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Efficient and Stable Solution-Processed Organic Light Emitting Transistors using a High-k Dielectric

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ABSTRACT:

We report the development of highly efficient and stable solution-processed organic light emitting transistors (OLETs) that combine a polymer heterostructure with the transparent high- k dielectric poly(vinylidene fluoride_{0.62}-trifluoroethylene_{0.31}-chlorotrifluoroethylene_{0.7}) (P(VDF-TrFE-CTFE)). The polymer heterostructure comprises of the poly[4-(4,4-dihexadecyl-4H-cyclopenta[1,2-b:5,4-b']dithiophen-2-yl)-alt-[1,2,5]thiadiazolo[3,4-c]pyridine] (PCDTPT) and Super Yellow as charge transporting and light emitting layers, respectively. Device characterization shows that the use of P(VDF-TrFE-CTFE) leads to larger channel currents (≈ 2 mA) and lower operating voltages (≈ 35 V) than for previously reported polymer based OLETs. Furthermore, the combined transparency of the dielectric and gate electrode, results in efficient bottom emission with external quantum efficiency of ≈ 0.88 % at a luminance $L \geq 2000$ cd m⁻². Importantly, the resulting OLETs exhibit excellent shelf life and operational stability. The present work represents a significant step forward in the pursuit of all-solution-processed OLET technology for lighting and display applications.

KEYWORDS: organic light-emitting transistors, high- k dielectric, polyvinylidene fluoride copolymer

1. INTRODUCTION

Organic light-emitting diodes (OLEDs) are now recognized as an important technology for application in ultrathin flexible displays on plastic or metal foil-base substrates. OLED based smart phones and large-size TVs have already entered the market, with the majority of these products utilizing rigid glass substrates. Despite the tremendous progress however, improvements in power efficiency, lifetime and cost are still needed in order to realize the full potential of the technology.¹⁻³ Such requirements are even more pertinent for lighting and have driven development of solution-based fabrication processes that will also impact on the display sector.⁴⁻⁵ Additionally, the control circuitry that drives the OLED pixels needs to transition from the rigid and brittle inorganic semiconductor technologies to soluble semiconductors deposited at low temperature which will enable integration with flexible plastic substrates. However, the low carrier mobility of organic semiconductors is not yet sufficient for such current-driven display backplanes⁶⁻⁸ and oxide semiconductors look more

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4 promising in the first instance, albeit with the added challenge to deposit these materials at
5 appropriate temperatures.⁹
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9 An alternative to an OLED with associated drive circuitry is to combine the functionality
10 of switching and light-emission in a single device, an organic light-emitting transistor
11 (OLET).¹⁰⁻¹² The latter combines both light-emitting and switching functions in a single
12 device. Because of this multi-functionality, OLETs provide great promise for simple
13 processing of advanced pixel architectures.¹³⁻¹⁶
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21 Despite the significant progress that has already been made in the performance of OLETs,
22 there are still numerous technical obstacles that need to be overcome, including the
23 simultaneous achievement of high luminance, high external quantum efficiency (EQE) and
24 low voltage operation.¹⁷⁻²¹ A key strategy to increase the driving current capabilities while
25 reducing the operating voltage is to introduce high-*k* gate dielectrics. Previously, OLETs with
26 low operation voltages (~10 V) have been reported employing Al₂O₃/PMMA and
27 Al₂O₃/ZrO_x bilayer gate dielectrics, however, the Al₂O₃ thin films were deposited by atomic
28 layer deposition (ALD), which has a slow deposition rate and the Al₂O₃/ZrO_x bilayer required
29 ozone exposure and a photochemical curing step.²²⁻²³ In addition, OLETs with a thermally
30 deposited trilayer organic stack and a poly(vinylidene fluoride-trifluoroethylene-
31 chlorotrifluoroethylene) P(VDF-TrFE-CFE) dielectric have been reported to exhibit higher
32 EQE with reduced operating voltage, when compared to those with a low-*k* poly(methyl
33 methacrylate) (PMMA) dielectric layer.²⁴⁻²⁶ However, the P(VDF-TrFE-CTFE) polymer has
34 not been used in OLETs before despite its intriguing properties, including high-*k*, outstanding
35 electromechanical response, and low dielectric leakage.²⁷⁻³¹
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55 Here, we demonstrate efficient and stable, multi-layer solution-processed OLETs in a non-
56 coplanar electrode structure comprising the high-*k* dielectric P(VDF-TrFE-CTFE), a charge-
57 transporting layer of poly[4-(4,4-dihexadecyl-4H-cyclopenta[1,2-b:5,4-b']dithiophen-2-yl)-
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4 alt-[1,2,5]thiadiazolo[3,4-c]pyridine] (PCDTPT), and a light-emissive layer of the Super
5 Yellow polymer. The P(VDF-TrFE-CTFE) dielectric exhibited excellent endurance and the
6
7 Yellow polymer. The P(VDF-TrFE-CTFE) dielectric exhibited excellent endurance and the
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9 devices consequently showed high operational stability and device lifetime. Carefully
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11 engineered bottom emission OLETs exhibited Luminance of >2000 cd/m² and an EQE value
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13 of ~ 0.88 %.

