2</sub> thin-films (Figure 3). The so-produced pyrite (cubic lattice) FeS<sub>2</sub> thin-films can contain traces of marcasite (orthorhombic FeS<sub>2</sub> lattice). Noteworthy, this orthorhombic FeS<sub>2</sub> structure has recently been reported by Hofmann and co-workers to enhance the photoresponse of cubic pyrite FeS<sub>2</sub> films [50].



Figure 3. Pyrite FeS<sub>2</sub> thin-film production by Atomic Layer Deposition (ALD) [49].

## 2.2.5. Resistance of FeS<sub>2</sub> Catalysts to Sulfide Poisoning

Traditional HER electrocatalysts such as platinum materials or other transition metal structures are usually susceptible to poisons such as arsenide or sulfur resulting in a substantial decrease in catalytic activity [51]. To investigate the sulfide-resistance of FeS<sub>2</sub> HER catalysts, the poisoning effect of sulfide on a FeS<sub>2</sub> surface was examined by Chua and Pumera by linear sweep voltammetry [52]. Remarkably, the authors found that FeS<sub>2</sub> electrocatalysts are extremely resistant to sulfide poisoning. The degradation effect of additional sulfide species on the FeS<sub>2</sub> surface is not as severe as observed for platinum-based catalyst materials and results only in a slight increment of overpotential.

#### 2.3. pH Dependence of Iron Sulfide Electrocatalysts

The performance of iron sulfide electrocatalysts does not only depend on the phase composition and morphology of the catalyst material, but also on the pH of the electrolyzed solution. Importantly, the pH range of a solution can change drastically during electrolysis processes. Since this effect is primarily observed directly at the electrodes on the catalyst surface, a locally high concentration of  $H^+$  or  $OH^-$  ions can chemically change the phase composition of a catalyst. As an example, Holt, de Leeuw and co-workers observed a pH-dependent effect with greigite (Fe<sub>3</sub>S<sub>4</sub>) electrocatalysts [53]. During their studies on Fe<sub>3</sub>S<sub>4</sub>-catalyzed HER they observed a large increase of the pH at the electrode surface from pH 7 to pH 12. Fe<sub>3</sub>S<sub>4</sub> is thermodynamically unstable under these basic conditions, being transformed to goethite (iron hydroxide) [24]. Consequently, iron hydroxide covers the electrode surface and must be considered to be a catalytic species for the HER process. Similarly, Giraud, Tard and co-workers monitored the pH dependence of a FeS<sub>2</sub>-catalyzed hydrogen evolution process [54]. The FeS<sub>2</sub> nanoparticles used in this process (synthesized from FeCl<sub>3</sub> and thiourea in a polyol process) show remarkable robustness and stability at very broad pH range (0.3–13). During this electrochemical reaction, at lower pH values a pH-dependent catalytic activity is observed, while above pH 5 the catalytic performance appears to be pH-independent. This observation indicates two entirely different electrochemical reaction mechanisms at low and high pH ranges, respectively.

# 2.4. HER Activity of Iron Sulfides in Comparison

Peron, Giraud, Tard, and co-workers carried out a detailed study to compare the performance of FeS<sub>2</sub>, Fe<sub>3</sub>S<sub>4</sub> and Fe<sub>9</sub>S<sub>10</sub> electrocatalysts for proton exchange membrane (PEM) electrolyzers [35,55]. Electrochemical experiments led to the conclusion that FeS<sub>2</sub> is more active for HER than Fe<sub>3</sub>S<sub>4</sub>, which itself is more active than Fe<sub>9</sub>S<sub>10</sub>. It has been ascertained in other studies on sulfide materials that a high density of sulfur atoms at the material surface and in particular the presence of disulfide S<sub>2</sub><sup>2-</sup> moieties causes a high catalytic activity (see [55] and references cited therein). A higher density of S<sub>2</sub><sup>2-</sup> active sites in FeS<sub>2</sub> compared to Fe<sub>3</sub>S<sub>4</sub> and Fe<sub>9</sub>S<sub>10</sub> becomes obvious from the different sulfur-iron ratios changing in the order 2 (FeS<sub>2</sub>) > 1.33 (Fe<sub>3</sub>S<sub>4</sub>) > 1.11 (Fe<sub>9</sub>S<sub>10</sub>). Furthermore, in pyrite FeS<sub>2</sub> the formal oxidation states of Fe<sup>+II</sup> and S<sup>-I</sup> as well as the structural motive Fe<sub>3</sub>–S–S–Fe<sub>3</sub> (with S<sub>2</sub><sup>2-</sup> dumbbells connecting six iron centers) explain the high natural occurrence of S<sub>2</sub><sup>2-</sup> moieties in FeS<sub>2</sub> [55]. The modest catalytic activity of Fe<sub>3</sub>S<sub>4</sub> is also the subject of a theoretical study by Roldan and de Leeuw. The authors show that Fe<sub>3</sub>S<sub>4</sub>-catalyzed water dissociation is indeed thermodynamically unfavorable and molecular adsorption of water on the Fe<sub>3</sub>S<sub>4</sub> surface is occurring instead [56].

