

Differing Impacts of Blended Fullerene Acceptors on the Performance of Ternary Organic Solar Cells

David A. Palacios-Gómez¹, Ali M. Huerta Flores¹, Roderick C. I. MacKenzie³, Christopher Pearson¹, Faisal Alanazi², Budhika G. Mendis² and Christopher Groves^{1}*

Departments of Engineering¹ and Physics², Durham University, South Road, Durham, DH1 3LE, UK; Faculty of Engineering³, University of Nottingham, NG7 2RD, UK

*email: chris.groves@durham.ac.uk

ABSTRACT:

Organic Photovoltaic (OPV) devices offer the ability to tune the electronic and optical properties of the active layer by selection of a wide range of molecules, however, their power conversion efficiencies currently lag other competing photovoltaic technologies. One method to enhance their performance is to add a further active material into the absorber layer, resulting in a ternary OPV. However, selecting appropriate ternary blend components to yield an improvement in performance is challenging due to the multitude of materials properties and physical processes that ternary blends can display. Here we perform a systematic set of experiments on OPV ternary blends incorporating either of the donor polymers P3HT or PTB7 and the fullerene acceptors PCBM and ICBA. Some combinations of ternary blends are shown to outperform the reference binaries in terms of open-circuit voltage or short-circuit current, however, this was not observed for all combinations. Improvements in IQE of the order of 25% were observed for PTB7-based ternaries compared to the reference binary, which is attributed to a reduction in charge recombination. All blends showed some improvement in open-circuit voltage with addition of ICBA due to alloying

of the fullerene components, but to differing degrees which is argued to be due to molecular morphology. These findings demonstrate that the benefits one can obtain using a ternary OPV approach vary depending on materials system to an extent that depends upon the ternary blend morphology.

KEYWORDS: ternary blends, power conversion efficiency, organic solar cells, organic photovoltaics, fullerene alloy, morphology.

1. INTRODUCTION

Solution processable organic photovoltaics (OPVs) have attracted a considerable amount of attention in the past decade due to its unique advantages as flexibility, large area fabrication and the promise of low-cost printing which makes this technology an encouraging option for renewable PV applications¹⁻⁵. Presently, power conversion efficiencies (PCEs) exceeding 16% have been realized^{6,7}, which whilst representing a significant improvement on the status quo only a few years ago, are still lower than other competing technologies. One method to address this challenge is to add a further material into the bulk heterojunction (BHJ) donor-acceptor layer to yield a ternary blend⁸⁻¹⁰. The addition of a third component in OPV blends offers the opportunity to optimize the optical and electrical properties beyond which is possible with a binary blend alone¹¹⁻¹³. From an electrical point of view, ternary blends have been shown to reduce recombination via spatial separation of charges¹⁴⁻¹⁷, whilst blends of fullerenes lead to the formation of an effective alloy which improves open-circuit voltage (V_{OC})¹⁸⁻²⁰. For ternary blends to maximize their usefulness as a methodology to improve OPVs will require better understanding of how materials selection, formulation, and device manufacture impact performance. Here we investigate this question by forming ternary blends in two exemplars binary OPV systems, changing blend composition and

additives to enable control over morphology. We demonstrate that the impact of this approach is substantially different for the two systems examined, with varying impacts on both V_{OC} and short-circuit current density (J_{SC}).

Among the different strategies to fabricate ternary blends, fullerene alloys which include polymers blended with a combination of fullerene acceptors (i.e. PCBM:ICBA) have demonstrated to be an effective route to further improve PV properties^{18, 19, 21}. Thompson et al.^{18, 19} first proposed the alloy model in P3HT with PCBM:ICBA, where the V_{OC} could be effectively tuned in ternary solar cells by varying the composition of fullerenes. For instance, a ~40 wt% ICBA resulted in a ~64mV increase in V_{OC} compared to the P3HT:PCBM binary blend, however, J_{SC} decreased for all ICBA combinations¹⁸. These results showed that V_{OC} was not pinned down to the smallest V_{OC} of the binary counterparts but evidenced that ternary OPVs could maximize V_{OC} and exceed binary BHJ efficiencies. Angmo et al.²⁰ further demonstrated that mixed fullerenes form an alloy in ternary blends of P3HT:ICBA:PCBM with solid-state NMR. Following device optimization, a PCE of 2.37% was realized with the ternary blend alloy exceeding its binary equivalents. Correspondingly, other authors have demonstrated that ternary blends with acceptor alloy and novel polymers can indeed lead to higher performing OPVs^{21, 22}. Cheng and co-workers²³ fabricated ternary blends with ICBA:PC₇₁BM and low bandgap PTB7 polymer. Contrarily to P3HT, the enhancement of V_{OC} was moderate with PTB7 as the donor, yet J_{SC} improved until a certain threshold (~20%-30%) prior to reducing. Further, although hole and electron mobilities remained relatively unchanged between the binary and ternary blends, ICBA enhanced the external quantum efficiency (EQE) which was attributed due to a cascade formation. The net result of these effects was a 12% improvement on the PCE. Similarly, Sharma et al.²⁴ introduced ICBA as a third component in PTB7:PC₇₁BM blend to elucidate critical information for OPV design. In their study, 10 wt%

ICBA increased EQE which was correlated to improved charge dissociation as revealed by transient absorption studies, while higher concentrations of ICBA yielded more recombination. More recently, blends of Non-fullerene acceptors (NFA) have also demonstrated encouraging results in state-of-the art OPVs²⁵⁻²⁷. Lu et al.²⁸ presented a ternary blend of PPBDTBT donor with ITIC:PC₇₁BM as the acceptor materials, in which ITIC extended the light absorption whilst PC₇₁BM reduced the aggregation of ITIC as seen with TEM images. Despite an increase in J_{SC} and PCE with the ternary blends, V_{OC} progressively decreased with the increasing weight ratio of the fullerene, thus evidencing no intimate acceptor alloy. Notwithstanding, the reductions in V_{OC} were pinned on to be dependent on the composition of the blend, rather than determined by the energy levels of the materials. These papers have given valuable insights into how specific ternary blend systems operate, however, it is not clear how the effects of a ternary component in an OPV generalise to a specific morphology in new systems. The novelty of our approach, is that we seek to provide insights into this question by comparing the performance of two ternary blend systems based on different polymer donors, but with the same fullerene acceptor and ternary components. We selected the ternary blends PTB7:ICBA:PC₇₁BM and P3HT:ICBA:PC₇₁BM as the reference binary devices (PTB7:PC₇₁BM and P3HT:PC₇₁BM) are well known, thus aiding interpretation of our data. Further, ICBA:PC₇₁BM has been shown to form a fullerene alloy under certain circumstances, and hence the selected ternary blend systems may display a wide range of behaviours that may occur in ternary OPV blends in general. The energy levels of these systems are shown in Fig. 1. In both blend systems, the fraction of ICBA was varied from 0% to 30% and devices were fabricated with and without the processing additive 1,8-diiodooctane (DIO). We have showed that increases in V_{OC} occur in both systems when ICBA is added, as is expected due to alloying, but to differing degrees; with P3HT-based ternaries gaining up to 200meV and PTB7-

based ternaries prepared in the same manner, gaining substantially less. Further, we have demonstrated that J_{SC} increases when ICBA is added for PTB7-based ternaries when the processing additive DIO is used, whereas the J_{SC} of other combinations reduces upon addition of ICBA. To the best of our knowledge, this is the first time the utility of a ternary blend is investigated by comparing the J_{SC} and V_{OC} of these blends and then relating it to the morphology. These data therefore demonstrate that nominally similar approaches to forming a ternary OPV blend can result in significantly different results depending upon the donor polymer.