17 18 **2. EXPERIMENTAL SECTION**

20 21 **2.1. Transistor Fabrication and Characterization.**

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23 Super Yellow (PDY-132), poly[4-(4,4-dihexadecyl-4H-cyclopenta[1,2-b:5,4-
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25 b']dithiophen-2-yl)-alt-[1,2,5] thiadiazolo[3,4-c]pyridine] (PCDTPT), and
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27 poly(vinylidene fluoride_{0.62}-trifluoroethylene_{0.31}-chlorotrifluoroethylene_{0.7}) (P(VDF-TrFE-
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29 CTFE)) polymers (see Figure 1 (a) for chemical structures) were purchased from Merck, 1-
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31 Material and Piezotech Arkema. Super Yellow and PCDTPT were dissolved in toluene and
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33 chlorobenzene at 7 mg/mL and 5 mg/mL concentration, respectively. A 2-butanone solution
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35 of P(VDF-TrFE-CTFE) was prepared at a concentration of 30 mg/mL and vigorously stirred
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37 overnight. The P(VDF-TrFE-CTFE) solution was spin-coated (1500 rpm, 60 s, 300 nm-
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39 thickness layer) on ITO-glass substrates pre-patterned to form the gate electrode and annealed
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41 at 100 °C for 20 min. PCDTPT was spin-coated (2000 rpm, 40 s, 50-nm thickness layer) on
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43 top of the P(VDF-TrFE-CTFE) layer, followed by annealing at 150 °C for 30 min. MoO₃ (~ 5
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45 nm) and Au (~ 20 nm) were then thermally deposited on top of the PCDTPT layer using a
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47 shadow mask to define the source electrode. Super Yellow was next spin-coated (2500 rpm,
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49 30s followed by 3000 rpm, 10 s, 100-nm thickness layer) and again annealed at 150 °C for 30
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51 min. The device structure was completed by shadow-masked vacuum deposition of the drain
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4 electrode, comprising Cs_2CO_3 (~ 5 nm) capped with Al (~ 20 nm). The transistor channel
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6 length and width were 75 μm and 14 mm, respectively.
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10 11 **2.2. Measurements.**

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13 The electrical characterisation of the devices was performed within a nitrogen-filled glove
14 box using an Agilent B2902A semiconductor parameter analyser. The capacitance-frequency
15 (C-f) response of the P(VDF-TrFE-CTFE) layer was measured using a Fluxim Paios
16 measurement system. Electroluminescence (EL) and photoluminescence (PL) spectra were
17 measured using a UV-vis spectrometer (Ocean-Optics USB4000-XR). Charge carrier
18 mobility was calculated from the transistors' transfer characteristics in the saturation regime.
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20 The luminance of the devices was determined from the photocurrent generated by a
21 calibrated photodiode, referenced to a standard luminance meter (Minolta LS-100), taking
22 into account the relative emission area. EQE was obtained from the luminance, source-drain
23 current (I_D), and emission spectra of the devices, assuming Lambertian emission. Tapping
24 mode atomic force microscope (AFM) images were recorded using a Digital Instruments
25 Dimension 3100 scanning probe microscope (SPM) with NanoScope IVa SPM control
26 station. Optical transmission spectra, over the wavelength range 400-850 nm, were measured
27 using a Shimadzu UV-1800 spectrophotometer.
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48 **3. RESULTS AND DISCUSSION**

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50 Figure 1 shows a schematic of the device structure, the energy level alignments and a
51 simplified operation model. Super Yellow was chosen as the emissive layer with PCDTPT as
52 hole-transporting layer and P(VDF-TrFE-CTFE) as the high- k gate dielectric. Improved
53 charge injection into the organic layers was facilitated by the non-coplanar device structure
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4 with asymmetric source and drain (S-D) contacts (Figure 1 (b)). The device operation can be
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6 considered in terms of a combination OLED-transistor structure (Figure 1(c)). The highest
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8 occupied (HOMO) and lowest unoccupied (LUMO) molecular orbital levels of each material
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10 (Figure 1 (b)) are taken from the literature.³²⁻³⁴ As described in Figure 1(c), holes are injected
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12 from the MoO₃/Au source contact into the HOMO energy level of the PCDTPT layer and
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14 subsequently transported across the channel by the PCDTPT layer. Electrons on the other
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16 hand are injected from the Cs₂CO₃/Al drain contact into the LUMO of the Super Yellow
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18 layer where they combine with holes to form excitons that decay radiatively generating
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20 optical photons. Here we note that the presence of MoO₃ layer at the interface between Au
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22 and PCDTPT promotes contact doping that increases the electrical conductivity of the bulk
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24 film near the contact, thereby reducing contact resistance.³⁵⁻³⁶ In addition, the Cs₂CO₃
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26 electron injection interlayer can play an important role in lowering the work function of Al
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28 (*ca.* -2.1 eV for Cs₂CO₃/Al *vs* -4.2 eV for Al) by forming an Al-O-Cs complex.³⁷
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36 **Figure 1**

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41 We first investigated the gate dielectric using a metal (ITO) - insulator (P(VDF-TrFE-
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43 CTFE)) - metal (Al) (MIM) structure, whose capacitance *vs* frequency (C-f) response over
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45 the range 10 to 10 MHz is shown in Figure 2(a). The capacitance decreased with increasing
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47 frequency as the polarization response time is limited by dipole alignment in the high-*k*
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49 dielectric.³⁸ The P(VDF-TrFE-CTFE) film capacitance (C) measured at 10 kHz is 86.3
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51 nF/cm², from which a *k* value of 29.3 was calculated using the parallel plate capacitor
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53 equation [$C = (k\epsilon_0 A) / d$], with ϵ_0 , A, and d, being the vacuum permittivity, area of overlap of
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55 the electrodes, and film thickness, respectively. These C and *k* values are comparable to those
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57 obtained from a Si/SiO₂/P(VDF-TrFE-CTFE)/Al capacitor (Figure S1). The relatively high
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breakdown voltage, demonstrating dielectric endurance up to 2.5 MV/cm (≈ 75 V), is shown in the current density-electric field (J-E) graph (Figure 2(b)) and a leakage current density of less than 10^{-5} A/cm² was observed. We note that the P(VDF-TrFE-CTFE) dielectric layer did not reach its breakdown voltage in the studied bias range.

For full transistor structures, as shown in Figure 2(c), typical output characteristics (drain voltage vs drain current, V_D - I_D) with distinct linear and saturation regimes were observed. Large I_D values (> 1 mA) were observed when V_G was biased at above -20 V (see Figure 2(d)). Based on height-mode AFM images of P(VDF-TrFE-CTFE) and PCDTPT layers on glass and ITO-glass substrates (see Figure S2) both polymers form smooth layers; a root mean square roughness 0.92 nm, 1.74 nm 1.24 nm and 1.24 nm was found for P(VDF-TrFE-CTFE) on glass, PCDTPT on glass, PCDTPT on P(VDF-TrFE-CTFE) coated glass and PCDTPT on P(VDF-TrFE-CTFE) coated ITO-glass, respectively. The surface morphology is uniform, facilitating lateral charge transport in the channel region and reducing charge trapping at the surface of the P(VDF-TrFE-CTFE) and PCDTPT layers. In addition, as shown in Figure 2(d), we observe the expected hysteresis between the forward and reverse scan transfer curves, associated with ferroelectric polarization switching in the high- k dielectric. The hole mobilities (μ_h) extracted from the transfer curves were 0.55 and 0.39 cm² V⁻¹ s⁻¹ and threshold voltages (V_{TH}) were -0.83 and 4.36 V for the forward and reverse scans, respectively. Both μ_h and V_{TH} were calculated in the saturation region using the gradual channel approximation model. The high capacitance of the P(VDF-TrFE-CTFE) layer allows low voltage operation due to the significant accumulation of holes at the PCDTPT/P(VDF-TrFE-CTFE) interface.