## 3. FeS<sub>2</sub>-TiO<sub>2</sub> Composite Materials

Titanium dioxide is a powerful photocatalyst (high photoactivity, non-toxic, chemically stable and inexpensive); however, due to its rather large band gap (3.2 eV for anatase TiO<sub>2</sub> [57]) it only responds to UV (ultraviolet) light. The decoration of TiO<sub>2</sub> nanomaterials with transition metal photocatalysts is a promising approach to extend the absorption of TiO<sub>2</sub> to the visible and even infrared regions of the electromagnetic spectrum [58]. A potential candidate for this application is FeS<sub>2</sub>, which exhibits a large optical absorption coefficient (>105 cm<sup>-1</sup>) and a narrow band gap of 0.95 eV [59].

Lee and Kang followed this approach and verified that the loading of FeS<sub>2</sub> nanoparticles on TiO<sub>2</sub> enhances the production of hydrogen from methanol/water mixtures [60]. The authors performed cyclic voltammetric and UV-Vis spectrometric measurements and explain the superior photoactivity of  $FeS_2/TiO_2$  hybrid catalysts compared to that of pure TiO<sub>2</sub> with a smaller band gap of the former (2.89 eV for FeS<sub>2</sub>/TiO<sub>2</sub> hybrid catalyst vs. 3.1 eV for TiO<sub>2</sub>). As a result, in FeS<sub>2</sub>/TiO<sub>2</sub> catalysts electrons are more easily promoted from the valence band to the conduction band, resulting in a higher photocatalytic activity. Similarly, Kuo, Li, Lin, Wang and co-workers prepared FeS<sub>2</sub>-TiO<sub>2</sub> nanocrystals, whose absorption range is even extended from the UV-visible to the near-infrared region [61]. As a part of this work, they studied the FeS<sub>2</sub>-TiO<sub>2</sub>-catalyzed HER from a 50% aqueous methanol solution and tested FeS<sub>2</sub>-TiO<sub>2</sub> composites with different FeS<sub>2</sub> loadings. Similar to the work by Lee and Kang, these FeS<sub>2</sub>-TiO<sub>2</sub> heterostructures show remarkably enhanced hydrogen production rates in comparison with pure  $FeS_2$  nanoparticles and pure  $TiO_2$  (see Figure 4A). Figure 4B depicts a schematic action mechanism of the FeS<sub>2</sub>-TiO<sub>2</sub> hybrid catalysts: when the material is irradiated with a broadband light source (covering UV, visible and IR (infrared) irradiation), photo-induced electrons are simultaneously promoted into the conduction bands of TiO<sub>2</sub> and FeS<sub>2</sub> by UV light and visible-NIR light, respectively. Subsequently, electron transfer occurs from the conduction band of FeS<sub>2</sub> to that of TiO<sub>2</sub>, where protons are reduced to hydrogen. In this process it is the heterostructure between  $TiO_2$  nanoparticles and  $FeS_2$ nanocrystals which is key for the enhanced photocatalytic activity since it enables fast electron transfer

from FeS<sub>2</sub> to TiO<sub>2</sub>. Zhang and co-workers developed a comparable FeS<sub>2</sub>/TiO<sub>2</sub> hybrid photocatalyst to use near-infrared light for photoelectrical water-splitting and demonstrate that TiO<sub>2</sub> nanotubes decorated with infrared light responsive FeS<sub>2</sub> exhibit excellent water-splitting performance under illumination of day light (Figure 4C) [62]. Under irradiation by solar light these FeS<sub>2</sub>/TiO<sub>2</sub> nanotube catalysts cause a photocurrent enhancement, which is more than three orders of magnitude higher than that reached with pristine TiO<sub>2</sub> nanotube electrodes under illumination of infrared light.