2. RESULTS AND DISCUSSION

The molecular structure of materials used here is presented in Fig. 1a. Two ternary OPV systems were examined, PTB7:ICBA:PC₇₁BM and P3HT:ICBA:PC₇₁BM the energy levels for which are shown in Fig. 1b²⁹. In all cases the active layer was incorporated into OPV devices with the layer structure glass/ITO/PEDOT:PSS/Active layer/Al, as shown in Fig. 1c. We note that as no interlayers (e.g. LiF or Ca) were used, this will result in lower V_{OC} values than that reported elsewhere in device which do^{30,31}. Ternary BHJ active layers were fabricated with a 1:1.5 donor and fullerenes acceptor ratios for PTB7-OPVs and 1:1 for P3HT-OPVs since these have been observed to be the optimum ratios for each type of blend³²⁻³⁶. Details of the materials, fabrication and measurements are given in the supporting information. A total of 8 nominally same repeats were prepared for each blend system yielding 96 OPV devices; 48 for each ternary blend system in which the wt% of ICBA was increased from 0% (i.e. a binary blend) to 30%. The impact of morphology was examined by fabricating the series of devices both with and without the processing additive DIO with a 3% concentration. Previous investigations with solvent additives have demonstrated that 3 vol% DIO when used as a co-solvent in PTB7 and P3HT blends can assist in achieving higher performances when compared to controls, other additives and different

volume concentrations³⁷⁻³⁹. The devices reported hereafter were chosen as typical examples of that composition, rather than ‘champion’ devices.

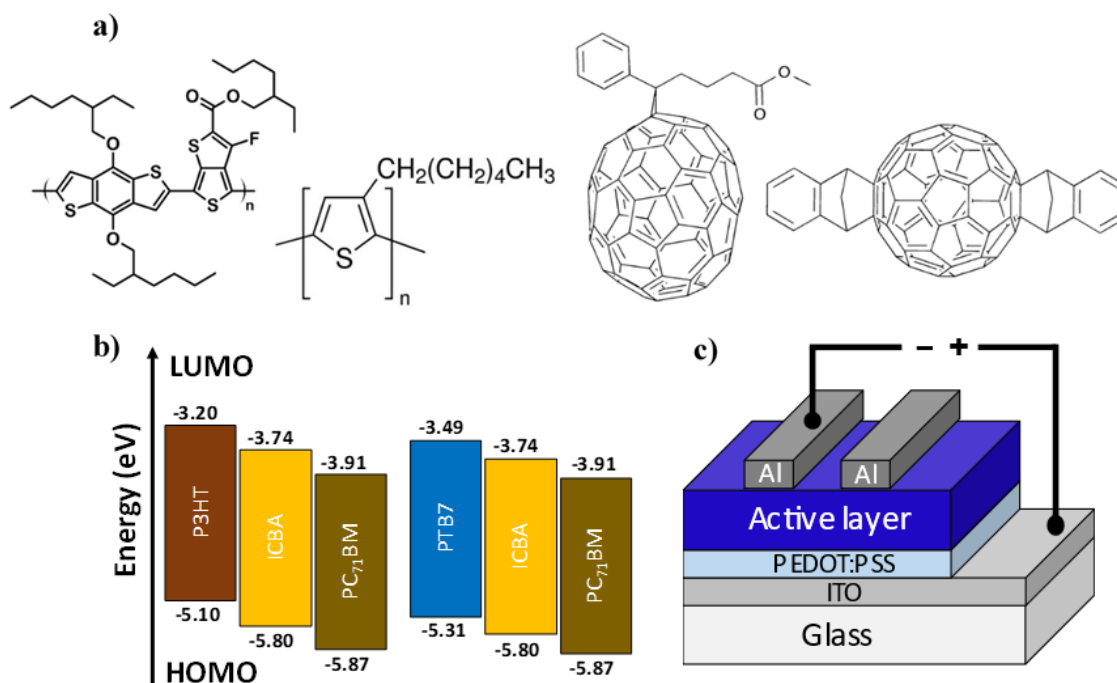


Figure 1. a) Molecular structure of materials used: PTB7, P3HT, PC₇₁BM and ICBA (from left to right), b) energy levels and c) OPV device configuration. Energy levels from^{23, 40}.

2.1 Open circuit voltage

Fig. 2 shows variation in PV parameters for the two ternary blend systems as a function of ICBA composition. Most notable is an increase in V_{OC} as ICBA composition increases. P3HT based blends show the greatest increase in V_{OC} , with the 25 wt% ICBA devices with DIO having an additional 100mV. PTB7-based blends also show an increase in V_{OC} over the majority of the ICBA composition range, but by contrast, the magnitude of increase is much smaller, with the 25 wt% ICBA devices with DIO having only an additional 20mV. Our results therefore also show

that V_{OC} appears to be determined by an effective fullerene alloy, rather than having the V_{OC} pinned to the lowest acceptor energy level^{18-20, 41}. The improvement in V_{OC} for the DIO based P3HT ternary blend lead is sufficient to yield a net improvement in PCE compared to the equivalent binary.

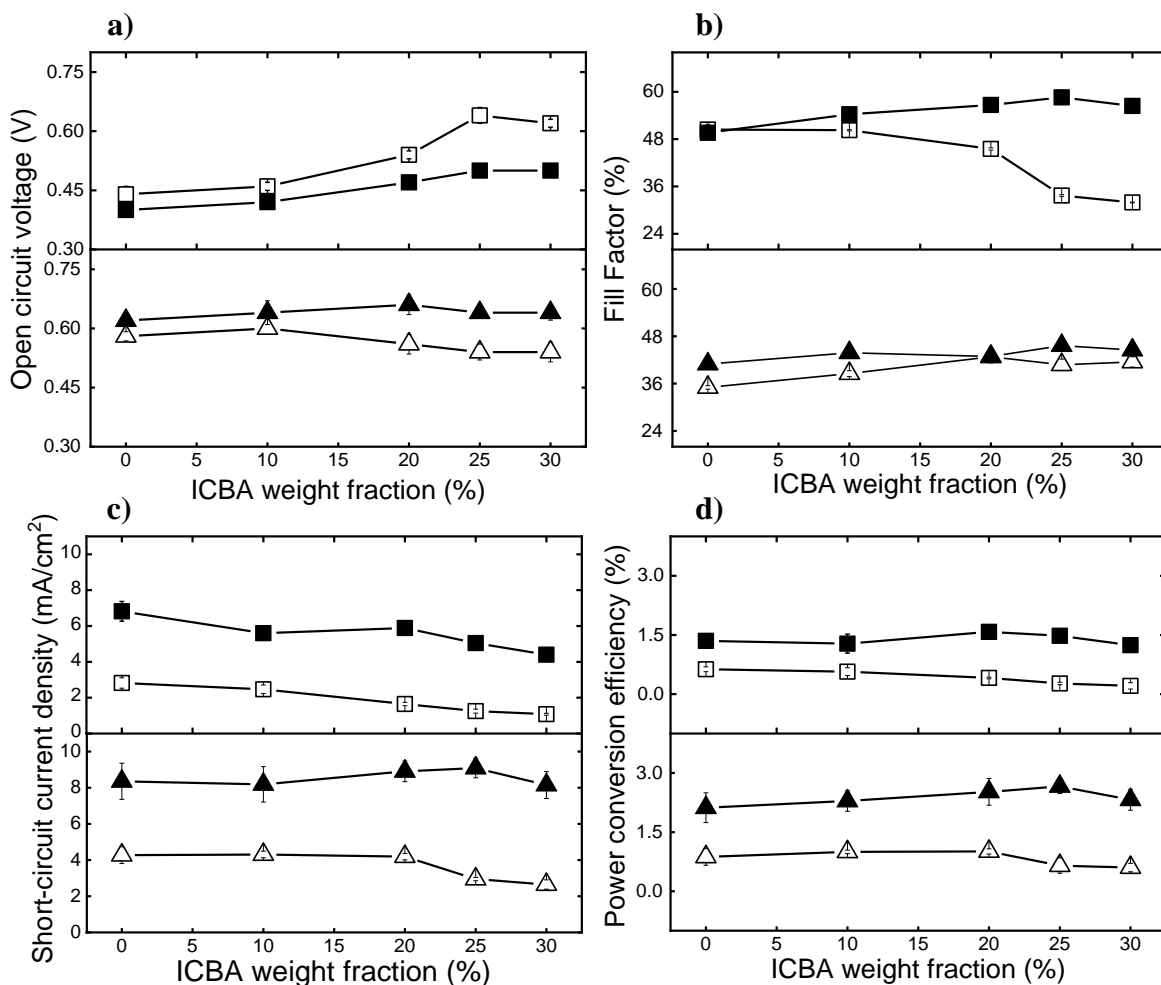


Figure 2. Variation of a) V_{OC} , b) FF , c) J_{SC} and d) PCE as a function of the amount of ICBA added to the P3HT:PC₇₁BM (black square) and PTB7:PC₇₁BM blends (black triangle) in 0%, 10%, 20%, 25% and 30%. Open figures denote OPVs without DIO and solid figures with 3% of DIO respectively.