Figure 2

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4 The variation in luminance and EQE at $V_D = -35$ V measured from the bottom and top of
5 the device (c.f. Figure S3) are shown as a function of V_G in Figures 3(a) and (b), respectively.
6
7 The bottom emission luminance ($L = 2368.9$ cd m⁻²) at $V_G = -35$ V was ~ 13.6 times higher
8 than the top emission ($L = 174.5$ cd m⁻²). The higher luminance for the bottom emission was
9 observed across a wide range of V_D and V_G values (see optical output characteristics in
10 Figure S4). In addition, the EQE value for the bottom emission (0.88 %) was approximately
11 one order of magnitude higher than for the top emission (0.066 %). Both EQE values saturate
12 for $|V_G| \geq 20$ V (Figure 3(b)), consistent with balanced hole and electron injection and
13 minimal exciton quenching at high electric fields. The EQE is presented as a function of
14 luminance in Figure 3(c), with good bottom emission performance achieved all the way up to
15 2500 cd/m². Also shown as an inset to Figure 3(c) is an image of Super Yellow emission
16 emanating from such a device biased at $V_D = V_G = -35$ V. The observed hysteresis in the
17 luminance / EQE vs V_G curves between forward and reverse scans can be attributed to the
18 ferroelectric properties of the P(VDF-TrFE-CTFE) layer leading to differing amounts of
19 charge recombination. The electroluminescence (EL) spectrum for such an OLET is shown in
20 Figure 3(d), closely overlapping but narrowed relative to the corresponding PL spectrum
21 (Figure S5). The EL peaks at $\lambda = 550$ nm, with the PL extending further into the red. Figure
22 3(d) also shows transmission spectra measured through an ITO-glass substrate coated with
23 P(VDF-TrFE-CTFE) and PCDTPT (T_{Bottom}) and through a Spectrosil substrate coated with
24 Cs₂CO₃ and Al (T_{Top}), with layer thicknesses as used in the device. These samples therefore
25 represent the light path from the Super Yellow emission layer in both directions through a
26 half-stack but without accounting for microcavity resonance effects in the full stack structure.
27 We observe that $T_{\text{Bottom}}(565 \text{ nm}) \approx 91.03$ % is significantly higher than $T_{\text{Top}}(565 \text{ nm}) \approx 19.3$
28 % . The even higher EQE ratio between bottom emission and top emission is a consequence
29 of the latter effects.
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Figure 3

Device operational stability was monitored over a 250 s time window for ITO-glass/P(VDF-TrFE-CTFE)/PCDTPT/Super Yellow/Cs₂CO₃/Al OLETs, with I_D and bottom emission luminance values continuously recorded under constant bias ($V_D = V_G = -35$ V) and in ambient conditions. As shown in Figure 4(a), I_D decreased by a relatively modest amount (~ 11%) over this period, indicating relatively good electrical stability for an un-encapsulated structure. More importantly, Figure 4(b) demonstrates that only a marginal reduction by ~ 2.1 % in the device luminance occurred over the same period. Typically, both extrinsic (e.g. oxygen, water, and temperature) and intrinsic (e.g. excited state and charge carrier interactions) factors are complicit in degradation, with reductions in charge carrier mobility and charge injection efficiency previously reported to result.³⁹ Our devices showed more stable charge carrier mobility and charge injection, leading to only a marginal reduction in device luminance over the period observed. A further study will be needed to explore the longer-term device lifetime and associated degradation mechanisms.

Figure 4

4. CONCLUSIONS

In conclusion, we have demonstrated efficient and stable solution-processed bottom-emitting organic light emitting transistors using a high- k dielectric in combination with an asymmetric electrode device architecture. The P(VDF-TrFE-CTFE) dielectric layer was shown to play a critical role in reducing the operating voltage while the asymmetric electrode

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4 architecture improved the hole and electron injection into the light emissive Super Yellow
5 layer, leading to efficient radiative recombination. Importantly, the OLETs showed high
6 maximum channel currents (≈ 2 mA), with a hole mobility μ_h value in the range 0.39 - 0.55
7 $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$. The high μ_h together with the obtained EQE of ≈ 0.88 % allows the luminance for
8 bottom emission devices to reach values in excess of 2000 cd/m^2 . Noticeably, the OLETs
9 exhibited promising operating stability with peak EQE and luminance measured at maximum
10 I_D . It is evident that introduction of a high-k dielectric can positively influence both the
11 operational stability and lifetime of the OLETs. Our results thus present a significant
12 opportunity to advance OLET performance towards efficient, bright and stable operation in
13 numerous applications, including solid-state lighting, optical communications, smart pixels
14 and integrated optoelectronic systems.