**Figure 4.** (**A**) Hydrogen evolution rate of FeS<sub>2</sub>, TiO<sub>2</sub> and FeS<sub>2</sub>-TiO<sub>2</sub> heterostructures under a 300 W xenon lamp; (**B**) schematic illustration of the photo-induced charge transfer in FeS<sub>2</sub>-TiO<sub>2</sub> heterostructures under UV, visible, and near-infrared light irradiation. (**A**) and (**B**) reprinted with permission from reference [61]; (**C**) TiO<sub>2</sub> nanotubes doped with FeS<sub>2</sub> exhibit high photoelectrochemical performance in the infrared light region, reprinted with permission from reference [62].

## 4. Iron Sulfide-2D Carbon Hybrid Materials

The deposition of catalyst components on a support material is a common strategy in heterogeneous catalysis, which helps minimize the agglomeration of the catalyst particles and hence increases the total catalytically active surface area. Particularly for the design of electrocatalysts the right choice of support material is of great significance to target catalyst composites with high electron transport capacity, which helps avoid recombination of photogenerated charge carriers to guarantee good electrocatalytic performance. In this context, 2D materials such as graphene, reduced graphene oxide, *h*-boron nitride, monolayer MS<sub>2</sub> (M = Mo, W) or monolayer MoCh (Ch = Se, Te) [63] are commonly used support media for electrocatalysts. For instance, Chen, Lee et al. have shown that platinum nanoparticles supported on reduced graphene oxide exhibit a higher electrocatalytic activity and superior stability than bare platinum nanoparticles [64] and other studies reveal the high efficiency of Mo<sub>2</sub>C/rGO hybrid catalysts for hydrogen evolution [65]. The number of reported iron sulfide-carbon hybrid structures is surprisingly small and this section highlights the value of such materials for HER, with a focus on composites involving the 2D substrates Graphene (G), Graphene Oxide (GO) and reduced Graphene Oxide (rGO) [66–68].

Hu and co-workers report the synthesis of nano-FeS<sub>2</sub>-rGO, a material with FeS<sub>2</sub> nanoparticles embedded in rGO, which has great potential as a high-performance electrocatalyst for HER [69]. Crucial is the high-temperature-assisted synthetic procedure delivering this nano-FeS2-rGO hybrid catalyst, which comprises the in situ synthesis of FeS<sub>2</sub> nanoparticles from microscopic FeS<sub>2</sub> powder in an rGO matrix (Figure 5). In the first reaction step microscopic  $FeS_2$  powder is obtained from bulk iron pyrite with a hammer and then mixed with GO, which has previously been exfoliated from graphite using an improved Hummers' method [68,70]. Subsequently, the obtained micro-FeS<sub>2</sub>-rGO material is exposed to a thermal shock (2470 K, 12 ms) initiating the decomposition of  $FeS_2$  into Fe and S atoms, which diffuse between the rGO layers. As rapid cooling takes place, the Fe and S atoms renucleate around the defects of the rGO sheets to form FeS<sub>2</sub> nanoparticles. The authors envisage that this synthetic strategy could also be applied to target other transition metal multicomponent composites. Nano-FeS<sub>2</sub>-rGO exhibits an excellent HER activity achieving 10 mA cm<sup>-2</sup> at 139 mV overpotential. Importantly, the rGO matrix does not only act as a physical support for the pyrite nanoparticles, but (1) leads to an outstanding chemical stability of the pyrite particles under acidic conditions, (2) increases the electronic conductivity of the catalyst material and (3) hinders the loss of pyrite nanoparticles from the material during the water-splitting process [69].



**Figure 5.** Ultrafast transformation of microscopic  $FeS_2$  mineral into  $FeS_2$  nanoparticles: A thermal shock (2470 K, 12 ms) initiates the decomposition of  $FeS_2$  into Fe and S atoms. Subsequent recrystallization of Fe and S atoms at rGO defects delivers nano-FeS<sub>2</sub>-rGO. Reprinted with permission from reference [69].