The difference in the degree of V_{OC} shift for the same change in fullerene composition is not explained in the literature. To explain why, we begin by reasoning that the formation of an alloy implies intimate mixing of the two fullerene components somewhere in the blend. If so, the differing morphology of the two blend systems may hold some clues as to why the alloying behaviour is different. Whilst PCBM has been shown to be miscible in PTB7⁴² and in amorphous P3HT⁴³, resulting in an amorphous polymer:fullerene phase in both blend systems, we note that the side-chain density of P3HT is high compared to many donor:acceptor polymers⁴⁴. An outcome of this high side-chain density is that PCBM does not intercalate into P3HT crystalline regions^{36, 43}. Consequently, we propose that fullerene molecules do not readily interpose between P3HT side-chains in either the crystalline or amorphous phases, and that as a consequence, fullerenes are excluded to other regions in the blend where they form an alloy. Conversely, we argue that the impact of alloying is less strong in the PTB7 blends as fullerenes can more easily interpose between the side-chains, which we term molecular intermixing. There is indirect evidence of molecular intermixing in PTB7:fullerene blends as optimised blends require an excess of fullerene³⁶, side-chains have weak orientational uniformity^{45, 46}, and that PCBM molecules disrupt π - π stacking in PTB type polymers generally⁴⁷.

To further investigate this hypothesis, we investigated literature ternary blend systems utilising an ICBA/PCBM fullerene alloy (Table S1 in the Supplementary Information) and whether the polymer donors supported molecular intermixing with fullerenes (Table S2 in the Supplementary Information). Interestingly we found that polymer donors which did support molecular intermixing (i.e. PTB7 and PBTTT-C14) all had similar small increases in V_{OC} when 30-40wt% of ICBA was added into the acceptor fraction compared to the binary control, in agreement with our hypothesis^{21, 23, 24}, which is consistent with our hypothesis. We note that PBTTT-C14 is an

interesting comparison to the current data as it is both crystalline (as P3HT) and has a structure that permits PCBM molecules to site between side-chains (as argued for PTB7) ⁴⁸, and shows more modest increases in V_{OC} when ICBA is added (like that observed for PTB7) ²¹. P3HT was the only example of a polymer donor that was not favourable to molecular intermixing which has been used in ICBA/PCBM ternary blends, but what data has been reported also agrees with our observation of a larger V_{OC} when ICBA is added as compared to non-intercalated blends ^{18, 19}. Thus, it may be the case that the degree of molecular mixing is a key element in whether the full benefit to V_{OC} provided by a fullerene alloy is observed. Further experiments on donor:acceptor blend systems that do, or do not, support molecular intermixing would be useful to test this assertion. However, we note from our review shown in Table S2, that P3HT is unusual in its high side-chain density, as many other blend systems (e.g. PTT, PQT, PBTTT, PCDTBT, and PFBT-T20TT) do intercalate to some degree ^{36, 49, 50}.

2.2 Fill Factor

Returning to the PV characteristics reported in Fig 2, we observed that the fill factor (FF) is largely unaffected by the addition of ICBA in all cases with the exception of the P3HT ternary where DIO is not used. The devices with DIO maintained a similar FF to the binary control. The reduction in FF for P3HT-based ternaries with more than 20 wt% ICBA may be due to poor charge transport, as TEM and AFM imaging shown in Fig 3 show that DIO promotes an extensive P3HT network of fibrils in high ICBA wt% devices, whilst devices without DIO do not show such structure. However, it is also possible that P3HT-based ternaries with more than 20wt% ICBA are impacted by a poor interface between the film surface and cathode, as the J-V curves for these devices have an S-shaped characteristic.

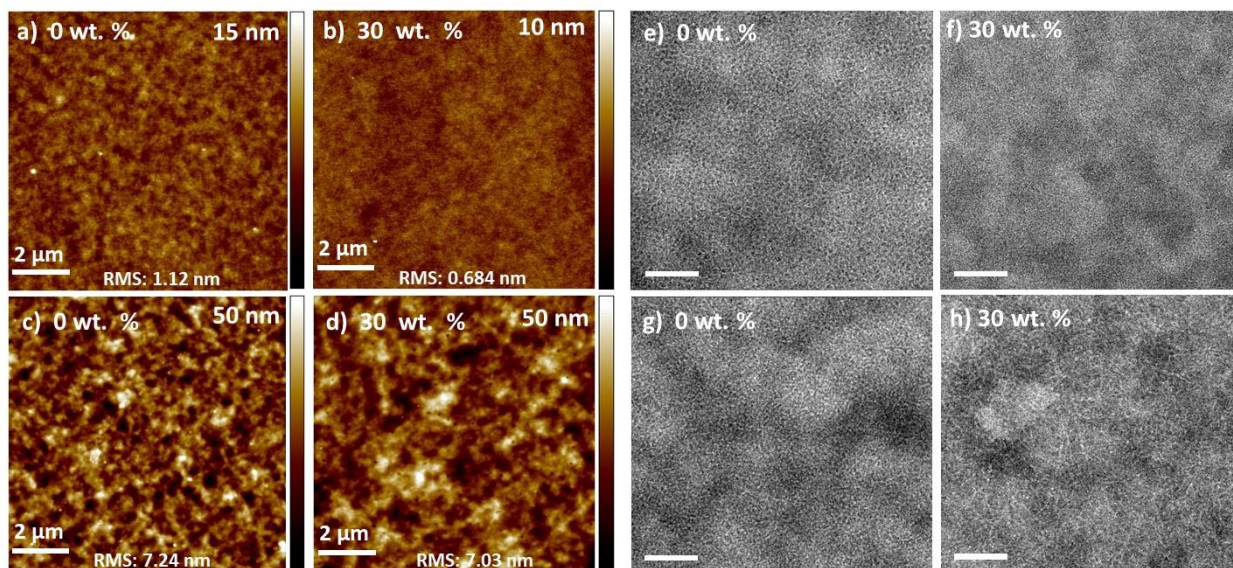


Figure 3. AFM topography and TEM morphology images of P3HT:PC₇₁BM binary blends without (a, e,) and with (c, g,) DIO, and ternary P3HT:ICBA:PC₇₁BM (30 wt%) without (b, f) and with (d, h) DIO. On TEM images, scale bar represents 200 nm.

2.3. Short-circuit current

Fig. 2 also shows that increasing ICBA wt% reduces the J_{SC} for all devices except for PTB7-based devices with DIO, for which the improvement in J_{SC} is sufficient to increase PCE as compared to the equivalent binary. To examine in more detail the changes in J_{SC} , Fig. 4 shows EQE measurements for PTB7:ICBA:PC₇₁BM and P3HT:ICBA:PC₇₁BM OPVs with ICBA content varying from 0 to 30%, both with and without the processing additive DIO. Here we observed that PTB7-based OPVs with added ICBA and DIO have an EQE enhanced by an order of 20% compared to the reference binary, whereas EQE is reduced for other device types.

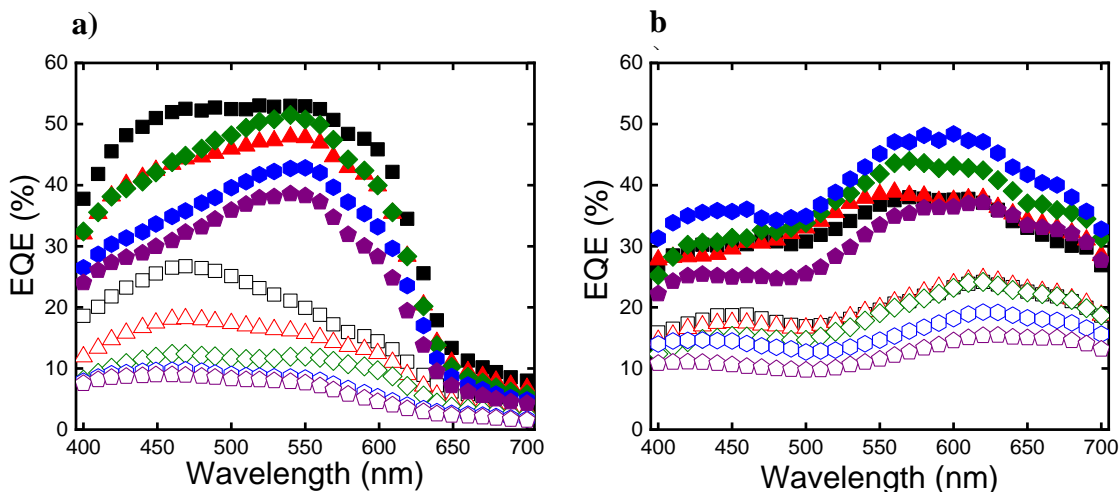


Figure 4. EQE spectra of a) P3HT:PC₇₁BM and b) PTB7:PC₇₁BM prepared with 0% (black) and an additional 10% (red), 20% (green), 25% (blue) and 30% (purple) concentration of ICBA with and without 3% of DIO. Solid figures represent OPVs with DIO and open figures represent OPVs without DIO.