31 ■ ASSOCIATED CONTENT

32 Supporting Information

33 This material is available free of charge via the Internet at <http://pubs.acs.org>.

34 ■ AUTHOR INFORMATION

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38 Notes

39 The authors declare no competing financial interest.

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21
22

23 ■ REFERENCES

- 24
25
26 (1) Kim, K. H.; Kim, J. J., Origin and control of orientation of phosphorescent and
27
28
29 tadf dyes for high-efficiency OLEDs. *Adv. Mater.* **2018**, *30*, e1705600.
30
31
32 (2) Liu, Y.; Li, C.; Ren, Z.; Yan, S.; Bryce, M. R., All-organic thermally activated
33
34
35
36 delayed fluorescence materials for organic light-emitting diodes. *Nat. Rev.*
37
38
39 *Mater.* **2018**, *3*, 18020.
40
41
42 (3) Tang, C. W.; VanSlyke, S. A., Organic electroluminescent diodes. *Appl. Phys.*
43
44
45
46 *Lett.* **1987**, *51*, 913-915.
47
48
49 (4) Maasoumi, F.; Jansen-van Vuuren, R. D.; Shaw, P. E.; Puttock, E. V.; Nagiri, R.
50
51
52
53 C. R.; McEwan, J. A.; Bown, M.; O’Connell, J. L.; Dunn, C. J.; Burn, P. L.;
54
55
56
57 Namdas, E. B., An external quantum efficiency of >20% from solution-
58
59
60

- 1
2
3
4 processed poly(dendrimer) organic light-emitting diodes. *npj Flexible Electron.*
5
6
7
8 **2018**, *2*, 27.
9
10
11 (5) Perumal, A.; Faber, H.; Yaacobi-Gross, N.; Pattanasattayavong, P.; Burgess,
12
13 C.; Jha, S.; McLachlan, M. A.; Stavrinou, P. N.; Anthopoulos, T. D.; Bradley, D.
14
15 D. C., High-efficiency, solution-processed, multilayer phosphorescent organic
16
17 light-emitting diodes with a copper thiocyanate hole-injection/hole-transport
18
19 layer. *Adv. Mater.* **2015**, *27*, 93-100.
20
21
22
23
24
25
26
27
28 (6) Cui, N.; Ren, H.; Tang, Q.; Zhao, X.; Tong, Y.; Hu, W.; Liu, Y., Fully transparent
29
30 conformal organic thin-film transistor array and its application as led front
31
32 driving. *Nanoscale* **2018**, *10*, 3613-3620.
33
34
35
36
37
38
39 (7) Liu, J.; Qin, Z.; Gao, H.; Dong, H.; Zhu, J.; Hu, W., Vertical organic field-effect
40
41 transistors. *Adv. Funct. Mater.* **2019**, *29*, 1808453.
42
43
44
45
46 (8) Choi, M.; Park, Y. J.; Sharma, B. K.; Bae, S.-R.; Kim, S. Y.; Ahn, J.-H., Flexible
47
48 active-matrix organic light-emitting diode display enabled by mos2 thin-film
49
50 transistor. **2018**, *4*, eaas8721.
51
52
53
54
55
56
57
58
59
60

- 1
2
3
4
5 (9) Adamopoulos, G.; Thomas, S.; Wobkenberg, P. H.; Bradley, D. D.; McLachlan,
6
7
8 M. A.; Anthopoulos, T. D., High-mobility low-voltage ZnO and Li-doped zno
9
10
11 transistors based on ZrO₂ high-k dielectric grown by spray pyrolysis in ambient
12
13
14
15 air. *Adv. Mater.* **2011**, *23*, 1894-1898.
16
17
18
19 (10) Zhang, C.; Chen, P.; Hu, W., Organic light-emitting transistors: Materials,
20
21
22 device configurations, and operations. *Small* **2016**, *12*, 1252-1294.
23
24
25
26 (11) Wakayama, Y.; Hayakawa, R.; Seo, H. S., Recent progress in photoactive
27
28
29 organic field-effect transistors. *Sci. Technol. Adv. Mater.* **2014**, *15*, 024202.
30
31
32
33 (12) Liu, C. F.; Liu, X.; Lai, W. Y.; Huang, W., Organic light-emitting field-effect
34
35
36 transistors: Device geometries and fabrication techniques. *Adv. Mater.* **2018**,
37
38
39 *30*, e1802466.
40
41
42
43 (13) Bisri, S. Z.; Takenobu, T.; Sawabe, K.; Tsuda, S.; Yomogida, Y.; Yamao, T.;
44
45
46 Hotta, S.; Adachi, C.; Iwasa, Y., p-i-n homojunction in organic light-emitting
47
48
49 transistors. *Adv. Mater.* **2011**, *23*, 2753-2758.
50
51
52
53 (14) Capelli, R.; Dinelli, F.; Gazzano, M.; D'Alpaos, R.; Stefani, A.; Generali, G.;
54
55
56
57 Riva, M.; Montecchi, M.; Giglia, A.; Pasquali, L., Interface functionalities in
58
59
60

- 1
2
3
4 multilayer stack organic light emitting transistors (OLETs). *Adv. Funct. Mater.*
5
6
7
8 **2014**, *24*, 5603-5613.
9
10
11 (15) Muccini, M., A bright future for organic field-effect transistors. *Nat. Mater.* **2006**,
12
13
14
15 *5*, 605-613.
16
17
18 (16) McCarthy, M. A.; Liu, B.; Donoghue, E. P.; Kravchenko, I.; Kim, D. Y.; Rinzler,
19
20
21
22 A. G., Low-voltage, low-power, organic light-emitting transistors for active
23
24
25
26
27
28
29 (17) Ullah, M.; Armin, A.; Tandy, K.; Yambem, S. D.; Burn, P. L.; Meredith, P.;
30
31
32
33
34
35
36
37
38
39
40 (18) Ullah, M.; Tandy, K.; Yambem, S. D.; Aljada, M.; Burn, P. L.; Meredith, P.;
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
- Namdas, E. B., Defining the light emitting area for displays in the unipolar regime of highly efficient light emitting transistors. *Sci. Rep.* **2015**, *5*, 8818.
- Namdas, E. B., Simultaneous enhancement of brightness, efficiency, and switching in rgb organic light emitting transistors. *Adv. Mater.* **2013**, *25*, 6213-6218.