A similar FeS<sub>2</sub>-rGO hybrid catalyst has been synthesized by Jiang and co-workers through a hydrothermal synthesis from rGO, Fe(NO<sub>3</sub>)<sub>3</sub> and thioacetamide. This FeS<sub>2</sub>-rGO material consists of hexagonal FeS<sub>2</sub> nanoclusters embedded in a lamellar rGO matrix and its electrochemical performance is superior to that of several iron-based catalysts [71]. The excellent electrical conductivity of rGO ensures a high electron transport capacity and small impedance ( $R_{ct} = 7.94 \Omega$ ). Furthermore, the hybrid catalyst has a smaller Tafel slope (61 mV dec<sup>-1</sup>) than FeP nanosheets (67 mV dec<sup>-1</sup>) [72] or disk-shaped FeS<sub>2</sub> nanoparticles (76 mV dec<sup>-1</sup>) [40].

# 4.2. MoS<sub>2</sub> and Ni Doped FeS<sub>2</sub>/rGO Hybrid Catalysts

Taking the idea of FeS<sub>2</sub>-rGO composites a step further, Wang and co-workers have developed a three-tiered hybrid catalyst FeS<sub>2</sub>@MoS<sub>2</sub>/rGO [73]. This electrocatalyst is fabricated from FeCl<sub>3</sub>, phosphomolybdic acid hydrate and L-cysteine and exhibits excellent HER activities as well as outstanding electrochemical durability. The rGO sheets provide the catalyst with excellent electrical

conductivity and lead to high dispersion of both  $MoS_2$  sheets and  $FeS_2$  particles guaranteeing a high abundance of active sites. Furthermore, rGO triggers strong interactions between the other two components  $MoS_2$  and  $FeS_2$  endowing the catalyst composite with a high mechanical stability. Besides  $MoS_2$  also nickel doping of iron sulfide-rGO composites can have a positive effect on the hydrogen evolution activity. Jiang, Han and co-workers studied different nickel-doped  $FeNi_xS_2$ -rGO electrocatalysts (x = 0.05–0.3) and found that the incorporation of nickel into iron sulfide nanostructures yields electrocatalysts with lower Tafel slope, smaller electrochemical impedance and reduced overpotential [74].

# 4.3. Fe<sub>1-x</sub>S/GO Hybrid Catalyst

Non-stoichiometric  $Fe_{1-x}S$  nanoparticles have been introduced into a sulfur-doped graphene oxide structure (S-GO) to produce nano–micro  $Fe_{1-x}S@S$ -GO. This catalyst material was synthesized by a one-pot solvothermal approach in which aqueous solutions of  $FeCl_3$ , GO, and thioacetamide are heated in an autoclave for 15 h at 200 °C [75]. Hereby, thioacetamide decomposes to the parent sulfide H<sub>2</sub>S, which firstly reacts with FeCl<sub>3</sub> to form  $Fe_{1-x}S$  nanoparticles and secondly converts GO to sulfur-doped GO (S-GO). The so-obtained  $Fe_{1-x}S@S$ -GO hybrid catalyst was used as cathode material in a dye-sensitized solar cell and achieves a high solar-to-electrical conversion efficiency up to 7.23% [75]. The  $Fe_{1-x}S@S$ -GO composite has not yet been tested for HER; however, Wang, Zhou, and co-workers highlight its potential for solar water-splitting.

#### 4.4. Fe<sub>4</sub>S<sub>4</sub>/Graphene Hybrid Catalyst

To finish this section about iron sulfide-carbon hybrid catalysts, one example of a molecular iron sulfide compound immobilized on graphene shall be highlighted. Begum and co-workers report  $Fe_4S_4$  cubane clusters on graphene as high-performance H<sub>2</sub> evolution catalyst in acidic water [76]. [PPh<sub>4</sub>]<sub>2</sub>[ $Fe_4(\mu_3-S)_4(DMET)_4$ ] (**A**) (DMET = *cis*-1,2-dicarbomethoxyethylene dithiolate) was prepared from [PPh<sub>4</sub>]<sub>4</sub>[ $Fe_2S_{12}$ ], dimethylacetylene dicarboxylate, and lithium sulfide. Subsequently, cluster compound **A** was ultrasonicated with functionalized graphene for 10 h under an argon atmosphere to yield **A**@graphene (Figure 6). **A** itself catalyzes the hydrogen evolution from tosylic acid (TON, 400); however, the hybrid catalyst **A**@graphene shows enhanced catalytic activity (TON, 3200) and higher stability in water compared to the naked complex **A**. Likely, the remarkable catalytic activity of **A**@graphene can be explained by continuous in situ reduction of **A** through the graphene support: **A** is oxidized during the proton reduction process and then immediately reduced by the graphene matrix, which acts as an external sacrificial electron donor. In extension to this work, research by Kanatzidis and co-workers shows that redox active  $Fe_4S_4$  clusters incorporated in dye-functionalized gels are capable of producing hydrogen under photochemical conditions [77,78].