We note that the data also show the positive impact of DIO of EQE in both P3HT- and PTB7-based blends in agreement with the literature^{38, 51-54}. In the case of P3HT based blends, DIO has been shown to improve the formation of P3HT fibrils^{39, 55}, in agreement with TEM and AFM images in Fig. 3. By contrast, PTB7-based blends are improved by the addition of DIO as it limits the size of PCBM aggregates that would otherwise reduce exciton dissociation^{42, 56}. Again, our AFM and TEM analysis in Fig. 5 is in agreement with these literature data, however, we additionally show that DIO also suppresses the size of ICBA aggregates more generally.

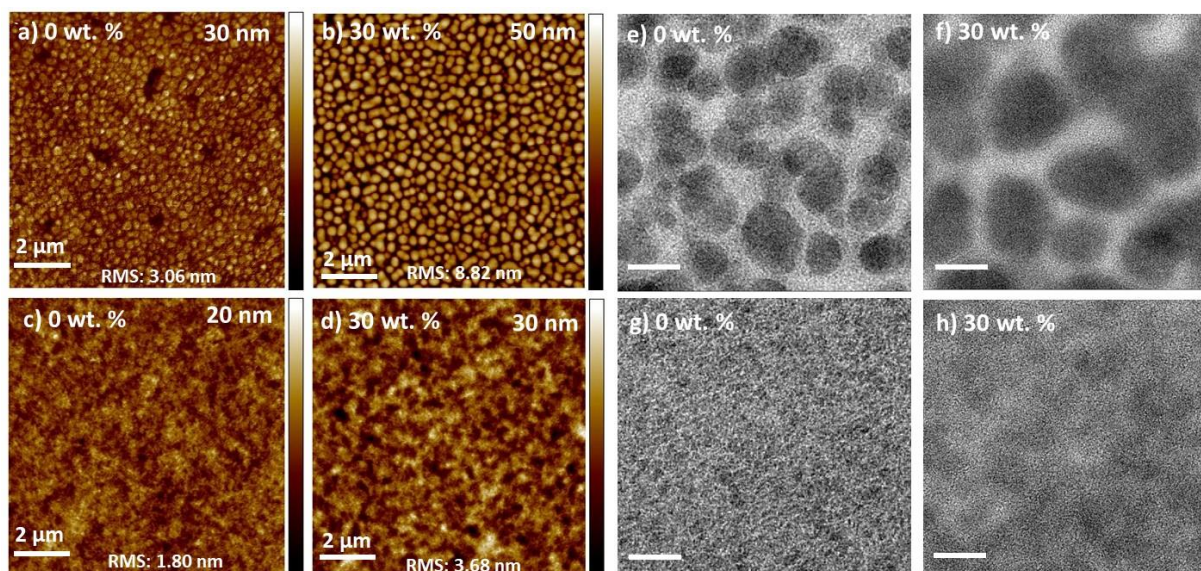


Figure 5. AFM topography and TEM morphology images of PTB7:PC₇₁BM binary blends without (a, e,) and with (c, g,) DIO, and ternary PTB7:ICBA:PC₇₁BM (30 wt%) without (b, f) and with (d, h) DIO. On TEM images, scale bar represents 200 nm.

Returning to the impact of the ternary blend on EQE, we note that the changes in EQE shown in Fig. 4 should be viewed in the context of reducing absorption as wt% of ICBA increases, shown in Fig. 6. In all cases the addition of ICBA reduced absorption as expected due to the weaker absorption in ICBA (Fig. S2) compared to the other active components^{18, 24, 57}.

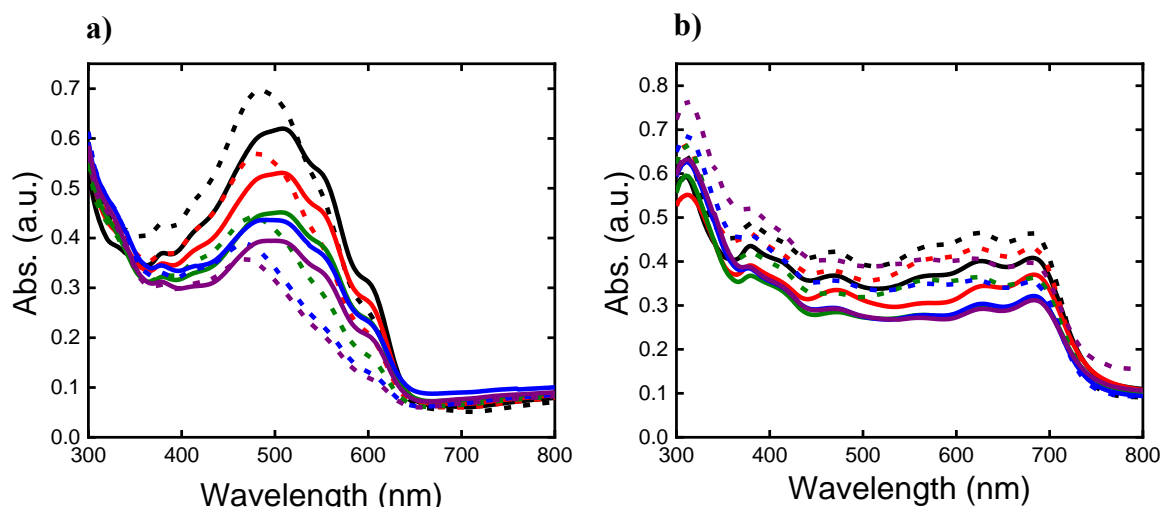


Figure 6. Optical absorption spectra of a) P3HT:PC₇₁BM and b) PTB7:PC₇₁BM blends with 0% (black), 10% (red), 20% (green), 25% (blue) and 30% (purple) ICBA loading, with and without 3% of DIO. Solid lines denote OPVs with DIO and short-dash lines denote OPVs without DIO.

Thus to account for the effect of changing absorption the internal quantum efficiency (IQE) is estimated by dividing EQE by the simulated absorption of the active layer using the transfer matrix method described by Burkhard et al.⁵⁸ as outlined in the supplementary information. Estimated IQE are shown for P3HT- and PTB7-based OPVs in Fig. 7a and 7b respectively. Focussing our attention on the impact of ICBA on the IQE for the reported blends, we observed that the IQE for the PTB7-based ternaries with DIO increase by as much as 25 wt% for when DIO is used. Hence, although absorption changes with composition, the IQE data suggest that ICBA is acting to improve recombination in the PTB7-based blend with DIO. We now consider possible explanations as to why the IQE of PTB7-based ternaries with DIO are improved by ICBA, whilst it is reduced for the other systems examined. Understanding the underpinning physics is critical should lessons be learned for the design of future ternary OPV devices.

First, we consider the possibility of changing exciton dissociation. TEM imaging (Fig. 3, 5) suggests fine intermixing of the donor and acceptors ($<20\text{nm}$) for P3HT based blends with and without DIO, and PTB7 blends with DIO, even with the addition of 30 wt% ICBA. By contrast, PTB7-based devices without DIO show large ($>150\text{nm}$) fullerene domains which increase in size with addition of ICBA (Fig 5e, f) which would be expected to reduce IQE through reduced exciton dissociation efficiency, as is observed in the data. Whilst changing exciton dissociation efficiency appears to explain reducing IQE as ICBA composition is increased for the PTB7-based ternaries without DIO, we note that the reduction in IQE up to 25 wt% ICBA is only slight ($\sim 4\%$), which contrasts to an increase in fullerene-rich domain size ($\sim 100\text{nm}$) which is many times larger than the exciton diffusion length ($\sim 10\text{nm}$)⁵⁹⁻⁶².