- 1
2
3
4
5 (19) Namdas, E. B.; Hsu, B. B. Y.; Liu, Z.; Lo, S. C.; Burn, P. L.; Samuel, I. D. W.,
6
7
8 Phosphorescent light-emitting transistors: Harvesting triplet excitons. *Adv.*
9
10
11 *Mater.* **2009**, *21*, 4957-4961.
12
13
14
15 (20) Zaumseil, J.; Friend, R. H.; Sirringhaus, H., Spatial control of the recombination
16
17
18 zone in an ambipolar light-emitting organic transistor. *Nat. Mater.* **2005**, *5*, 69-
19
20
21
22 74.
23
24
25
26 (21) Muhieddine, K.; Ullah, M.; Pal, B. N.; Burn, P.; Namdas, E. B., All solution-
27
28
29 processed, hybrid light emitting field-effect transistors. *Adv. Mater.* **2014**, *26*,
30
31
32 6410-6415.
33
34
35
36 (22) Ullah, M.; Wawrzinek, R.; Maasoumi, F.; Lo, S.-C.; Namdas, E. B.,
37
38
39 Semitransparent and low-voltage operating organic light-emitting field-effect
40
41
42 transistors processed at low temperatures. *Adv. Opt. Mater.* **2016**, *4*, 1022-
43
44
45
46 1026.
47
48
49
50 (23) Chaudhry, M. U.; Tetzner, K.; Lin, Y. H.; Nam, S.; Pearson, C.; Groves, C.;
51
52
53 Petty, M. C.; Anthopoulos, T. D.; Bradley, D. D. C., Low-voltage solution-
54
55
56
57
58
59
60

- 1
2
3
4 processed hybrid light-emitting transistors. *ACS Appl. Mater. Interfaces* **2018**,
5
6
7
8 *10*, 18445-18449.
9
- 10
11 (24) Soldano, C.; D'Alpaos, R.; Generali, G., Highly efficient red organic light-
12
13
14
15 emitting transistors (OLETs) on high-k dielectric. *ACS Photon.* **2017**, *4*, 800-
16
17
18 805.
19
- 20
21 (25) Pattanasattayavong, P.; Yaacobi-Gross, N.; Zhao, K.; Ndjawa, G. O.; Li, J.;
22
23
24
25 Yan, F.; O'Regan, B. C.; Amassian, A.; Anthopoulos, T. D., Hole-transporting
26
27
28
29 transistors and circuits based on the transparent inorganic semiconductor
30
31
32 copper(i) thiocyanate (CuSCN) processed from solution at room temperature.
33
34
35
36 *Adv. Mater.* **2013**, *25*, 1504-1509.
37
38
- 39 (26) Li, J.; Sun, Z.; Yan, F., Solution processable low-voltage organic thin film
40
41
42
43 transistors with high-k relaxor ferroelectric polymer as gate insulator. *Adv.*
44
45
46
47 *Mater.* **2012**, *24*, 88-93.
48
- 49 (27) Cho, Y.; Ahn, D.; Park, J. B.; Pak, S.; Lee, S.; Jun, B. O.; Hong, J.; Lee, S. Y.;
50
51
52
53 Jang, J. E.; Hong, J.; Morris, S. M.; Sohn, J. I.; Cha, S. N.; Kim, J. M.,
54
55
56
57 Enhanced ferroelectric property of P(VDF-TrFE-CTFE) film using room-
58
59
60

- 1
2
3
4 temperature crystallization for high-performance ferroelectric device
5
6
7
8 applications. *Adv. Electron. Mater.* **2016**, *2*, 1600225.
9
10
11 (28) Hwang, S. K.; Bae, I.; Cho, S. M.; Kim, R. H.; Jung, H. J.; Park, C., High
12
13 performance multi-level non-volatile polymer memory with solution-blended
14
15 ferroelectric polymer/high-k insulators for low voltage operation. *Adv. Funct.*
16
17
18
19
20
21
22
23 *Mater.* **2013**, *23*, 5484-5493.
24
25 (29) Kim, J.; Lee, J. H.; Ryu, H.; Lee, J.-H.; Khan, U.; Kim, H.; Kwak, S. S.; Kim, S.-
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60
W., High-performance piezoelectric, pyroelectric, and triboelectric
nanogenerators based on P(VDF-TrFE) with controlled crystallinity and dipole
alignment. *Adv. Funct. Mater.* **2017**, *27*, 1700702.
Enhanced performance of
ferroelectric-based all organic capacitors and transistors through choice of
solvent. *Appl. Phys. Lett.* **2014**, *104*, 233301.
Non-volatile polymer
electroluminescence programmable with ferroelectric field-induced charge
injection gate. *Adv. Funct. Mater.* **2016**, *26*, 5391-5399.

- 1
2
3
4
5 (32) Tseng, H. R.; Ying, L.; Hsu, B. B.; Perez, L. A.; Takacs, C. J.; Bazan, G. C.;
6
7
8 Heeger, A. J., High mobility field effect transistors based on macroscopically
9
10
11 oriented regioregular copolymers. *Nano Lett.* **2012**, *12*, 6353-6357.
12
13
14
15 (33) Tseng, H. R.; Phan, H.; Luo, C.; Wang, M.; Perez, L. A.; Patel, S. N.; Ying, L.;
16
17
18 Kramer, E. J.; Nguyen, T. Q.; Bazan, G. C.; Heeger, A. J., High-mobility field-
19
20
21 effect transistors fabricated with macroscopic aligned semiconducting
22
23
24
25 polymers. *Adv. Mater.* **2014**, *26*, 2993-2998.
26
27
28
29 (34) Ullah, M.; Tandy, K.; Yambem, S. D.; Muhieddine, K.; Ong, W. J.; Shi, Z.; Burn,
30
31
32 P. L.; Meredith, P.; Li, J.; Namdas, E. B., Efficient and bright polymer light
33
34
35 emitting field effect transistors. *Org. Electron.* **2015**, *17*, 371-376.
36
37
38
39 (35) Choi, S.; Fuentes-Hernandez, C.; Wang, C. Y.; Khan, T. M.; Larrain, F. A.;
40
41
42 Zhang, Y.; Barlow, S.; Marder, S. R.; Kippelen, B., A study on reducing contact
43
44
45 resistance in solution-processed organic field-effect transistors. *ACS Appl.*
46
47
48
49 *Mater. Interfaces* **2016**, *8*, 24744-24752.
50
51
52
53
54
55
56
57
58
59
60