**Figure 6.** H-bonding interactions in  $[PPh_4]_2[Fe_4(\mu_3-S)_4(DMET)_4]$ @graphene. Reprinted with permission from reference [76].

#### 5. Metal Doping of Iron Sulfide Nanomaterials

It has been shown in the previous section how the right choice of support material can significantly enhance the activity of iron sulfide electrocatalysts. A different approach to further improve these materials for electrochemical applications is their combination with other metallic elements or metal compounds. Most notably, the incorporation of elemental cobalt or cobalt phases into iron sulfide structures delivers new materials with remarkable HER activity.

## 5.1. Cobalt-Doped FeS<sub>2</sub> Structures

Huang, Luiso, Fedkiw and co-workers synthesized and investigated cobalt-doped iron sulfides  $Fe_xCo_{1-x}S_2$  (x = 0.98-0.32) on a carbon black support (Ketjenblack) and carried out a systematic study on the role of cobalt dopant for the hydrogen evolution activity of these materials [79]. Indeed, electrochemical measurements indicate a significant enhancement in HER activity of the Co-doped FeS<sub>2</sub> structures in comparison to un-doped FeS<sub>2</sub>. Among the catalysts examined,  $Fe_{0.50}Co_{0.50}S_2$  (on HNO<sub>3</sub> oxidized Ketjenblack) exhibited the highest HER activity. The overpotential necessary to reach a current density of 10 mA/cm<sup>2</sup> was 150 mV (229 mV for un-doped FeS<sub>2</sub>) and only decreased by 1 mV after 500 catalytic cycles, indicating a long-term durability of the catalyst material in acidic environment.

A cobalt-doped iron pyrite catalyst with a phosphide surface modification (P/Co-FeS<sub>2</sub>) has been prepared by Chen, Wang and co-workers via solvothermal and chemical vapor deposition approaches [80]. In a hydrothermal process cobalt-doped iron oxide is grown on carbon fiber paper by reaction of iron nitrate and cobalt nitrate. This material is then treated with elemental sulfur at 450 °C to form the cobalt-doped iron pyrite structure Co-FeS<sub>2</sub>. Finally, annealing of this material with sodium hypophosphite results in the formation of P/Co-FeS<sub>2</sub>, a phosphide-modified nanocomposite of Co-FeS<sub>2</sub>. This catalyst material exhibits high HER activity with a low overpotential of 90 mV (at 100 mA cm<sup>-2</sup>) and a high durability in acidic solutions.

Cobalt-doped FeS<sub>2</sub> nanosheets hybridized with carbon nanotubes (CNT)  $Fe_{1-x}Co_xS_2/CNT$  also hold great potential for efficient HER in acidic solutions. Hwang, Chen, Dai, and co-workers report a scalable solvothermal synthesis of such  $Fe_{0.9}Co_{0.1}S_2$ -CNT structures and performed DFT (density functional theory) calculations to elucidate the role of cobalt doping for HER activity [81]. The theoretical investigations showed that the reaction in acidic solution first proceeds via proton adsorption and reduction on the catalyst surface to form hydrogen atoms, which are adsorbed on catalyst edges. Subsequently, H<sub>2</sub> is formed and finally released in a desorption step, which also occurs on the catalyst edges (Figure 7). Notably, sulfur atoms on the edge of the catalyst were found to be the HER active sites. Comparison of the kinetic energy barrier profiles of the  $Fe_{0.9}Co_{0.1}S_2$ -CNT (Figure 7B top) and FeS<sub>2</sub>-CNT (Figure 7B bottom) catalyzed HER reveals a significant difference in the activation barrier of hydrogen atom adsorption leading to transition state 1 (TS1). In case of the FeS<sub>2</sub>-CNT catalyst (1.62 eV). Most likely, this substantial difference (0.39 eV) explains the higher catalytic activity of Fe<sub>1-x</sub>Co<sub>x</sub>S<sub>2</sub>/CNT hybrid catalysts compared to that of FeS<sub>2</sub>-CNT composites.