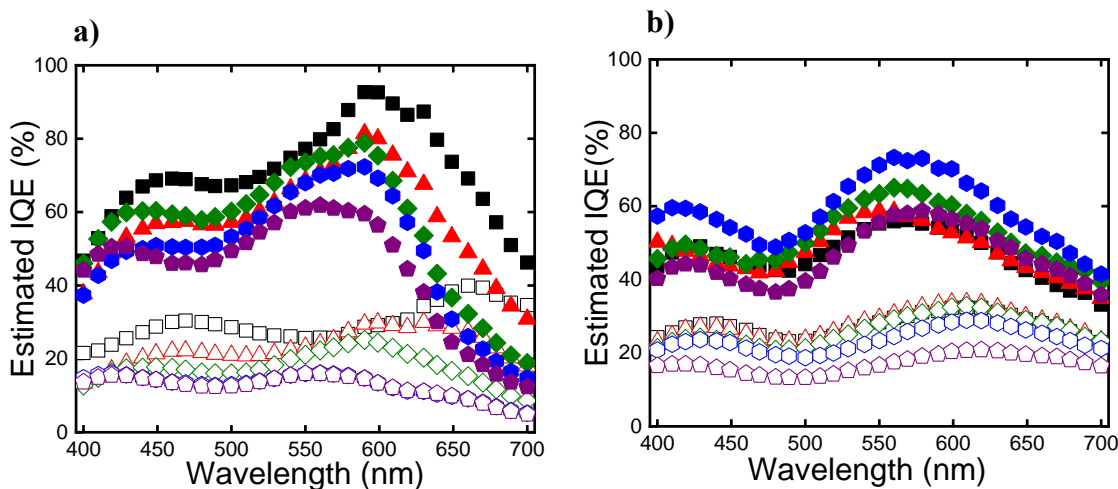


Figure 7. Estimated IQE curves of a) P3HT:PC₇₁BM and b) PTB7:PC₇₁BM prepared with 0% (black) and an additional 10% (red), 20% (green), 25% (blue) and 30% (purple) concentration of ICBA with and without 3% of DIO. Solid figures represent OPVs with DIO and open figures represent OPVs without DIO.

It may therefore be the case that charge recombination reduces in the PTB7-based device without DIO as well as the PTB7-based device with DIO. However, without a detailed morphological model of the PTB7-based blend without DIO as ICBA is increased it is not possible to quantitatively examine the competition between exciton and charge further. Next, we consider the possible impact of changing charge transport. In other studies of ternary blends, changes in IQE have been shown to be due to improvements or otherwise in charge transport through the blend ^{63, 64}. When considering the current data, adding ICBA increases the overall fraction of acceptor and reduces the fraction of donor. Recalling the literature, P3HT:PCBM devices are susceptible to being limited by hole transport ⁶⁵, whilst thick PTB7:PCBM devices are susceptible to being limited by electron transport ⁶⁶. Thus, it is possible that increasing ICBA content may be more detrimental to the performance of P3HT based blends rather than PTB7 based blends as the former is more sensitive to poor hole transport. We observe hallmarks of this hypothesis in the data, as dark forward diode current increases with ICBA for PTB7-based devices (Fig. S1c), whilst it reduces for P3HT-based devices (Fig. S1a). However, we point out that the reference binary devices have the optimal polymer:PCBM ratio, thus one may expect that the addition of ICBA would push the blend away from this optimum and so reduce efficiency from a charge transport point of view. To investigate this balance further, we performed drift diffusion modelling using the General-Purpose Photovoltaic Device Model (GPVDM). This software has been used to evaluate in detail a number of different OPV devices ⁶⁷⁻⁶⁹. For interested readers, complete details on how the model operates are presented here ⁷⁰. In brief, GPVDM solves drift-diffusion equations for electrons and holes along with Shockley-Read-Hall capture/escape on a finite difference mesh. The transfer matrix method and Poisson's equation are solved for optical and electrostatic effects respectively. With this approach, the model describes the distribution of carriers in position and energy space, as well

as simulate J-V characteristics. In this work, the modelled architecture of devices was ITO (100nm) / PEDOT:PSS (30nm) / Active layer (150nm) / Al (100nm). In the supplementary information (Fig. S3a), we fit to dark J-V curves for PTB7-based blends for the range of ICBA compositions by changing the electron mobility alone. Whilst we could recreate the increase in diode current observed with increasing ICBA content, this approach did not recreate the observed increase in photocurrent (Fig S3b). Hence, we view it unlikely that the improvement in IQE observed for the PTB7-based blends with ICBA is due to changing mobility alone.

Finally, it has been proposed that ternary blends can act to reduce recombination if the energy levels of the constituent materials form a ‘staircase’ in which separation of the charge pair is energetically favoured^{14, 15, 71, 72}, in much the same manner as photosynthesis. This ‘cascade’ effect, and a consequent reduction in recombination has been observed directly in some evaporated systems⁷³⁻⁷⁵. The question remains, however, as to why we observe an increase in IQE for PTB7 based blends with increasing ICBA, but not for P3HT based blends. We propose that this difference can be attributed to the alloying effect discussed previously. In P3HT based blends, fullerene alloying appears to occur readily, and as such, we may expect that any possible advantage from a cascade heterojunction to be lost. Contrastingly, PTB7 based blends appear to show fullerene alloying to a lesser degree, suggesting that electronically distinct fullerene domains may be more prevalent, in turn leading to the possibility of cascade heterojunctions.

3. CONCLUSIONS

In this paper we have analysed the performance of two ternary OPV blend systems, PTB7:ICBA:PC₇₁BM and P3HT:ICBA:PC₇₁BM, and compared them to reference binary OPVs

(PTB7:PC₇₁BM and P3HT:PC₇₁BM) to better understand the contrasting impacts of a ternary approach to optimising performance on OPVs. The impact of the ICBA ternary component was shown to be different for the two donor polymers. P3HT-based ternaries showed the largest increase in V_{OC} as compared to PTB7-based ternaries, notwithstanding the same concentrations of ternary and acceptor components in each series. Further, the EQE of the PTB7-based devices with DIO was shown to increase by 20% upon addition of ICBA, whilst the EQE reduced for all other investigated systems. We hypothesise that the observed differences are rooted in the molecular morphology of the systems examined, with the fullerene compounds being able to interpose in between the side-chains of PTB7, but not in the case of P3HT. This is argued to lead to larger V_{OC} shifts for the P3HT-based devices, due to greater molecular mixing of the fullerene components; and greater reduction in recombination due to a cascade effect in the PTB7-based devices that arises due to more electronically distinct fullerene molecules. These data demonstrate both that ternary blends can have differing impacts upon the physical operation of the OPV, and that the behaviour observed depends upon the donor polymer. More speculatively, the data may point to a common cause to the differing impacts of adding a ternary component to an OPV blend, namely the degree of intermixing between the ternary component and other constituents. This motivates further experiments of ternary OPV systems to investigate whether choice of acceptors that can, and cannot, interpose between donor side-chains can control the effects of ternary blends.

ASSOCIATED CONTENT

Supporting information:

More details on the characterization of materials, OPV fabrication, UV-vis of neat films, dark/light J-Vs, PV table, AFM, TEM and modelling fits are available in the supporting information.

AUTHOR INFORMATION

Corresponding author

Chris Groves – Department of Engineering, Durham University, South Road, Durham, DH1 3LE, United Kingdom. Orcid: 0000-0003-2402-1618 Email: chris.groves@durham.ac.uk

Authors

David A. Palacios-Gómez – Department of Engineering, Durham University, South Road, Durham, DH1 3LE, United Kingdom. Orcid: 0000-0003-1432-9168

Ali M. Huerta-Flores – Department of Engineering, Durham University, South Road, Durham, DH1 3LE, United Kingdom. Orcid: 0000-0001-8304-0590

Roderick C. I. MacKenzie – Faculty of Engineering, University of Nottingham, NG7 2RD, United Kingdom. Orcid: 0000-0002-8833-2872

Chris Pearson – Department of Engineering, Durham University, South Road, Durham, DH1 3LE, United Kingdom. Orcid:

Faisal Alanazi – Department of Physics, Durham University, South Road, Durham, DH1 3LE, United Kingdom. Orcid: 0000-0002-9478-2394

Budhika G. Mendis – Department of Physics, Durham University, South Road, Durham, DH1 3LE, United Kingdom. Orcid: 0000-0003-2334-2866

Author Contributions

The manuscript was written by contributions of all authors. All authors have given approval for the final version of the manuscript.