- 1
2
3
4
5 (36) Ablat, A.; Kyndiah, A.; Houin, G.; Alic, T. Y.; Hirsch, L.; Abbas, M., Role of
6
7
8 oxide/metal bilayer electrodes in solution processed organic field effect
9
10
11 transistors. *Sci. Rep.* **2019**, *9*, 6685.
12
13
14
15 (37) Huang, J.; Xu, Z.; Yang, Y., Low-work-function surface formed by solution-
16
17
18 processed and thermally deposited nanoscale layers of cesium carbonate. *Adv.*
19
20
21 *Funct. Mater.* **2007**, *17*, 1966-1973.
22
23
24
25 (38) Baeg, K. J.; Khim, D.; Kim, J.; Han, H.; Jung, S. W.; Kim, T. W.; Kang, M.;
26
27
28 Facchetti, A.; Hong, S. K.; Kim, D. Y.; Noh, Y. Y., Controlled charge transport
29
30
31 by polymer blend dielectrics in top-gate organic field-effect transistors for low-
32
33
34 voltage-operating complementary circuits. *ACS Appl. Mater. Interfaces* **2012**, *4*,
35
36
37
38 6176-6184.
39
40
41
42 (39) Niu, Q.; Rohloff, R.; Wetzelaer, G. A. H.; Blom, P. W. M.; Craciun, N. I., Hole
43
44
45 trap formation in polymer light-emitting diodes under current stress. *Nat. Mater.*
46
47
48
49 **2018**, *17*, 557-562.
50
51
52
53
54
55
56
57
58
59
60

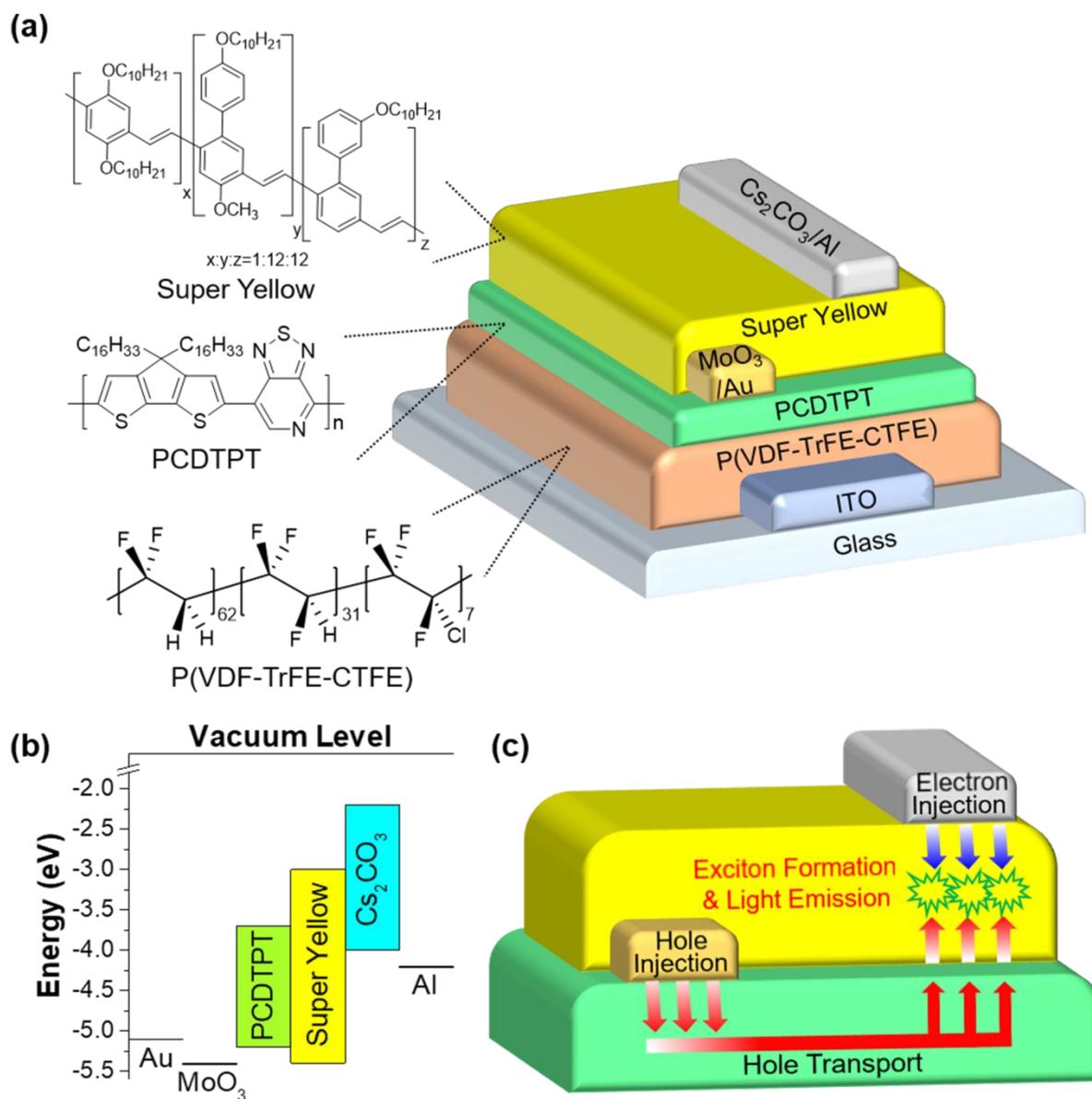


Figure 1. (a) Chemical structures of Super Yellow, PCDTPT, and P(VDF_{0.62}-TrFE_{0.31}-CTFE_{0.7}) polymer light-emitting, hole-transporting, and gate dielectric layer materials, respectively; also shown is a schematic device structure. (b) Energy level diagram for the studied organic light-emitting transistor structures with asymmetric source-drain electrodes. (c) Schematic depiction of the operating mechanism of the OLETs, showing hole injection from the source electrode into PCDTPT, lateral hole transport, electron injection from the drain electrode into Super Yellow, exciton formation within that layer and light emission therefrom.