#### 5.2. FeS<sub>2</sub>-CoS<sub>2</sub> Hybrid Structures

Xi and co-workers demonstrated that even overall water-splitting is possible with cobalt-doped pyrite FeS<sub>2</sub> heterostructures. They developed FeS<sub>2</sub>/CoS<sub>2</sub> interface nanosheets with abundant defects; a bifunctional electrocatalyst, which exhibits superior catalytic properties for both HER and OER in alkaline conditions [82]. A water-splitting device equipped with such FeS<sub>2</sub>/CoS<sub>2</sub> electrodes requires at 25 °C only a voltage of 1.47 V to drive a current density of 10 mA cm<sup>-2</sup>. At elevated temperature (60 °C) the activity of the catalyst is even superior, and a lower voltage of 1.37 V is needed to afford the same current density. Remarkably, the performance of this FeS<sub>2</sub>/CoS<sub>2</sub> catalyst is better than that of most reported non-noble metal overall water-splitting catalysts (see Figure 8). This

example highlights the significance of interface engineering for developing highly efficient and stable water-splitting electrocatalysts.



**Figure 7.** (A) Schematic reaction of hydrogen evolution occurring on a sulfur atom of  $Fe_{0.9}Co_{0.1}S_2$ . (B) Kinetic energy barrier profiles of HER on  $Fe_{0.9}Co_{0.1}S_2$ -CNT (top) and  $FeS_2$ -CNT (bottom) catalysts. Reprinted with permission from reference [81].



**Figure 8.** Comparison of the voltage for  $FeS_2/CoS_2$  interface nanosheets catalyst (red) with other reported non-noble metal overall water-splitting catalysts (at 10 mA cm<sup>-2</sup>). Reprinted with permission from reference [82].

#### 5.3. Cobalt-Doped FeS<sub>2</sub>-CoS<sub>2</sub> Hybrid Structures

Yan, Shi and co-workers merged the two concepts of cobalt doping and the hybridization of  $CoS_2$  and  $FeS_2$  pyrite structures (*vide supra* 5.1 and 5.2) to produce heterostructures of the type Co-FeS<sub>2</sub>/CoS<sub>2</sub>, which can be prepared from cobalt-doped FeS<sub>2</sub> and CoS<sub>2</sub>. Benefitting from a three-dimensional hierarchical nanoarchitecture, which maximizes the exposure of accessible active

sites, these Co-FeS<sub>2</sub>/CoS<sub>2</sub> electrocatalysts exhibited excellent activity for hydrogen and oxygen evolution reactions [83]. The materials' distinctive nanostructure not only endows the catalyst with a large surface area, but also assists the release of hydrogen and oxygen from the electrode surface. The superior oxygen evolution performance of Co-FeS<sub>2</sub>/CoS<sub>2</sub> heterostructures (278 mV overpotential at 10 mA cm<sup>-2</sup> in 1 M KOH) is mainly explained by the presence of the CoS<sub>2</sub> phase and its co-existence as heterostructure with Co-FeS<sub>2</sub>. The overpotential for HER in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution is 103 mV (10 mA cm<sup>-2</sup>) and both overpotential values of OER and HER change only marginally after 1000 catalytic cycles by 5 mV and 3 mV, respectively.

# 5.4. FeS<sub>2</sub>-Doped MoS<sub>2</sub> Nanoflowers

Instead of doping iron disulfide with metals or metal compounds, FeS<sub>2</sub> itself can be employed as dopant and be incorporated in trace levels into a guest crystal lattice. MoS<sub>2</sub> nanomaterials are known to be efficient catalysts for hydrogen evolution [17] and to further enhance the HER efficiency of MoS<sub>2</sub>, Yang and co-workers have prepared MoS<sub>2</sub> nanoflowers (predominately 1T-MoS<sub>2</sub>) and doped them with FeS<sub>2</sub> [84]. Due to a unique morphology and electronic structure this FeS<sub>2</sub>-doped molybdenum sulfide material exhibits exceptional performance as HER electrocatalyst, which is superior to that of pure MoS<sub>2</sub> and pure FeS<sub>2</sub>.