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REFERENCES:

1. Mazziro, K. A.; Luscombe, C. K., The Future of Organic Photovoltaics. *Chem. Soc. Rev.* **2015**, *44*, 78-90.
2. Wang, Q.; Xie, Y.; Soltani-Kordshuli, F.; Eslamian, M., Progress in Emerging Solution-Processed Thin Film Solar Cells – Part I: Polymer Solar Cells. *Renew. Sustain. Energy Rev.* **2016**, *56*, 347-361.
3. Espinosa, N.; Hösel, M.; Angmo, D.; Krebs, F. C., Solar Cells with One-Day Energy Payback for the Factories of the Future. *Energy Environ. Sci.* **2012**, *5*, 5117-5132.
4. Krebs, F. C.; Tromholt, T.; Jorgensen, M., Upscaling of Polymer Solar Cell Fabrication Using Full Roll-to-Roll Processing. *Nanoscale* **2010**, *2*, 873-86.

5. Forrest, S., The Path to Ubiquitous and Low-Cost Organic Electronic Appliances on Plastic. *Nature* **2004**, *428*, 911-918.
6. Pan, M.-A.; Lau, T.-K.; Tang, Y.; Wu, Y.-C.; Liu, T.; Li, K.; Chen, M.-C.; Lu, X.; Ma, W.; Zhan, C., 16.7%-Efficiency Ternary Blended Organic Photovoltaic Cells with PCBM as the Acceptor Additive to Increase the Open-Circuit Voltage and Phase Purity. *J. Mater. Chem. A* **2019**, *7*, 20713-20722.
7. Yuan, J.; Zhang, Y.; Zhou, L.; Zhang, G.; Yip, H.-L.; Lau, T.-K.; Lu, X.; Zhu, C.; Peng, H.; Johnson, P. A.; Leclerc, M.; Cao, Y.; Ulanski, J.; Li, Y.; Zou, Y., Single-Junction Organic Solar Cell with over 15% Efficiency Using Fused-Ring Acceptor with Electron-Deficient Core. *Joule* **2019**, *3*, 1140-1151.
8. Scaccabarozzi, A. D.; Stingelin, N., Semiconducting:Insulating Polymer Blends for Optoelectronic Applications—A Review of Recent Advances. *J. Mater. Chem. A* **2014**, *2*, 10818-10824.
9. Ferenczi, T. A.; Muller, C.; Bradley, D. D.; Smith, P.; Nelson, J.; Stingelin, N., Organic Semiconductor:Insulator Polymer Ternary Blends for Photovoltaics. *Adv. Mater.* **2011**, *23*, 4093-4097.
10. An, Q.; Zhang, F.; Sun, Q.; Zhang, M.; Zhang, J.; Tang, W.; Yin, X.; Deng, Z., Efficient Organic Ternary Solar Cells with the Third Component as Energy Acceptor. *Nano Energy* **2016**, *26*, 180-191.
11. Ameri, T.; Khoram, P.; Min, J.; Brabec, C. J., Organic Ternary Solar Cells: A Review. *Adv. Mater.* **2013**, *25*, 4245-4266.
12. An, Q.; Zhang, F.; Zhang, J.; Tang, W.; Deng, Z.; Hu, B., Versatile Ternary Organic Solar Cells: A Critical Review. *Energy Environ. Sci.* **2016**, *9*, 281-322.

13. Bi, P.; Hao, X., Versatile Ternary Approach for Novel Organic Solar Cells: A Review. *Sol. RRL* **2019**, *3*, 1800263.
14. Tan, Z. K.; Johnson, K.; Vaynzof, Y.; Bakulin, A. A.; Chua, L. L.; Ho, P. K.; Friend, R. H., Suppressing Recombination in Polymer Photovoltaic Devices Via Energy-Level Cascades. *Adv. Mater.* **2013**, *25*, 4131-4138.
15. Groves, C., Suppression of Geminate Charge Recombination in Organic Photovoltaic Devices with a Cascaded Energy Heterojunction. *Energy Environ. Sci.* **2013**, *6*, 1546-1551.
16. Huang, T.-Y.; Patra, D.; Hsiao, Y.-S.; Chang, S. H.; Wu, C.-G.; Ho, K.-C.; Chu, C.-W., Efficient Ternary Bulk Heterojunction Solar Cells Based on Small Molecules Only. *J. Mater. Chem. A* **2015**, *3*, 10512-10518.
17. An, Q.; Zhang, F.; Li, L.; Zhuo, Z.; Zhang, J.; Tang, W.; Teng, F., Enhanced Performance of Polymer Solar Cells by Employing a Ternary Cascade Energy Structure. *Phys. Chem. Chem. Phys.* **2014**, *16*, 16103-16109.
18. Khlyabich, P. P.; Burkhart, B.; Thompson, B. C., Efficient Ternary Blend Bulk Heterojunction Solar Cells with Tunable Open-Circuit Voltage. *J. Am. Chem. Soc.* **2011**, *133*, 14534-14537.
19. Street, R. A.; Davies, D.; Khlyabich, P. P.; Burkhart, B.; Thompson, B. C., Origin of the Tunable Open-Circuit Voltage in Ternary Blend Bulk Heterojunction Organic Solar Cells. *J. Am. Chem. Soc.* **2013**, *135*, 986-989.
20. Angmo, D.; Bjerring, M.; Nielsen, N. C.; Thompson, B. C.; Krebs, F. C., Fullerene Alloy Formation and the Benefits for Efficient Printing of Ternary Blend Organic Solar Cells. *J. Mater. Chem. C* **2015**, *3*, 5541-5548.

21. Khlyabich, P. P.; Sezen-Edmonds, M.; Howard, J. B.; Thompson, B. C.; Loo, Y.-L., Formation of Organic Alloys in Ternary-Blend Solar Cells with Two Acceptors Having Energy-Level Offsets Exceeding 0.4 eV. *ACS Energy Lett.* **2017**, *2*, 2149-2156.
22. Peng, Z.; Xia, Y.; Gao, F.; Xiong, K.; Hu, Z.; James, D. I.; Chen, J.; Wang, E.; Hou, L., A Dual Ternary System for Highly Efficient ITO-Free Inverted Polymer Solar Cells. *J. Mater. Chem. A* **2015**, *3*, 18365-18371.
23. Cheng, P.; Li, Y.; Zhan, X., Efficient Ternary Blend Polymer Solar Cells with Indene-C60 Bisadduct as an Electron-Cascade Acceptor. *Energy Environ. Sci.* **2014**, *7*, 2005-2011.
24. Sharma, R.; Lee, H.; Gupta, V.; Kim, H.; Kumar, M.; Sharma, C.; Chand, S.; Yoo, S.; Gupta, D., Photo-Physics of PTB7, PCBM and ICBA Based Ternary Solar Cells. *Org. Electron.* **2016**, *34*, 111-117.
25. Chen, Y.; Ye, P.; Jia, X.; Gu, W.; Xu, X.; Wu, X.; Wu, J.; Liu, F.; Zhu, Z.-G.; Huang, H., Tuning Voc for High Performance Organic Ternary Solar Cells with Non-Fullerene Acceptor Alloys. *J. Mater. Chem. A* **2017**, *5*, 19697-19702.
26. Liu, T.; Guo, Y.; Yi, Y.; Huo, L.; Xue, X.; Sun, X.; Fu, H.; Xiong, W.; Meng, D.; Wang, Z.; Liu, F.; Russell, T. P.; Sun, Y., Ternary Organic Solar Cells Based on Two Compatible Nonfullerene Acceptors with Power Conversion Efficiency >10%. *Adv. Mater.* **2016**, *28*, 10008-10015.
27. Baran, D.; Ashraf, R. S.; Hanifi, D. A.; Abdelsamie, M.; Gasparini, N.; Rohr, J. A.; Holliday, S.; Wadsworth, A.; Lockett, S.; Neophytou, M.; Emmott, C. J.; Nelson, J.; Brabec, C. J.; Amassian, A.; Salleo, A.; Kirchartz, T.; Durrant, J. R.; McCulloch, I., Reducing the Efficiency-Stability-Cost Gap of Organic Photovoltaics with Highly Efficient and Stable Small Molecule Acceptor Ternary Solar Cells. *Nat. Mater.* **2017**, *16*, 363-369.