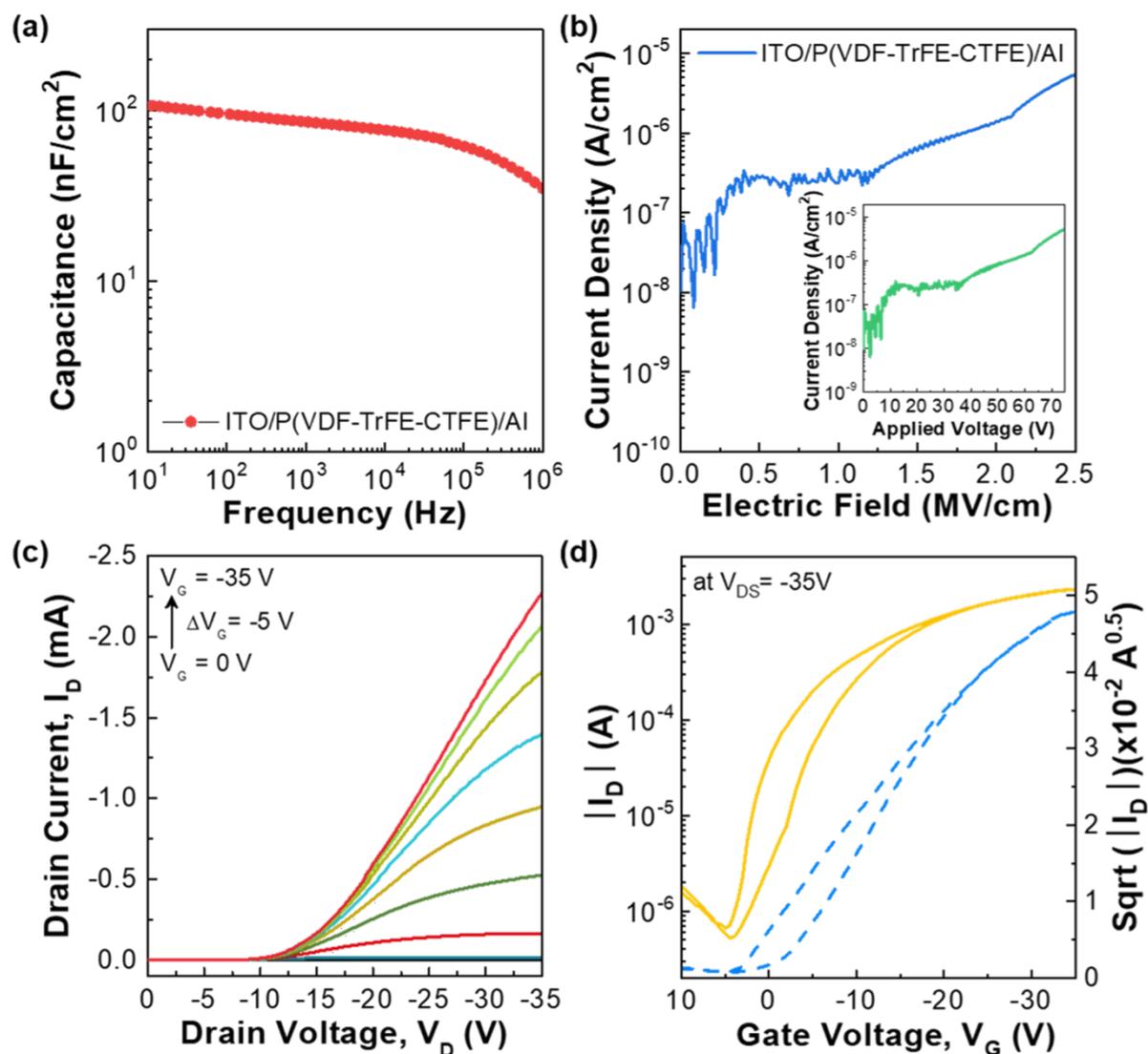


Figure 2. (a) Capacitance-frequency, and (b) current density-electric field characteristics for MIM capacitor structures comprising ITO/P(VDF-TrFE-CTFE)/Al. (c) Output and (d) transfer curves for the studied OLETs.