#### 6. Conclusions and Outlook

The development of iron sulfide catalysts for electrochemical hydrogen evolution is a rapidly growing research field, of which selected milestones are summarized in this article. Even though iron sulfide electrocatalysts are still less developed and typically less effective HER catalysts than noble metal systems, numerous examples of well-designed iron sulfide materials exhibit surprising catalytic performance. In particular, the presence of disulfide moieties  $S_2^{2-}$  causes a high catalytic HER activity and thus especially FeS<sub>2</sub> materials are attractive electrocatalysts for hydrogen evolution. Importantly, iron sulfide structures with 2D-carbon nanomaterials exhibit frequently better electrocatalytic performance and chemical robustness than pure iron sulfides. Comparably, doping of iron sulfide phases with other metals or other transition metal compounds delivers highly reactive HER catalysts. All in all, these examples show that the development of new iron sulfide containing electrocatalysts and further improvement of existing materials is urgently needed to exploit the maximum potential of a hitherto overlooked catalyst family for hydrogen evolution.

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# Appendix A. Summary Table of Discussed Iron Sulfide Materials

Synthetic procedures for all discussed iron sulfide nanomaterials and specific characteristic data such as HER overpotential and Tafel slope (the slope of a curve of overpotential in Volts versus the logarithm of current density) are collected in this appendix of the review and provide a summary of their activity and performance as HER electrocatalysts. Importantly, quantitative comparison of the catalysts' HER performance is a challenging task and care should be taken because of differences in the electrode preparation, the amount of employed catalyst material and the electrolyte (pH conditions).

Catalyst	Synthetic Procedure	Electrolyte	Tafel Slope	η HER (X mA cm <sup>-2</sup> )	Ref.					
Iron sulfides										
Fe@FeO <sub>x</sub> S <sub>y</sub>	FeS nanosheets on iron foam	1 M KOH	77	243 (100)	[34]					
FeS NPs on Nafion	Decomposition of Fe <sub>2</sub> S <sub>2</sub> (CO) <sub>6</sub>	pH 7	150	~450 (0.71)	[35]					
$FeS_2$ discs $FeS_2$ wires $FeS_2$ cubes	Hot-injection procedure with $FeI_2$ and $S_8$ in different rations	pH 7	76 91 200	-	[40]					
FeS <sub>2</sub> /C NPs	Hydrothermal process from FeCl <sub>3</sub> ·(H <sub>2</sub> O) <sub>6</sub> and diethyl- dithio-carbamate trihydrate	1 М КОН	98	202 (10)	[44]					
Mesoporous FeS <sub>2</sub>	$\begin{array}{c} Conversion \ of \\ mesoporous \ Fe_2O_3 \ with \\ H_2S \ and \ S_8 \ into \\ mesoporous \ FeS_2 \end{array}$	pH 13	78	96(10)	[43]					
FeS <sub>2</sub> thin-films	Atomic Layer Deposition	-	-	-	[49]					
Cathode in PEM electrolyzer FeS <sub>2</sub> /C Fe <sub>3</sub> S <sub>4</sub> /C Fe <sub>9</sub> S <sub>10</sub> /C	Polyol method from FeCl <sub>3</sub> and thiourea	pH 7	204 224 234	-	[55]					
pH dependence of FeS <sub>2</sub> -catalyzed HER	Polyol method from FeCl <sub>3</sub> and thiourea	pH 0.3 1 2 3 4 5 6 7 9 11 13	85 127 170 290 306 340 287 280 274 329 192	-	[54]					
FeS <sub>2</sub> -TiO <sub>2</sub> materials										
FeS <sub>2</sub> /TiO <sub>2</sub> core–shell composites	Deposition of FeS <sub>2</sub> NPs (solvothermal from FeCl <sub>3</sub> and Na <sub>2</sub> S <sub>2</sub> O <sub>3</sub> ) on nano-sized TiO <sub>2</sub>	MeOH/H <sub>2</sub> O (1:1)	-	-	[60]					
FeS <sub>2</sub> -TiO <sub>2</sub> heterostructures	Injection of S <sub>8</sub> -oleylamine mixture in solution containing TiO <sub>2</sub> -octadecene and Fe-oleic acid precursors	MeOH EtOH H <sub>2</sub> O Na <sub>2</sub> S aq. Na <sub>2</sub> S/Na <sub>2</sub> SO <sub>3</sub> solution	-	-	[61]					
FeS <sub>2</sub> sensitized TiO <sub>2</sub> nanotubes	Deposition of FeS <sub>2</sub> (from FeCl <sub>3</sub> and thiourea) on TiO <sub>2</sub> nanotubes	pH 7.4	-	-	[62]					

**Table A1.** Synthetic procedures and HER performance data (Tafel slope (mV dec<sup>-1</sup>), over potential  $\eta$  (mV)) for discussed iron sulfide materials.