28. Lu, H.; Zhang, J.; Chen, J.; Liu, Q.; Gong, X.; Feng, S.; Xu, X.; Ma, W.; Bo, Z., Ternary-Blend Polymer Solar Cells Combining Fullerene and Nonfullerene Acceptors to Synergistically Boost the Photovoltaic Performance. *Adv. Mater.* **2016**, *28*, 9559-9566.
29. Cheng, P.; Yan, C.; Wu, Y.; Wang, J.; Qin, M.; An, Q.; Cao, J.; Huo, L.; Zhang, F.; Ding, L.; Sun, Y.; Ma, W.; Zhan, X., Alloy Acceptor: Superior Alternative to PCBM toward Efficient and Stable Organic Solar Cells. *Adv. Mater.* **2016**, *28*, 8021-8028.
30. Brabec, C. J.; Shaheen, S. E.; Winder, C.; Sariciftci, N. S.; Denk, P., Effect of LiF/Metal Electrodes on the Performance of Plastic Solar Cells. *Appl. Phys. Lett.* **2002**, *80* (7), 1288-1290.
31. Yu, H.; Li, Y.; Dong, Y.; Huang, X., Fabrication and Optimization of Polymer Solar Cells Based on P3HT:PC70BM System. *Int. J. Photoenergy* **2016**, *2016*, 1-8.
32. Liang, Y.; Xu, Z.; Xia, J.; Tsai, S. T.; Wu, Y.; Li, G.; Ray, C.; Yu, L., For the Bright Future-Bulk Heterojunction Polymer Solar Cells with Power Conversion Efficiency of 7.4%. *Adv. Mater.* **2010**, *22*, E135-E138.
33. Padinger, F., Rittberger, R. and Sariciftci, N., Effects of Postproduction Treatment on Plastic Solar Cells. *Adv. Funct. Mater.* **2003**, *13*, 85-88.
34. van Bavel, S.; Sourty, E.; de With, G.; Frolic, K.; Loos, J., Relation Between Photoactive Layer Thickness, 3D Morphology, and Device Performance in P3HT/PCBM Bulk-Heterojunction Solar Cells. *Macromolecules* **2009**, *42*, 7396-7403.
35. Li, G.; Shrotriya, V.; Huang, J.; Yao, Y.; Moriarty, T.; Emery, K.; Yang, Y., High-Efficiency Solution Processable Polymer Photovoltaic Cells by Self-Organization of Polymer Blends. *Nat. Mater.* **2005**, *4*, 864-868.
36. Mayer, A. C.; Toney, M. F.; Scully, S. R.; Rivnay, J.; Brabec, C. J.; Scharber, M.; Koppe, M.; Heeney, M.; McCulloch, I.; McGehee, M. D., Bimolecular Crystals of Fullerenes in

Conjugated Polymers and the Implications of Molecular Mixing for Solar Cells. *Adv. Funct. Mater.* **2009**, *19*, 1173-1179.

37. Zhao, X.; Xiang, J.; Liu, D.; Zhou, D.; Wang, G.; Zhou, G.; Alameh, K.; Ding, B.; Song, Q., Impact of Alkyl Chain Length of 1,n-diiodoalkanes on PC71BM Distribution in Both Bulk and Air Surface of PTB7:PC71BM Film. *Org. Electron.* **2016**, *37*, 358-365.

38. Kim, W.; Kim, J. K.; Kim, E.; Ahn, T. K.; Wang, D. H.; Park, J. H., Conflicted Effects of a Solvent Additive on PTB7:PC71BM Bulk Heterojunction Solar Cells. *J. Phys. Chem. C* **2015**, *119*, 5954-5961.

39. Fan, X.; Zhao, S.-L.; Chen, Y.; Zhang, J.; Yang, Q.-Q.; Gong, W.; Yuan, M.-Y.; Xu, Z.; Xu, X.-R., Nano Structure Evolution in P3HT:PC61BM Blend Films Due to the Effects of Thermal Annealing or by Adding Solvent. *Chin. Phys. B* **2015**, *24*, 078401.

40. Jin Young Kim, K. L., Nelson E. Coates, Daniel Moses, Thuc-Quyen Nguyen, Mark Dante, Alan J. Heeger, Efficient Tandem Polymer Solar Cells Fabricated by All-Solution Processing. *Science* **2007**, *317*, 222-225.

41. Lami, V.; Hofstetter, Y. J.; Butscher, J. F.; Vaynzof, Y., Energy Level Alignment in Ternary Organic Solar Cells. *Adv. Electron. Mater.* **2020**, *6* (8).

42. Collins, B. A.; Li, Z.; Tumbleston, J. R.; Gann, E.; McNeill, C. R.; Ade, H., Absolute Measurement of Domain Composition and Nanoscale Size Distribution Explains Performance in PTB7:PC71BM Solar Cells. *Adv. Energy Mater.* **2013**, *3*, 65-74.

43. Collins, B. A.; Gann, E.; Guignard, L.; He, X.; McNeill, C. R.; Ade, H., Molecular Miscibility of Polymer–Fullerene Blends. *J. Phys. Chem. Lett.* **2010**, *1* (21), 3160-3166.

44. Kleinschmidt, A. T.; Root, S. E.; Lipomi, D. J., Poly(3-hexylthiophene) (P3HT): Fruit Fly or Outlier in Organic Solar Cell Research? *J. Mater. Chem. A* **2017**, *5* (23), 11396-11400.

45. Szarko, J. M.; Rolczynski, B. S.; Lou, S. J.; Xu, T.; Strzalka, J.; Marks, T. J.; Yu, L.; Chen, L. X., Photovoltaic Function and Exciton/Charge Transfer Dynamics in a Highly Efficient Semiconducting Copolymer. *Adv. Funct. Mater.* **2014**, *24* (1), 10-26.
46. Matthew R. Hammond, R. J. K., Andrew A. Herzing, Lee J. Richter, David S. Germack, Hyun-Wook Ro, Christopher L. Soles, Daniel A. Fischer, Tao Xu, Luping Yu, Michael F. Toney, and Dean M. DeLongchamp, Molecular Order in High-Efficiency Polymer/Fullerene Bulk Heterojunction Solar Cells. *ACS Nano* **2011**, *5* (10), 8248-8257.
47. Szarko, J. M.; Guo, J.; Liang, Y.; Lee, B.; Rolczynski, B. S.; Strzalka, J.; Xu, T.; Loser, S.; Marks, T. J.; Yu, L.; Chen, L. X., When Function Follows Form: Effects of Donor Copolymer Side Chains on Film Morphology and BHJ Solar Cell Performance. *Adv. Mater.* **2010**, *22* (48), 5468-5472.
48. Miller, N. C.; Cho, E.; Gysel, R.; Risko, C.; Coropceanu, V.; Miller, C. E.; Sweetnam, S.; Sellinger, A.; Heeney, M.; McCulloch, I.; Brédas, J.-L.; Toney, M. F.; McGehee, M. D., Factors Governing Intercalation of Fullerenes and Other Small Molecules Between the Side Chains of Semiconducting Polymers Used in Solar Cells. *Adv. Energy Mater.* **2012**, *2*, 1208-1217.
49. Beiley, Z. M.; Hoke, E. T.; Noriega, R.; Dacuña, J.; Burkhard, G. F.; Bartelt, J. A.; Salleo, A.; Toney, M. F.; McGehee, M. D., Morphology-Dependent Trap Formation in High Performance Polymer Bulk Heterojunction Solar Cells. *Adv. Energy Mater.* **2011**, *1*, 954-962.
50. Xiao, T.; Xu, H.; Grancini, G.; Mai, J.; Petrozza, A.; Jeng, U. S.; Wang, Y.; Xin, X.; Lu, Y.; Choon, N. S.; Xiao, H.; Ong, B. S.; Lu, X.; Zhao, N., Molecular Packing and Electronic Processes in Amorphous-Like Polymer Bulk Heterojunction Solar Cells with Fullerene Intercalation. *Sci. Rep.* **2014**, *4*, 5211.