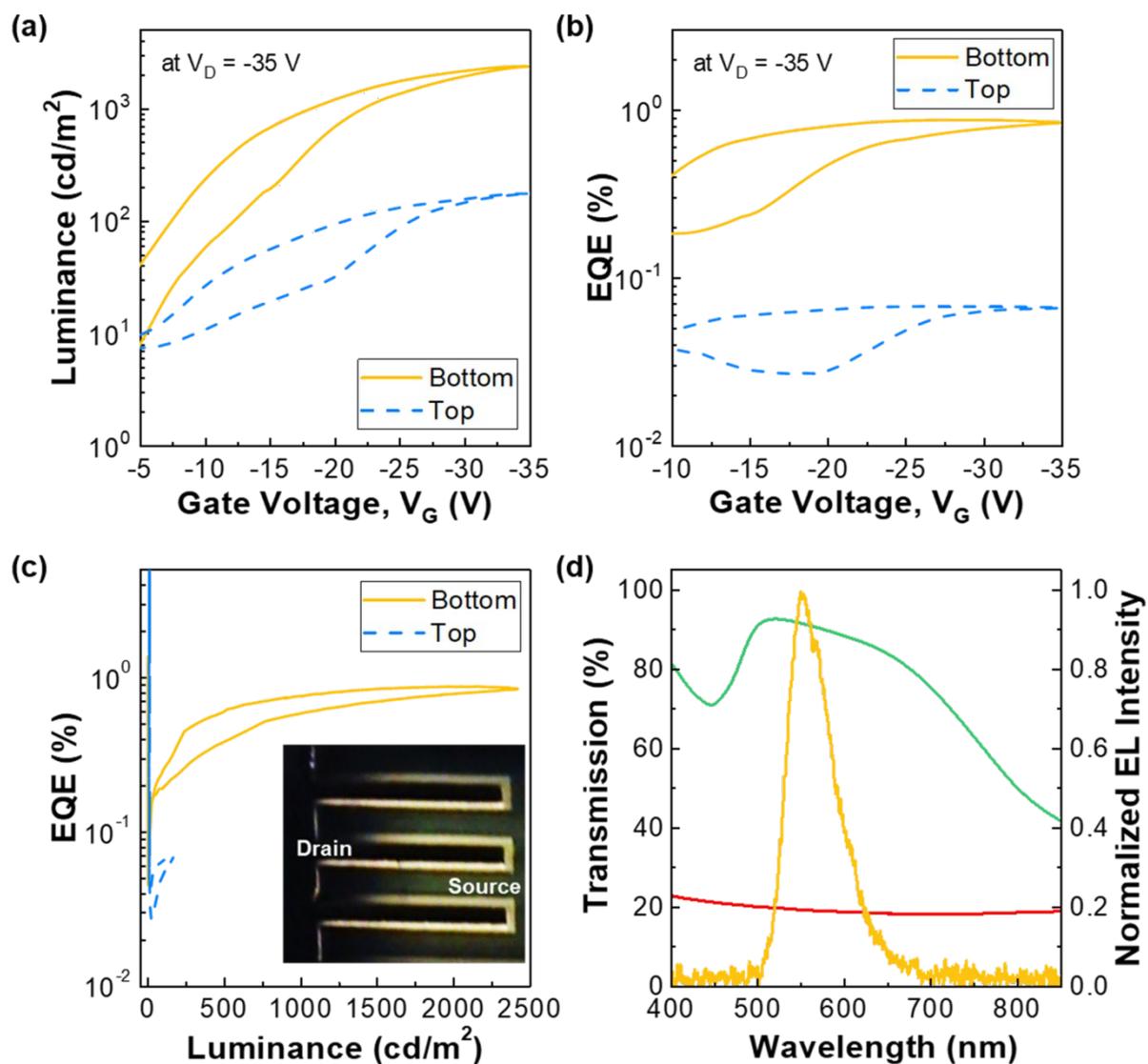


Figure 3. (a) Optical transfer characteristics, (b) external quantum efficiency (EQE), and (c) EQE as a function of device luminance for both top and bottom emission directions. Inset in (c) shows an OLET during operation. (d) Normalized electroluminescence (EL) spectrum and transmittance spectra for the top ($\text{Cs}_2\text{CO}_3/\text{Al}$; red line) and bottom (glass/ITO/P(VDF-TrFE-CTFE)/PCDTPT; green line) electrode stacks.

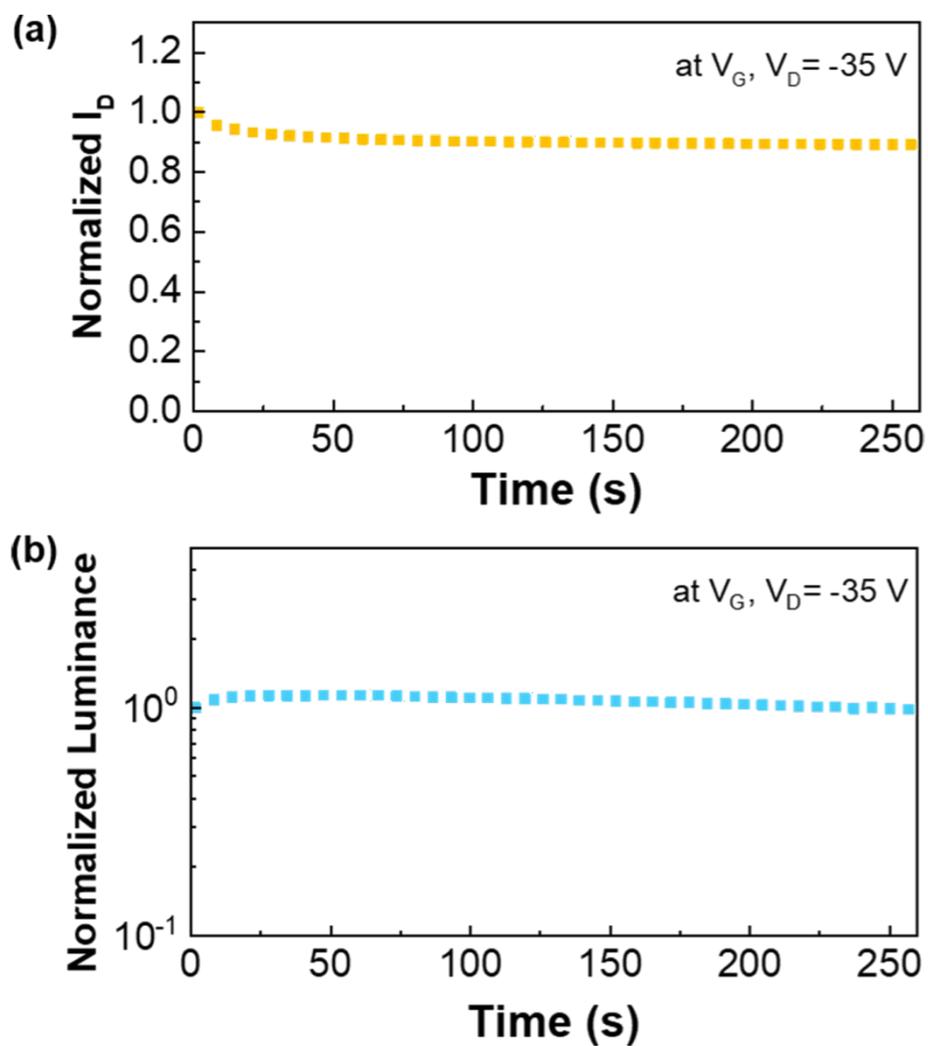


Figure 4. Operational stability for a PCDTPT/Super Yellow OLET. (a) Normalized I_D , and (b) normalized luminance under constant driving conditions of $V_G = V_D = -35$ V.

Table 1. Performance parameters of high-k dielectric-based OLETs (Note that all transistor parameters were extracted from more than 5 devices and the averaged values are displayed).

Mobility ($\text{cm}^2\text{V}^{-1}\text{s}^{-1}$)	On/off ratio	V_{th} (V)	EQE (%)	Luminance (cd m^{-2})
0.55 / 0.39 (Forward / Backward)	4340 / 3460 (Forward / Backward)	-0.83 / 4.36 (Forward / Backward)	0.11 / 0.88 (Top / Bottom Emission)	180 / 2422 (Top / Bottom Emission)

<TOC Graphic>