Catalyst	Synthetic Procedure	Electrolyte	Tafel Slope	η HER (X mA cm <sup>-2</sup> )	Ref.				
Iron sulfide-2D carbon hybrid materials									
nano-FeS <sub>2</sub> -rGO	Current-induced high-temperature thermal shock process	0.5 M H <sub>2</sub> SO <sub>4</sub>	66	139 (10)	[69]				
FeS <sub>2</sub> -rGO	Hydrothermal reduction process	0.5 M H <sub>2</sub> SO <sub>4</sub>	61	226 (10)	[71]				
FeS2@MoS2/rGO	Autoclave reaction of FeCl <sub>3</sub> , phosphomolybdic acid hydrate and L-cysteine	0.5 M H <sub>2</sub> SO <sub>4</sub>	38.4	123 (10)	[73]				
FeNi <sub>0.05</sub> S <sub>2</sub> -rGO FeNi <sub>0.10</sub> S <sub>2</sub> -rGO FeNi <sub>0.15</sub> S <sub>2</sub> -rGO FeNi <sub>0.20</sub> S <sub>2</sub> -rGO FeNi <sub>0.25</sub> S <sub>2</sub> -rGO FeNi <sub>0.30</sub> S <sub>2</sub> -rGO	High-temperature hydrothermal method from GO, FeCl <sub>3</sub> , NiCl <sub>2</sub> , and thioacetamide	0.5 M H <sub>2</sub> SO <sub>4</sub>	74.46 71.96 76.96 78.36 82.82 79.19	232 225 198 183 209 260 (10)	[74]				
nano-micro Fe <sub>1-x</sub> S@S-GO	Autoclave reaction of aqueous solutions of FeCl <sub>3</sub> , GO, and thioacetamide	-	-	-	[75]				
[PPh4]2[Fe4(µ3- S)4(DMET)4]@graphene	Ultrasonication of Fe <sub>4</sub> S <sub>4</sub> cubane type cluster with functionalized graphene	<i>p</i> -toluene sulfonic acid	-	-	[76]				
Metal-doped iron sulfide materials									
Fe <sub>0.50</sub> Co <sub>0.50</sub> S <sub>2</sub> on Ketjenblack oxidized with 10 N nitric acid	Chemical sulfurization of metal precursors using $H_2S$	0.5 M H <sub>2</sub> SO <sub>4</sub>	52	150 (10)	[79]				
P/Co-FeS <sub>2</sub>	Hydrothermal process of iron nitrate and cobalt nitrate followed by CVD treatment with S <sub>8</sub>	0.5 M H <sub>2</sub> SO <sub>4</sub>	41.5	90 (100)	[80]				
Fe <sub>0.9</sub> Co <sub>0.1</sub> S <sub>2</sub> /CNT	Solvothermal approach	$0.5 \text{ M H}_2\text{SO}_4$	46	120 (20)	[81]				
FeS <sub>2</sub> /CoS <sub>2</sub> interface nanosheets	Annealing of CoFe <sub>2</sub> O <sub>4</sub> NPs with sulfur	1 M KOH	44	78.2 (10)	[82]				
Co-FeS <sub>2</sub> /CoS	Hydrothermal process from FeSO <sub>4</sub> , Co(NO <sub>3</sub> ) <sub>2</sub> , thiourea, and sulfur	0.5 M H <sub>2</sub> SO <sub>4</sub>	56	103 (10)	[83]				
FeS <sub>2</sub> -doped MoS <sub>2</sub>	Autoclave process of $Na_2Mo_4.2H_2O$ , thiourea, and $Fe_3O_4$ microspheres	0.5 M H <sub>2</sub> SO <sub>4</sub>	82	136 (10)	[84]				

# Table A1. Cont.

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