51. Das, S.; Keum, J. K.; Browning, J. F.; Gu, G.; Yang, B.; Dyck, O.; Do, C.; Chen, W.; Chen, J.; Ivanov, I. N.; Hong, K.; Rondinone, A. J.; Joshi, P. C.; Geohegan, D. B.; Duscher, G.; Xiao, K., Correlating High Power Conversion Efficiency of PTB7:PC71BM Inverted Organic Solar Cells with Nanoscale Structures. *Nanoscale* **2015**, *7*, 15576-15583.
52. Zheng, Y.; Wang, G.; Huang, D.; Kong, J.; Goh, T.; Huang, W.; Yu, J.; Taylor, A. D., Binary Solvent Additives Treatment Boosts the Efficiency of PTB7:PCBM Polymer Solar Cells to Over 9.5%. *Sol. RRL* **2018**, *2*, 1700144.
53. Guo, X.; Cui, C.; Zhang, M.; Huo, L.; Huang, Y.; Hou, J.; Li, Y., High Efficiency Polymer Solar Cells Based on poly(3-hexylthiophene)/indene-C70 Bisadduct With Solvent Additive. *Energy Environ. Sci.* **2012**, *5*, 7943-7949.
54. Arca, F.; Loch, M.; Lugli, P., Enhancing Efficiency of Organic Bulkheterojunction Solar Cells by Using 1,8-Diiodooctane as Processing Additive. *IEEE J. Photovolt.* **2014**, *4*, 1560-1565.
55. Park, M. S.; Kim, F. S., Synergistic Effects of Processing Additives and Thermal Annealing on Nanomorphology and Hole Mobility of Poly(3-hexylthiophene) Thin Films. *Polymers* **2019**, *11*, 112.
56. Liu, F.; Zhao, W.; Tumbleston, J. R.; Wang, C.; Gu, Y.; Wang, D.; Briseno, A. L.; Ade, H.; Russell, T. P., Understanding the Morphology of PTB7:PCBM Blends in Organic Photovoltaics. *Adv. Energy Mater.* **2014**, *4*, 1301377.
57. Xu, B.; Sai-Anand, G.; Gopalan, A. I.; Qiao, Q.; Kang, S. W., Improving Photovoltaic Properties of P3HT:IC60BA through the Incorporation of Small Molecules. *Polymers* **2018**, *10*, 121.

58. Burkhard, G. F.; Hoke, E. T.; McGehee, M. D., Accounting for Interference, Scattering, and Electrode Absorption to Make Accurate Internal Quantum Efficiency Measurements in Organic and Other Thin Solar Cells. *Adv. Mater.* **2010**, *22*, 3293-3297.
59. Scully, S. R.; McGehee, M. D., Effects of Optical Interference and Energy Transfer on Exciton Diffusion Length Measurements in Organic Semiconductors. *J. App. Phys.* **2006**, *100*, 034907.
60. Peumans, P., Uchida, S. & Forrest, S., Efficient Bulk Heterojunction Photovoltaic Cells Using Small-Molecular-Weight Organic Thin Films. *Nature* **2003**, *425*, 158-162.
61. Denis E. Markov, E. A., Paul W. M. Blom, Alexander B. Sieval, and Jan C. Hummelen, Accurate Measurement of the Exciton Diffusion Length in a Conjugated Polymer Using a Heterostructure with a Side-Chain Cross-Linked Fullerene Layer. *J. Phys. Chem. A* **2005**, *109*, 5266-5274.
62. Dang, M. T.; Hirsch, L.; Wantz, G.; Wuest, J. D., Controlling the Morphology and Performance of Bulk Heterojunctions in Solar Cells. Lessons Learned from the Benchmark Poly(3-hexylthiophene):[6,6]-phenyl-C61-butyric Acid Methyl Ester System. *Chem. Rev.* **2013**, *113*, 3734-3765.
63. Li, Z.; Xu, X.; Zhang, W.; Meng, X.; Genene, Z.; Ma, W.; Mammo, W.; Yartsev, A.; Andersson, M. R.; Janssen, R. A. J.; Wang, E., 9.0% Power Conversion Efficiency From Ternary All-Polymer Solar Cells. *Energy Environ. Sci.* **2017**, *10*, 2212-2221.
64. Fan, R.; Huai, Z.; Sun, Y.; Li, X.; Fu, G.; Huang, S.; Wang, L.; Yang, S., Enhanced Performance of Polymer Solar Cells Based on PTB7-Th:PC71BM by Doping with 1-Bromo-4-Nitrobenzene. *J. Mater. Chem. C* **2017**, *5*, 10985-10990.

65. Mihailetschi, V. D.; Xie, H. X.; de Boer, B.; Koster, L. J. A.; Blom, P. W. M., Charge Transport and Photocurrent Generation in Poly(3-hexylthiophene): Methanofullerene Bulk-Heterojunction Solar Cells. *Adv. Funct. Mater.* **2006**, *16*, 699-708.
66. Foster, S.; Deledalle, F.; Mitani, A.; Kimura, T.; Kim, K.-B.; Okachi, T.; Kirchartz, T.; Oguma, J.; Miyake, K.; Durrant, J. R.; Doi, S.; Nelson, J., Electron Collection as a Limit to Polymer:PCBM Solar Cell Efficiency: Effect of Blend Microstructure on Carrier Mobility and Device Performance in PTB7:PCBM. *Adv. Energy Mater.* **2014**, *4*, 1400311.
67. MacKenzie, R. C. I.; Kirchartz, T.; Dibb, G. F. A.; Nelson, J., Modeling Nongeminate Recombination in P3HT:PCBM Solar Cells. *J. Phys. Chem. C* **2011**, *115*, 9806-9813.
68. MacKenzie, R. C. I.; Balderrama, V. S.; Schmeisser, S.; Stoof, R.; Greedy, S.; Pallarès, J.; Marsal, L. F.; Chanaewa, A.; von Hauff, E., Loss Mechanisms in High Efficiency Polymer Solar Cells. *Adv. Energy Mater.* **2016**, *6*, 1501742.
69. Gao, Y.; MacKenzie, R. C. I.; Liu, Y.; Xu, B.; van Loosdrecht, P. H. M.; Tian, W., Engineering Ultra Long Charge Carrier Lifetimes in Organic Electronic Devices at Room Temperature. *Adv. Mater. Interfaces* **2015**, *2*, 1400555.
70. MacKenzie, R. C. I.; Shuttle, C. G.; Chabynyc, M. L.; Nelson, J., Extracting Microscopic Device Parameters from Transient Photocurrent Measurements of P3HT:PCBM Solar Cells. *Adv. Energy Mater.* **2012**, *2*, 662-669.
71. Izawa, S.; Nakano, K.; Suzuki, K.; Hashimoto, K.; Tajima, K., Dominant Effects of First Monolayer Energetics at Donor/Acceptor Interfaces on Organic Photovoltaics. *Adv. Mater.* **2015**, *27*, 3025-3031.
72. Nakano, K.; Suzuki, K.; Chen, Y.; Tajima, K., Roles of Energy/Charge Cascades and Intermixed Layers at Donor/Acceptor Interfaces in Organic Solar Cells. *Sci. Rep.* **2016**, *6*, 29529.

73. Schlenker, C. W.; Barlier, V. S.; Chin, S. W.; Whited, M. T.; McAnally, R. E.; Forrest, S. R.; Thompson, M. E., Cascade Organic Solar Cells. *Chem. Mater.* **2011**, *23*, 4132-4140.
74. Sista, S.; Yao, Y.; Yang, Y.; Tang, M. L.; Bao, Z., Enhancement in Open Circuit Voltage Through a Cascade-Type Energy Band Structure. *Appl. Phys. Lett.* **2007**, *91*, 223508.
75. Kinoshita, Y.; Hasobe, T.; Murata, H., Control of Open-Circuit Voltage in Organic Photovoltaic Cells by Inserting an Ultrathin Metal-Phthalocyanine Layer. *Appl. Phys. Lett.* **2007**, *91*, 083518.

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